X-Ray Nano-Analytics and Microscopy

Part I

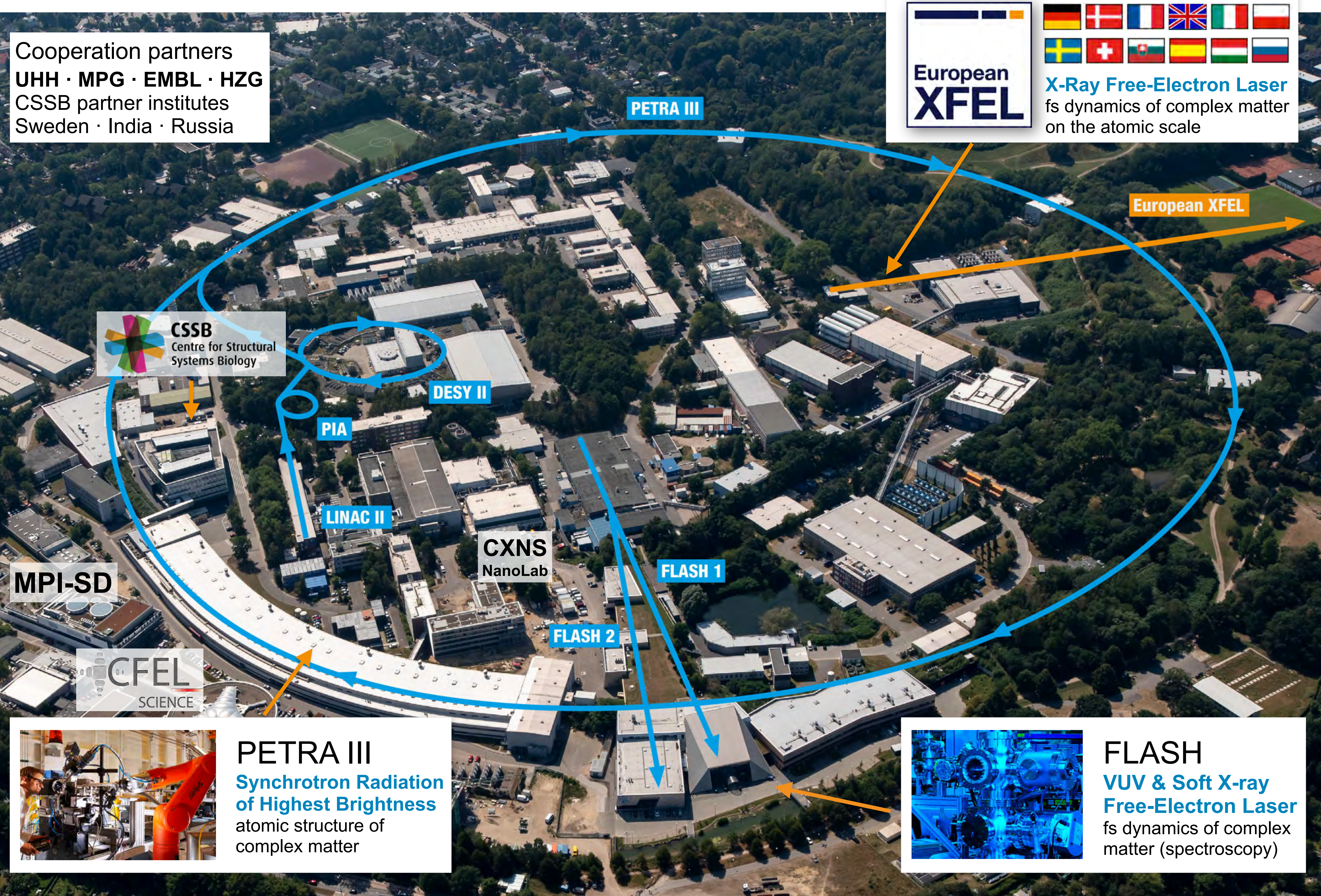
Christian G. Schroer
DESY & Universität Hamburg

HELMHOLTZ RESEARCH FOR GRAND CHALLENGES

Universität Hamburg
DER FORSCHUNG | DER LEHRE | DER BILDUNG

DESY.

DESY: Bright Light for Science



X-ray Scanning Microscopy

Broad field of applications:

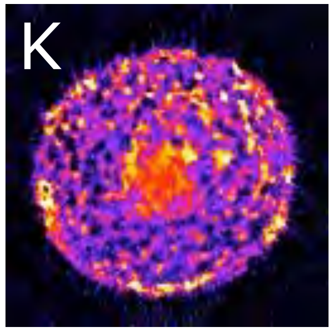
- Main advantage: large penetration depth
  - → *in-situ* and *operando* studies
  - → 3D bulk analysis without destructive sample preparation
- X-ray analytical contrasts: XRD, XAS, XRF, ...
  - → elemental, chemical, and structural information

Today: „mesoscopic gap“

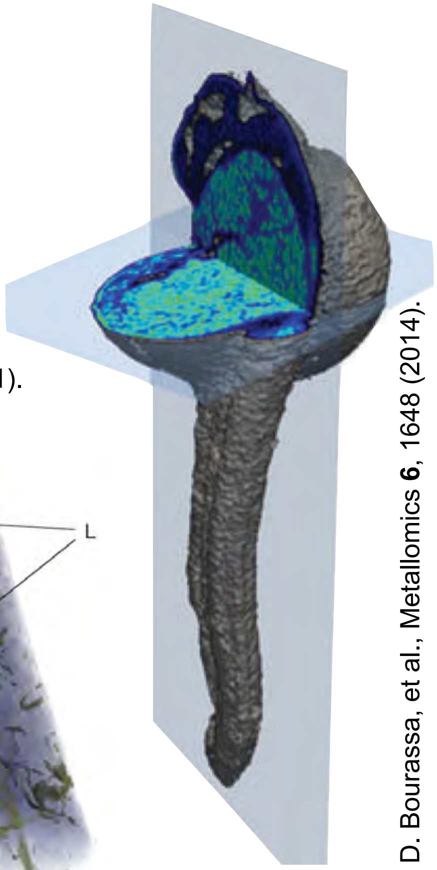
real-space resolution: down to about 10 nm

XRD and XAS: atomic scale

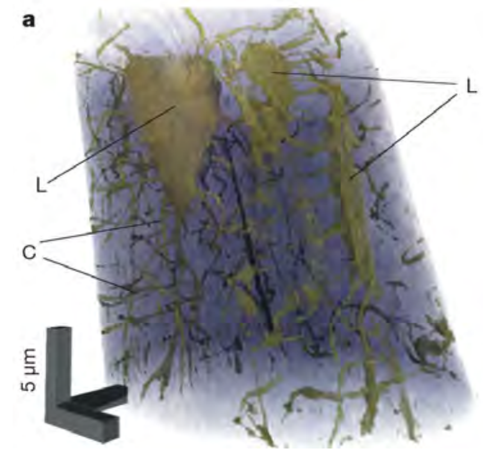
Many interesting physics and chemistry (e. g. catalysis) at the 1 - 10 nm scale!



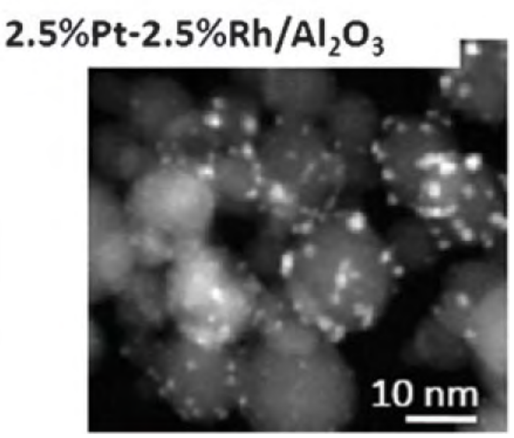
C. G. Schroer, APL **79**, 1912 (2001).



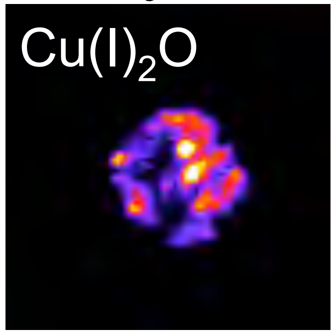
D. Bourassa, et al., Metallomics **6**, 1648 (2014).



M. Dierolf, et al., Nature **467**, 436 (2010).



catalysts



C. G. Schroer, et al., APL **82**, 3360 (2003).

X-ray Microscopy

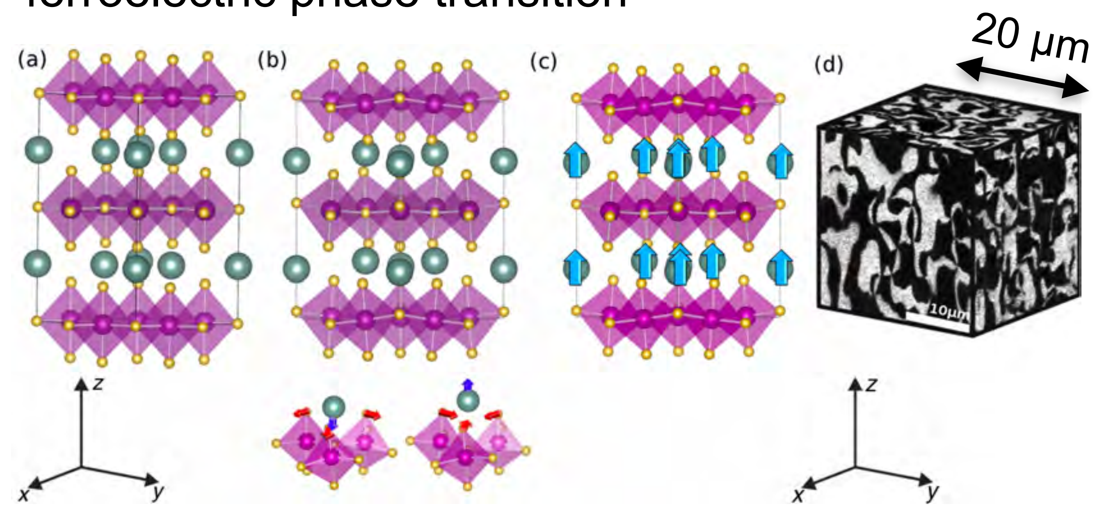
Many interesting physics and chemistry questions:

investigate local states:

- > individual defects (0D): changes in electron density, charge ordering
- > (structural) domain boundaries (2D), e. g., in multiferroics
- > mesoscopic dynamics at (solid-state) phase transitions
- > catalytic nanoparticles (under reaction conditions)
- > ...

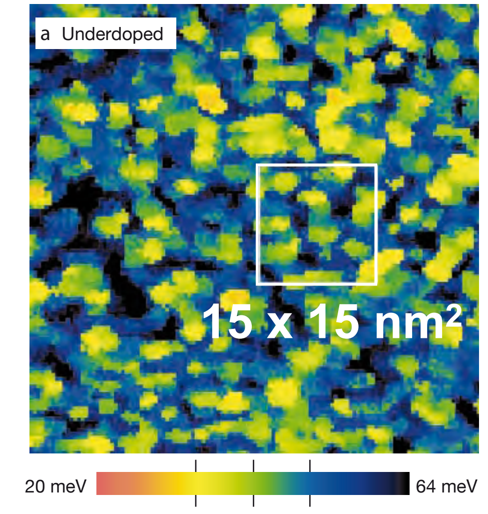
Mesoscale also very important for nanotechnology (e. g., defects in devices)!

ferroelectric phase transition



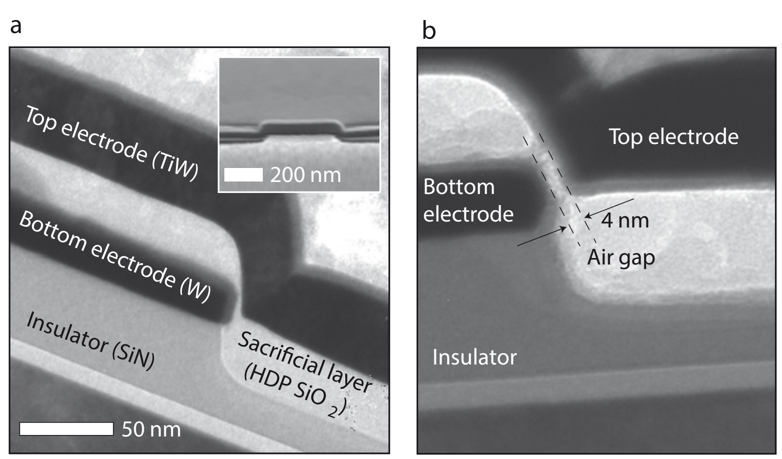
Griffin, et al., PRX **2**, 041022 (2012).

variation of supercond. gap



Lang, et al., Nature **415**, 412 (2002).

nanoelectromechanical switch



Lee, et al., Nature Nanotech. **8**, 36 (2012).

Current State of X-Ray Microscopy

Conventional X-ray microscopy

→ optics limit spatial resolution: diffraction limit

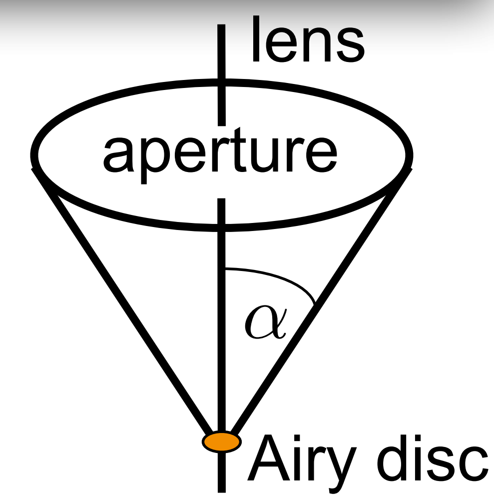
$$d = \frac{\lambda}{2n \sin\alpha}$$

(typically: a few tens of nanometers)

optics are technology limited!

Theoretical extrapolation of X-ray optical performance to the atomic level.

[PRB **74**, 033405 (2006); H. Yan, et al., PRB **76**, 115438 (2007)]



Coherent X-ray imaging techniques (CXDI, ptychography)

→ no imaging optics needed!

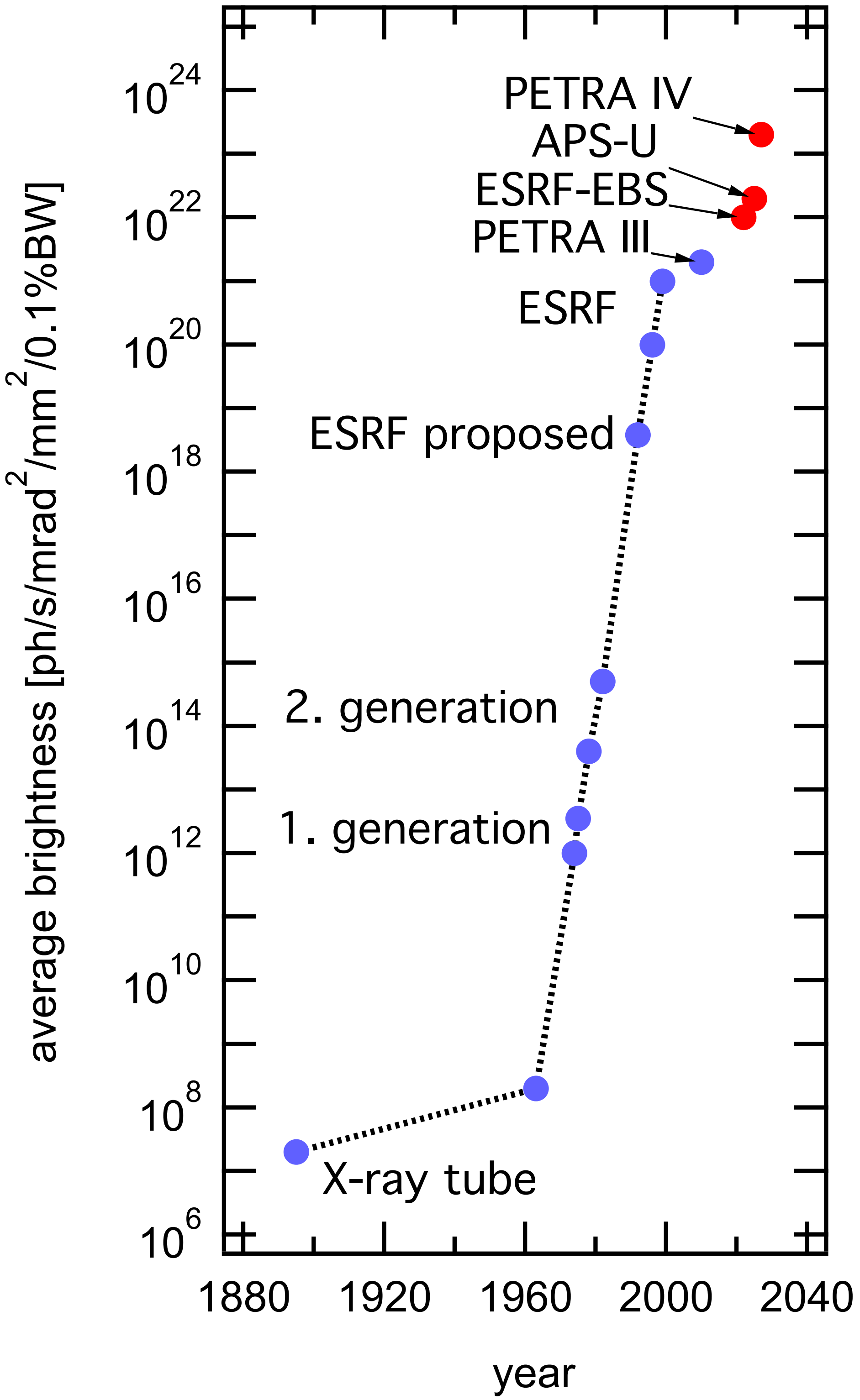
→ limited by statistics of far-field diffraction patterns ...

highest resolution: a few nanometers, focusing coherent beam

[PRL **101**, 090801 (2008); Y. Takahashi, et al., PRB **80**, 054103 (2009);
A. Schropp, et al., APL 100, 253112 (2012)]

Spectral Brightness

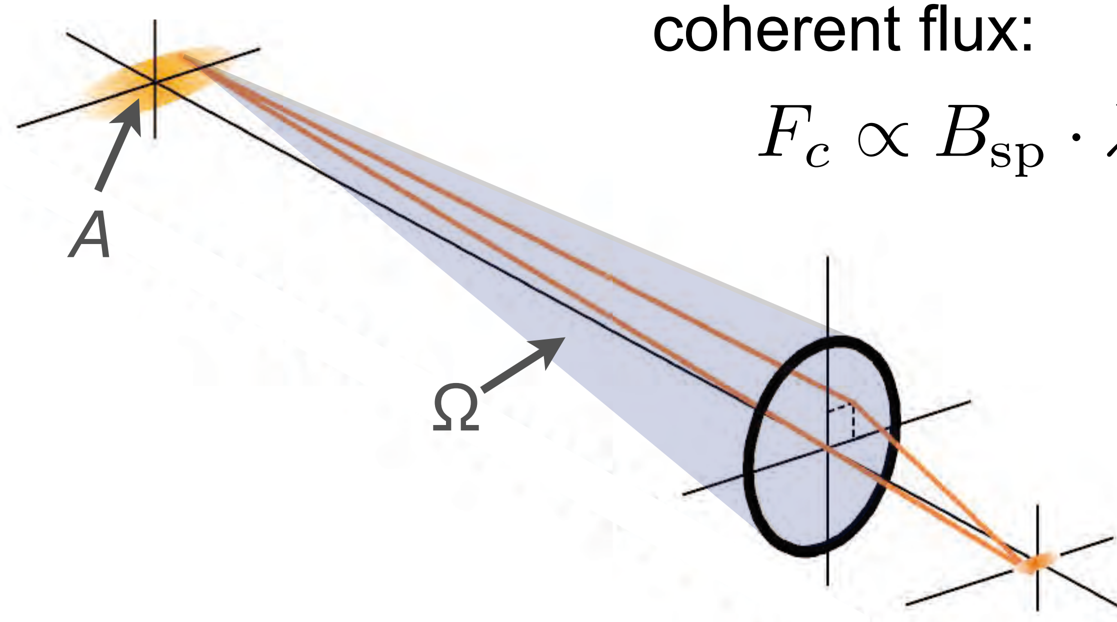
10000x more light per decade (since 1965)



Spectral brightness:

$$B_{\mathrm{sp}} = \frac{F}{\Omega \cdot A \cdot \Delta E/E}$$

Flux per phase-space volume



coherent flux:

$$F_c \propto B_{\mathrm{sp}} \cdot \lambda^2 \cdot \frac{\Delta E}{E}$$

Improvements in brightness:

- > faster measurements (time resolution)
- > nano-imaging (spatial resolution)
- > spectroscopy (energy resolution)

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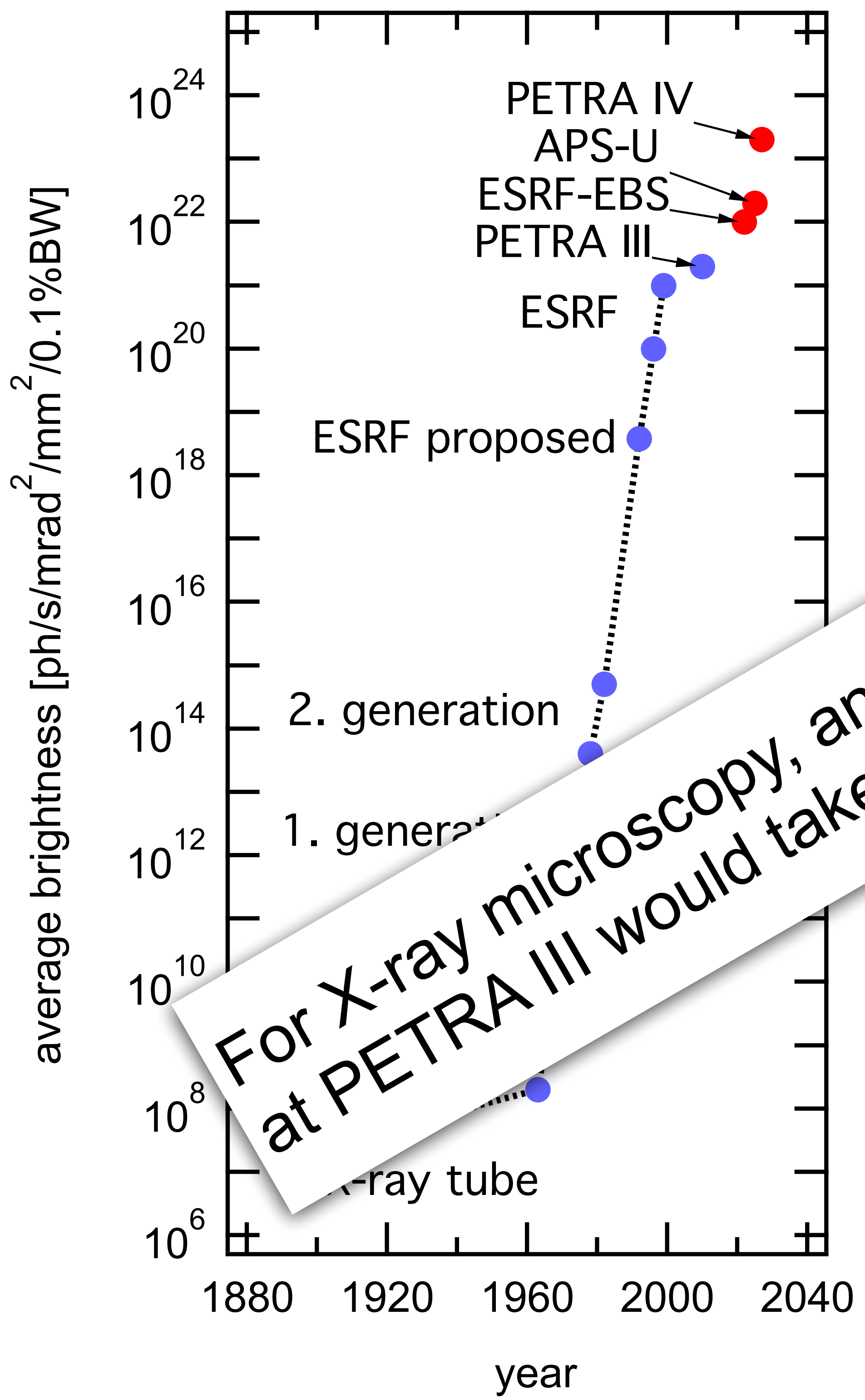


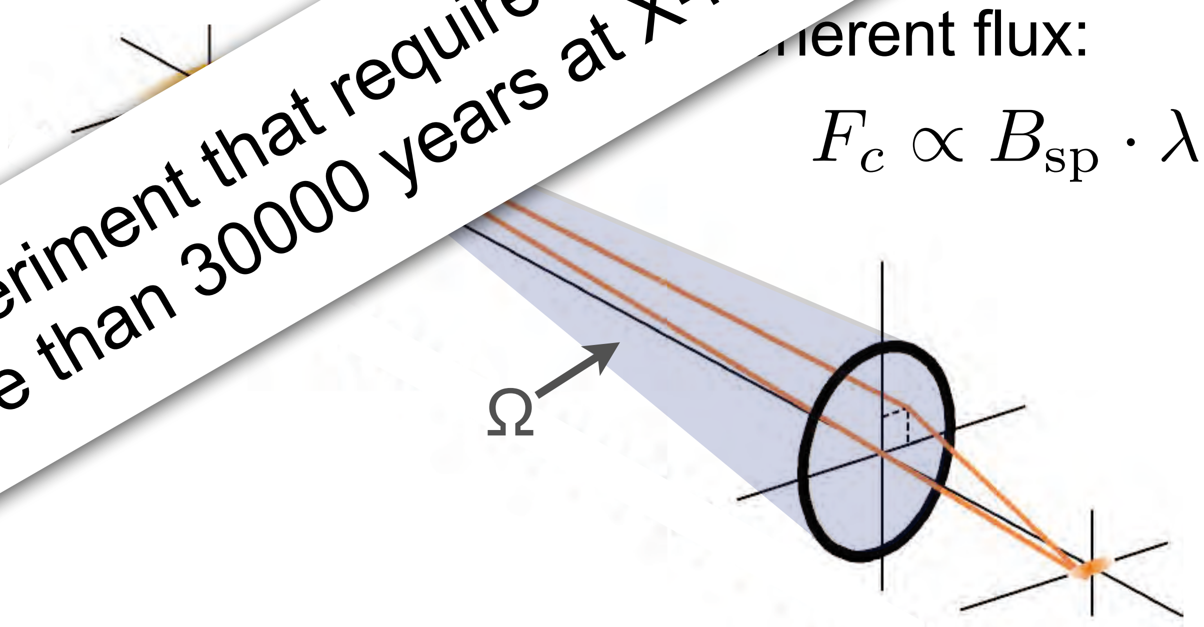
Spectral brightness:

$$B_{\mathrm{sp}} = \frac{F}{\Omega \cdot \ldots}$$

Flux per phase ...

... erent flux:

$$F_c \propto B_{\mathrm{sp}} \cdot \lambda^2 \cdot \frac{\Delta E}{E}$$



For X-ray microscopy, an experiment that requires 1 s of exposure time at PETRA III would take more than 30000 years at X-ray tube!!!

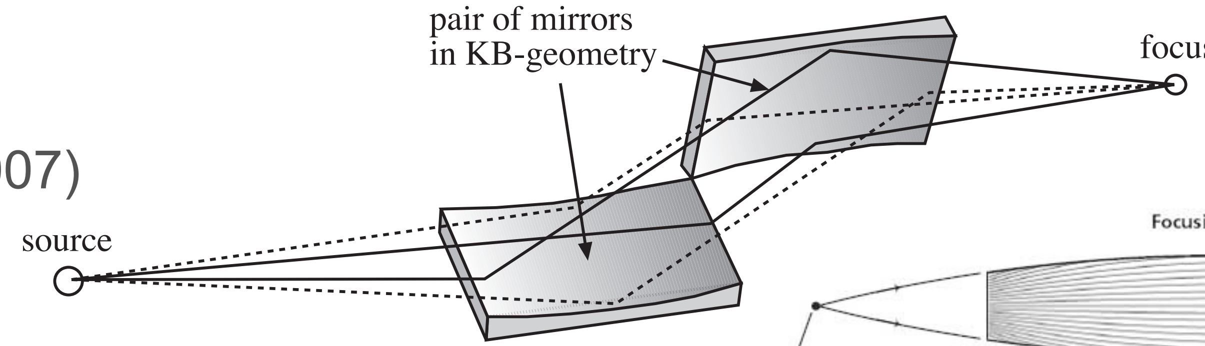
Improvements in brightness:

- > faster measurements (time resolution)
- > nano-imaging (spatial resolution)
- > spectroscopy (energy resolution)

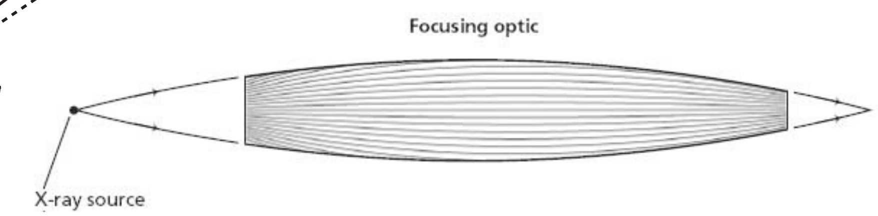
Nanofocusing Optics

reflection:

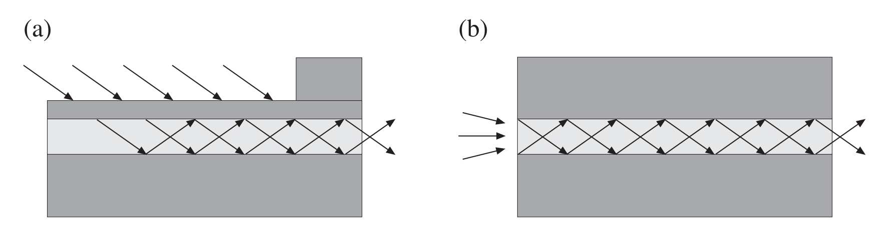
> mirrors (25 nm)
H. Mimura, et al., APL **90**, 051903 (2007)



> capillaries

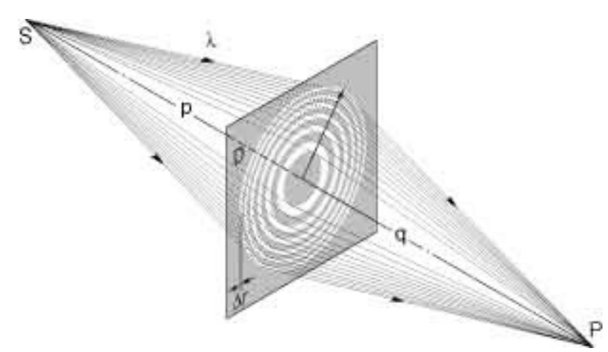


> wave guides (~10 nm)
S. P. Krüger, et al., J. Synchrotron Rad. **19**, 227 (2012)

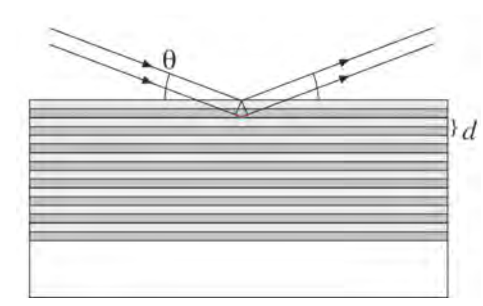


diffraction:

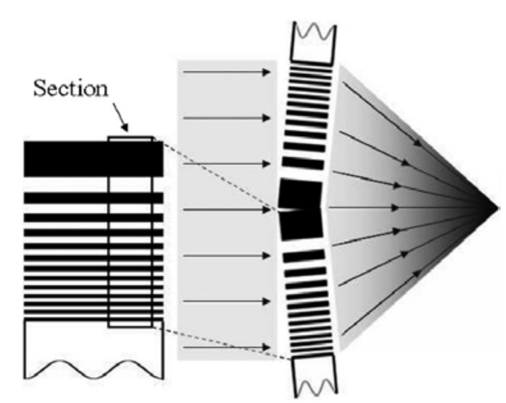
> Fresnel zone plates (< 10 nm)
J. Vila-Comamala, et al., Ultramic. **109**, 1360 (2009)



> multilayer mirrors (7 nm)
H. Mimura, et al., Nat. Phys. **6**, 122 (2010)



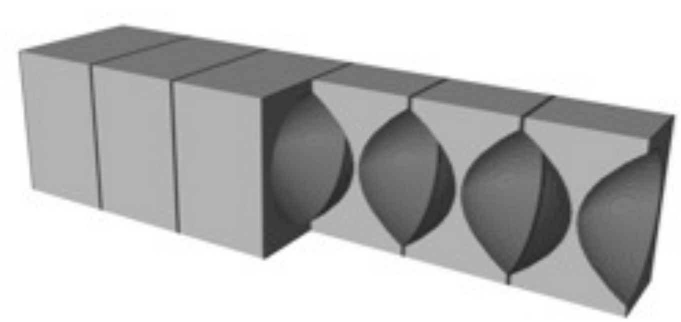
> multilayer Laue lenses (8 x 7 nm)
S. Bajt, et al., Light: Sci. & App. **7**, 17162 (2018)



> bent crystals

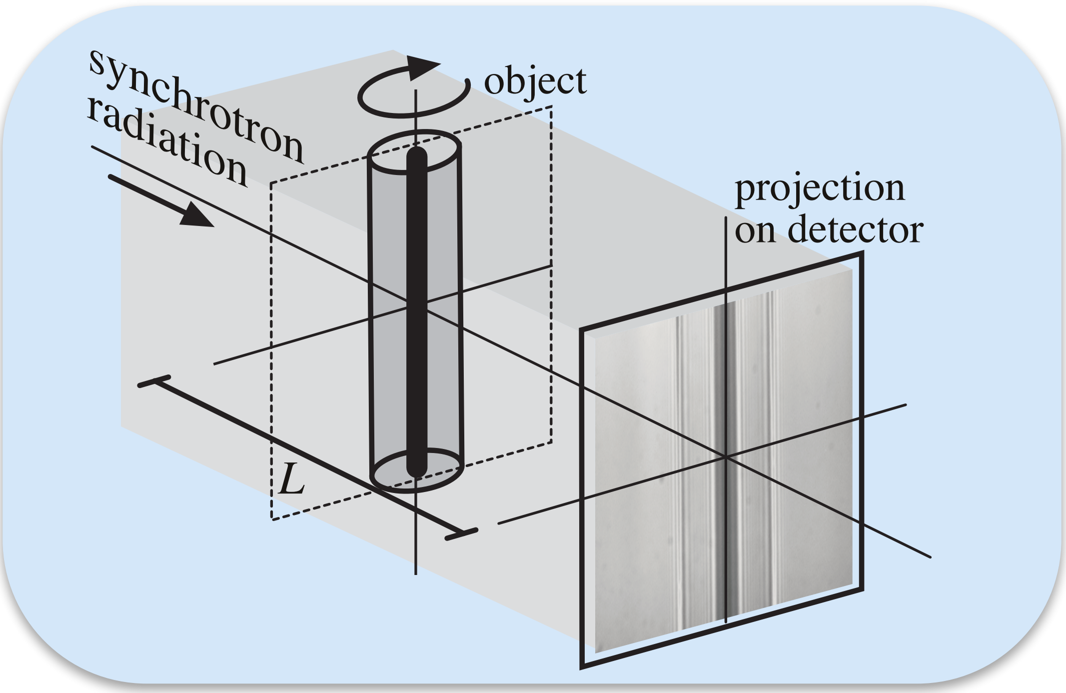
refraction:

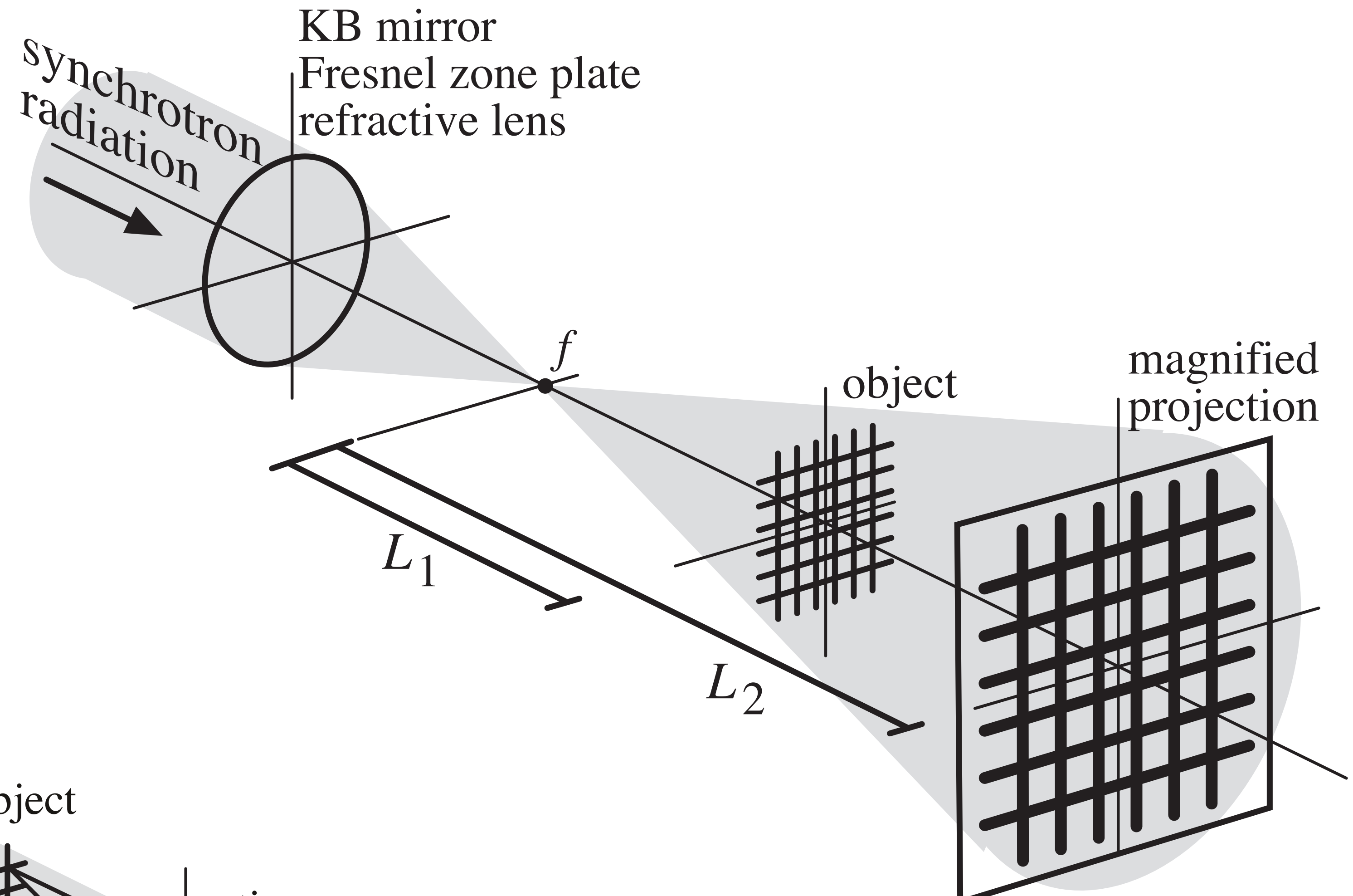
> lenses (43 nm, 18 nm)
C. G. Schroer, et al., AIP Conf. Ser. **1365**, 227 (2011)
J. Patommel, et al., APL **110**, 101103 (2017)



X-Ray Microscopy Techniques: Full-Field Imaging

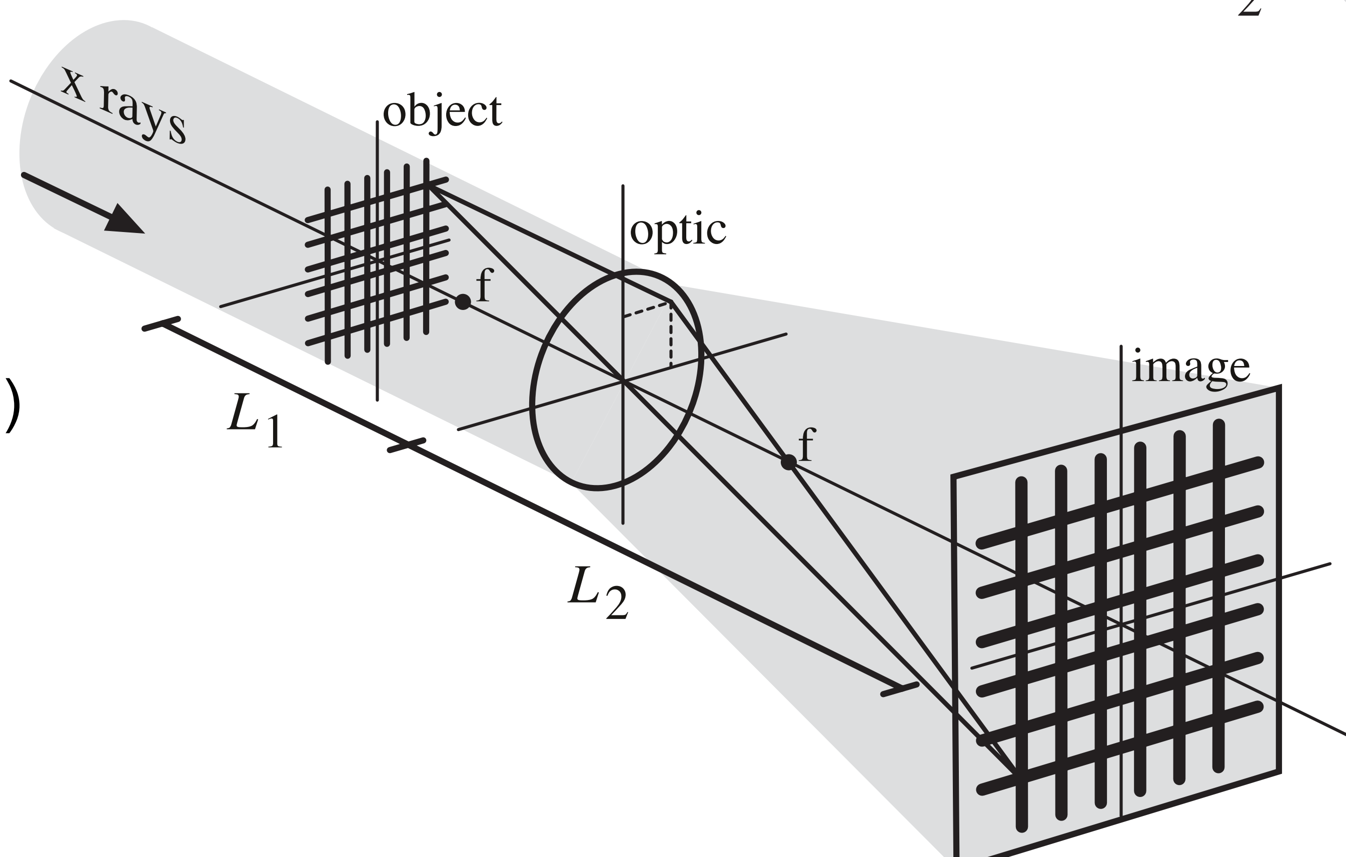
Projection imaging:





Imaging through objective lens:

x-rays focused by condensor (aperture matching)



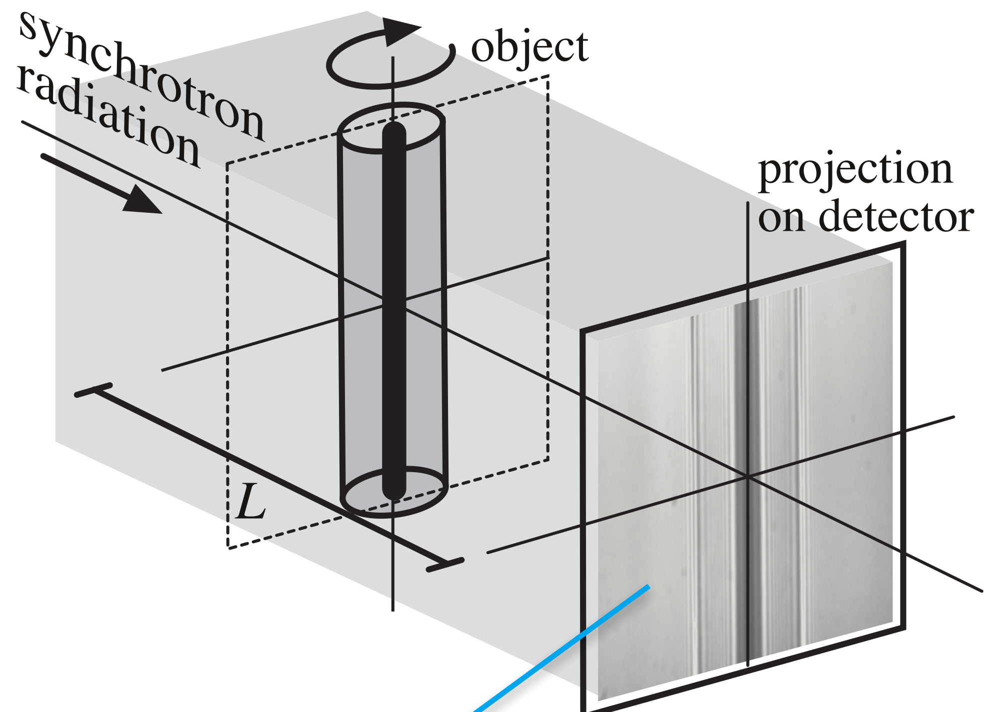
Example: Projection Imaging (Phase Contrast)

Example: plant physiology

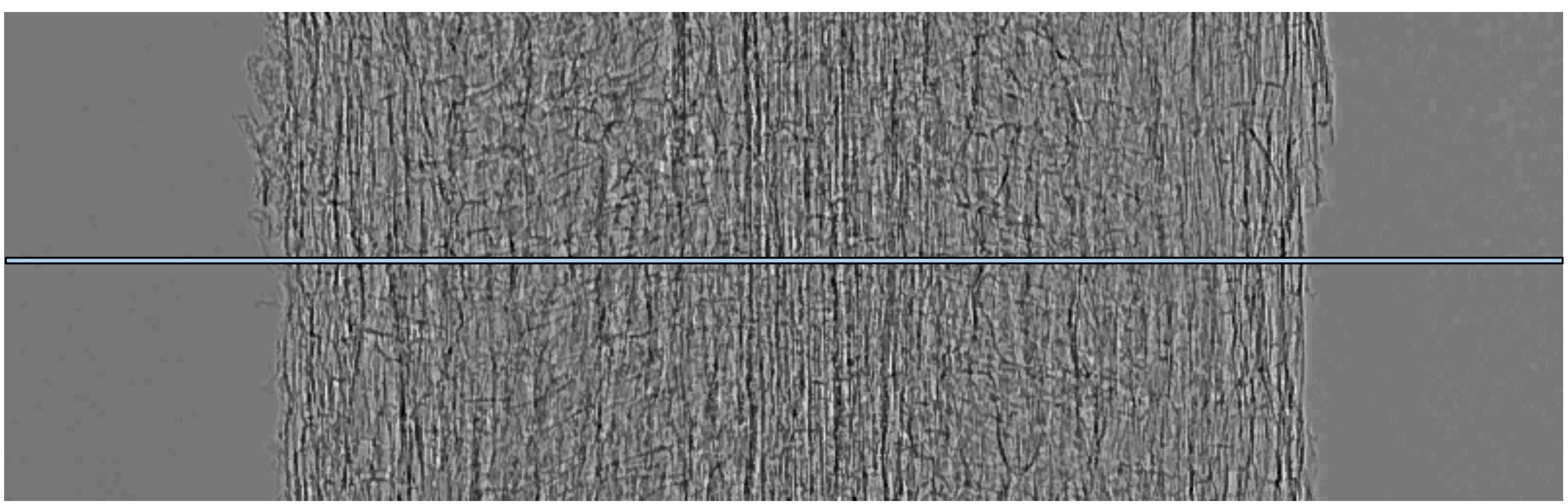
0.5 mm

root of mahogany tree
(W. H. Schröder, FZ Jülich)

Reveal inner structures of object:



1250 projections over 180°



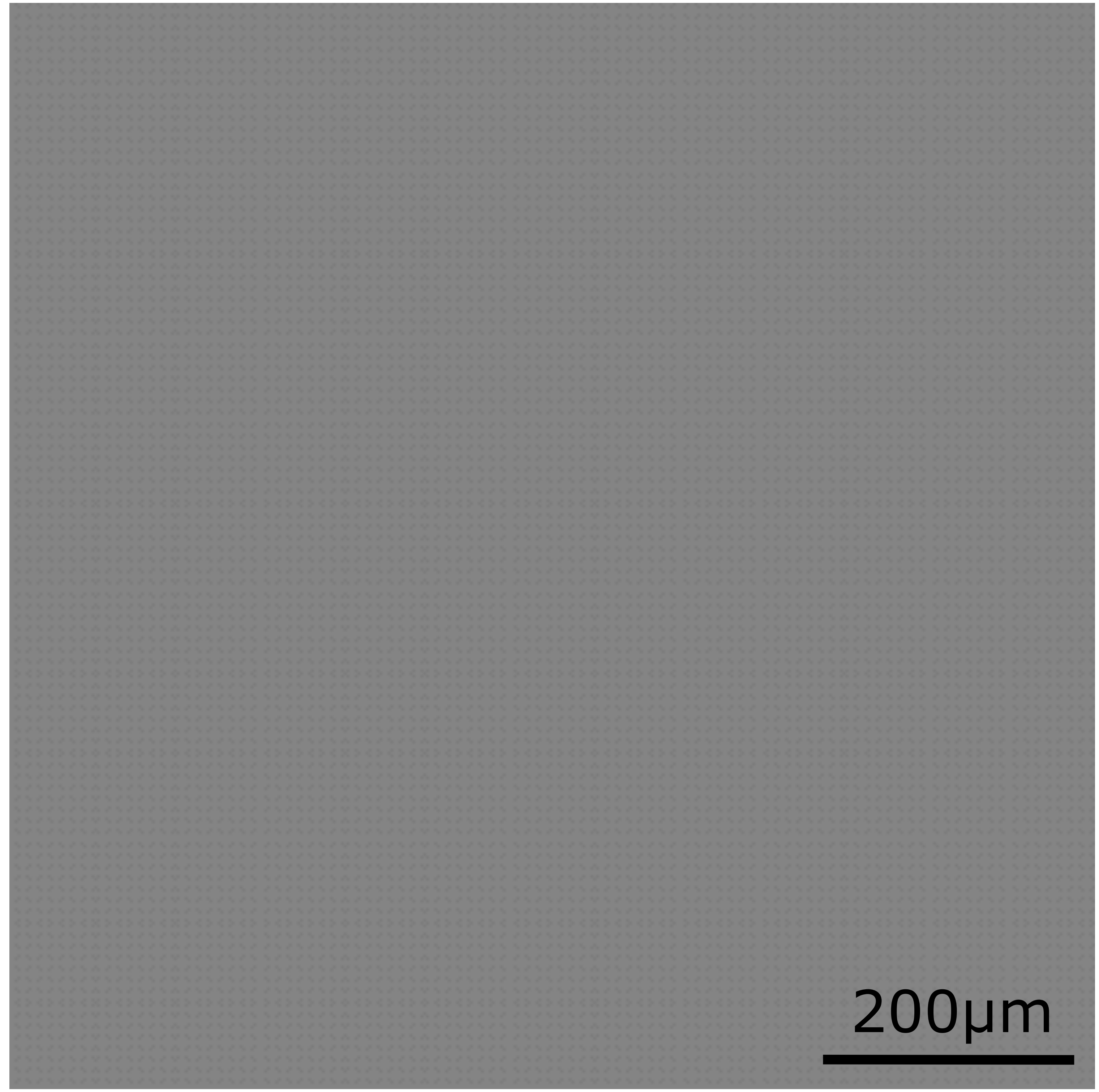
energy: 20 keV
L = 50 mm
pixel size: 1.4 µm

Tomographic Reconstruction

Single slice:

reconstructed from
1250 projections

root of mahogany tree
(W. H. Schröder, FZ Jülich)

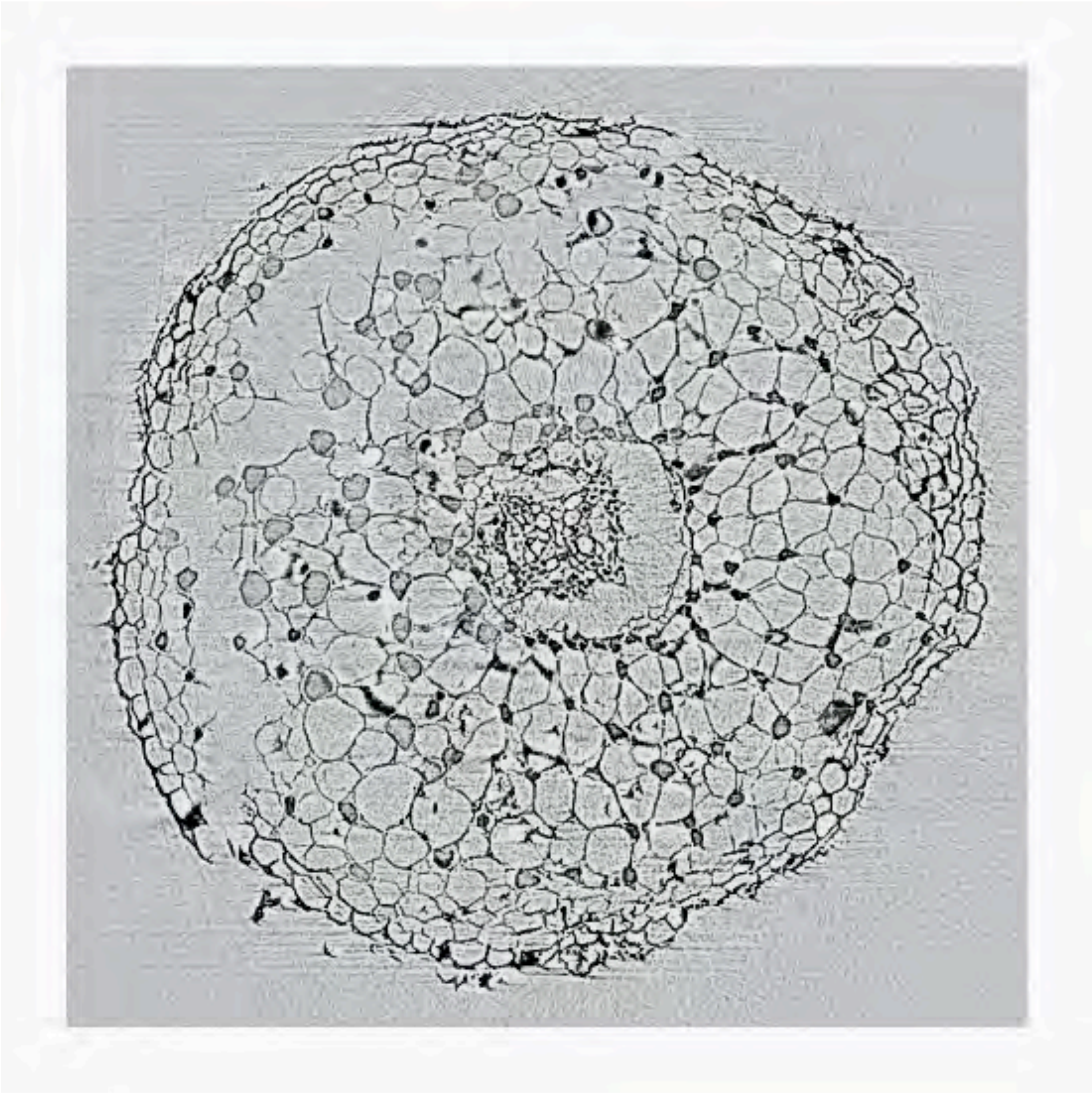


3D Reconstruction

Many slices:

3D structure

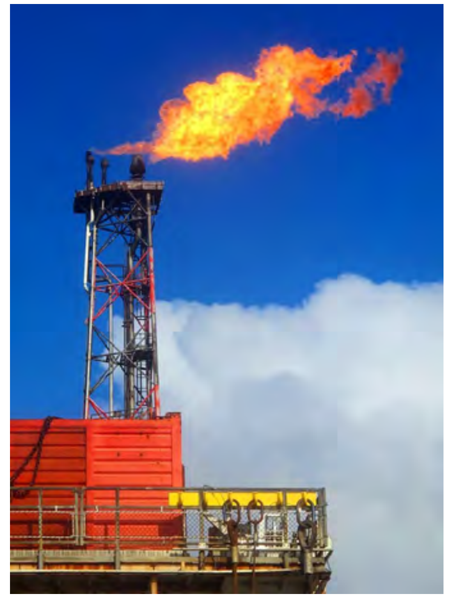
root of mahogany tree
(W. H. Schröder, FZ Jülich)



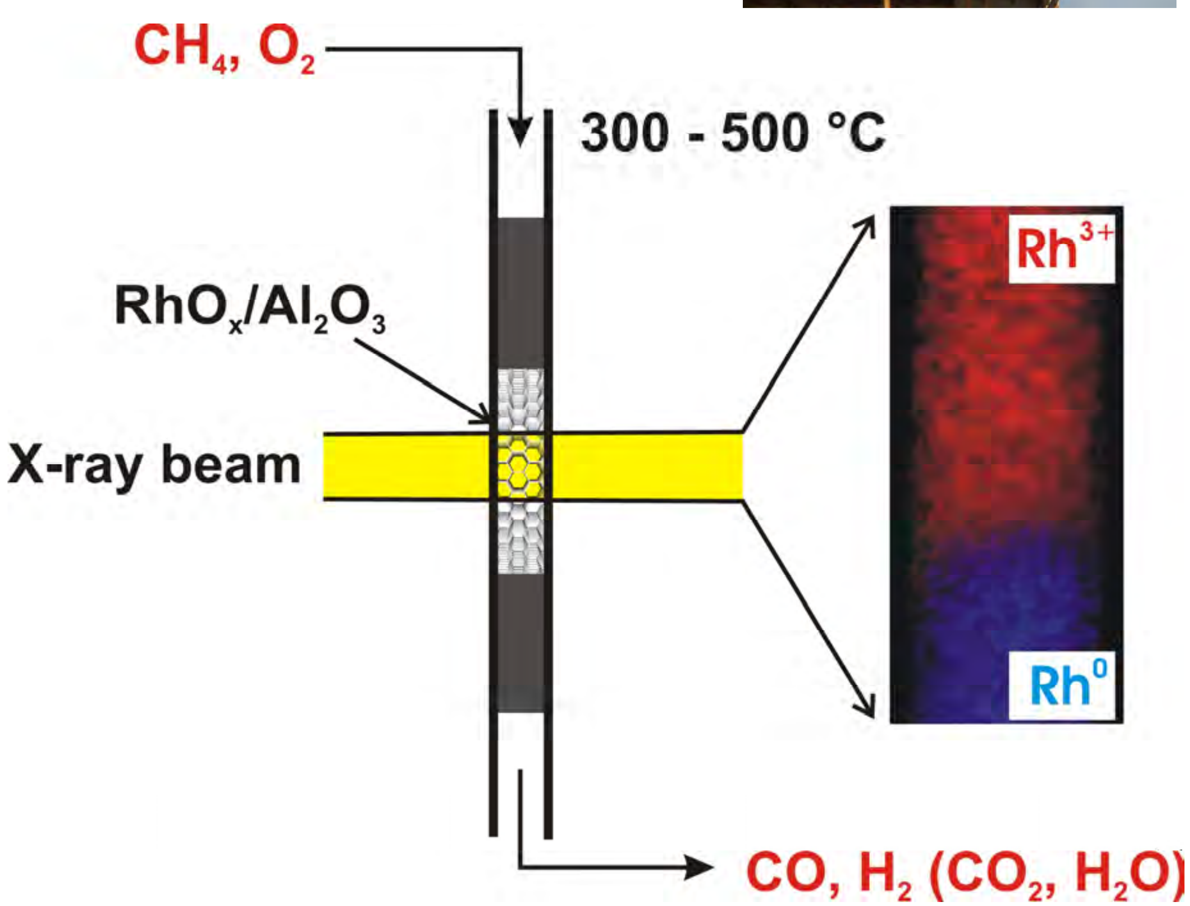
resolution: ~ 3 µm

Visualize Catalysts in Action

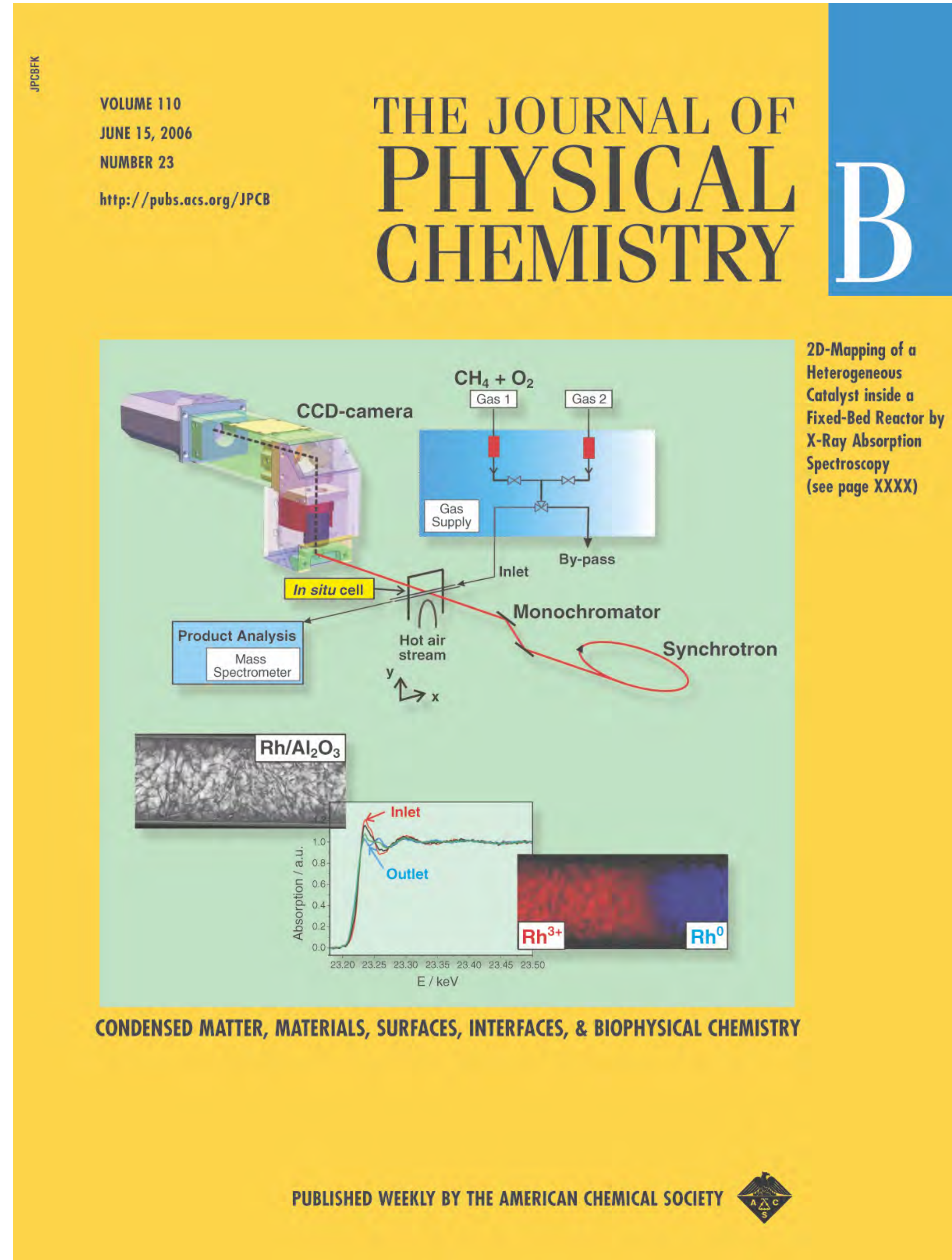
Methane often wasted during oil production:



First step to convert methane into liquid fuels (syngas production):



Grunwaldt, et al.,
J. Chem. Phys. B **110,** 8674 (2006)



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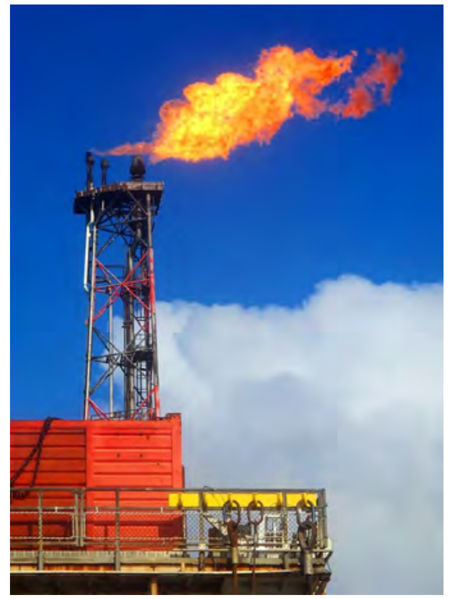

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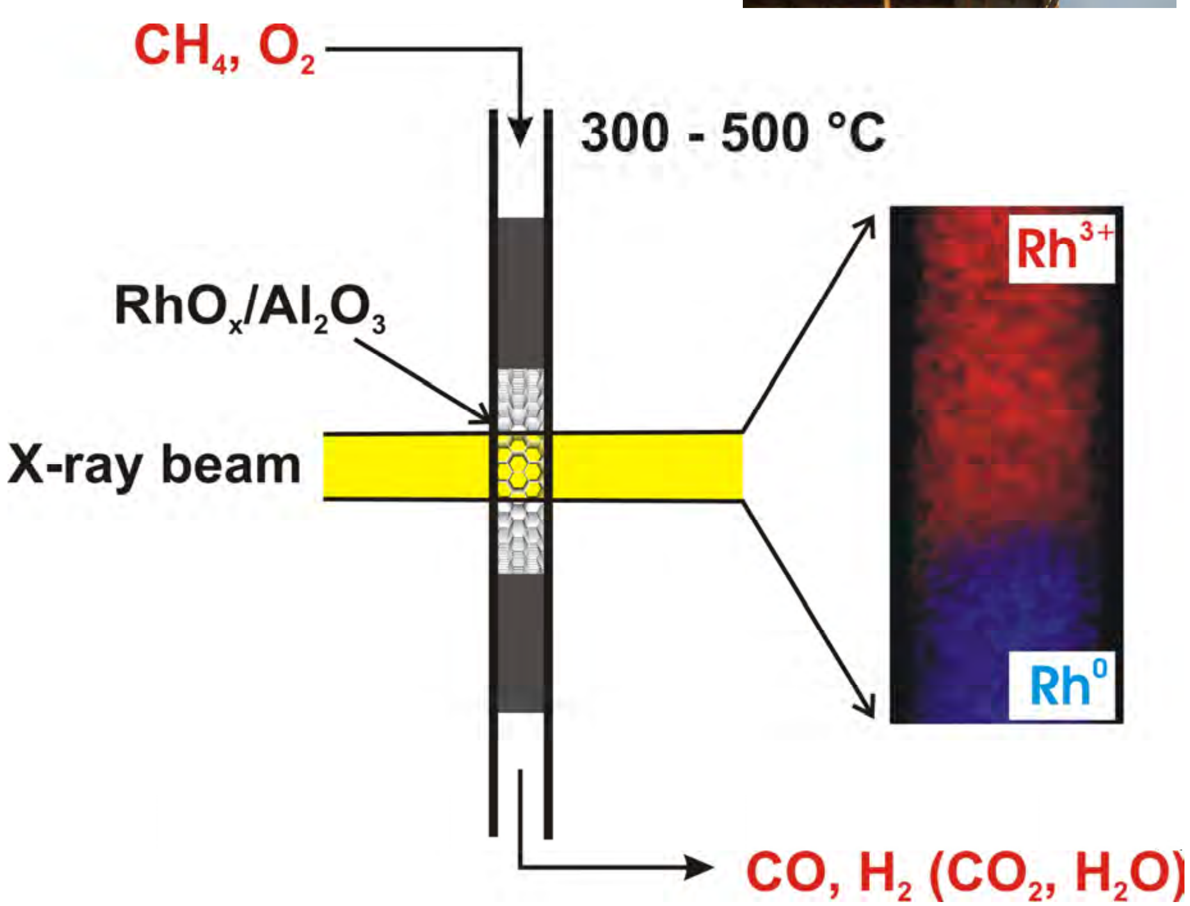

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Combustion of methane:

$$CH_4 + 2O_2 \longrightarrow CO_2 + 2H_2O$$

(exothermal: -801,7kJ/mol)

reforming of methane to H_2:

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potentially other reaction:
direct partial oxidation:

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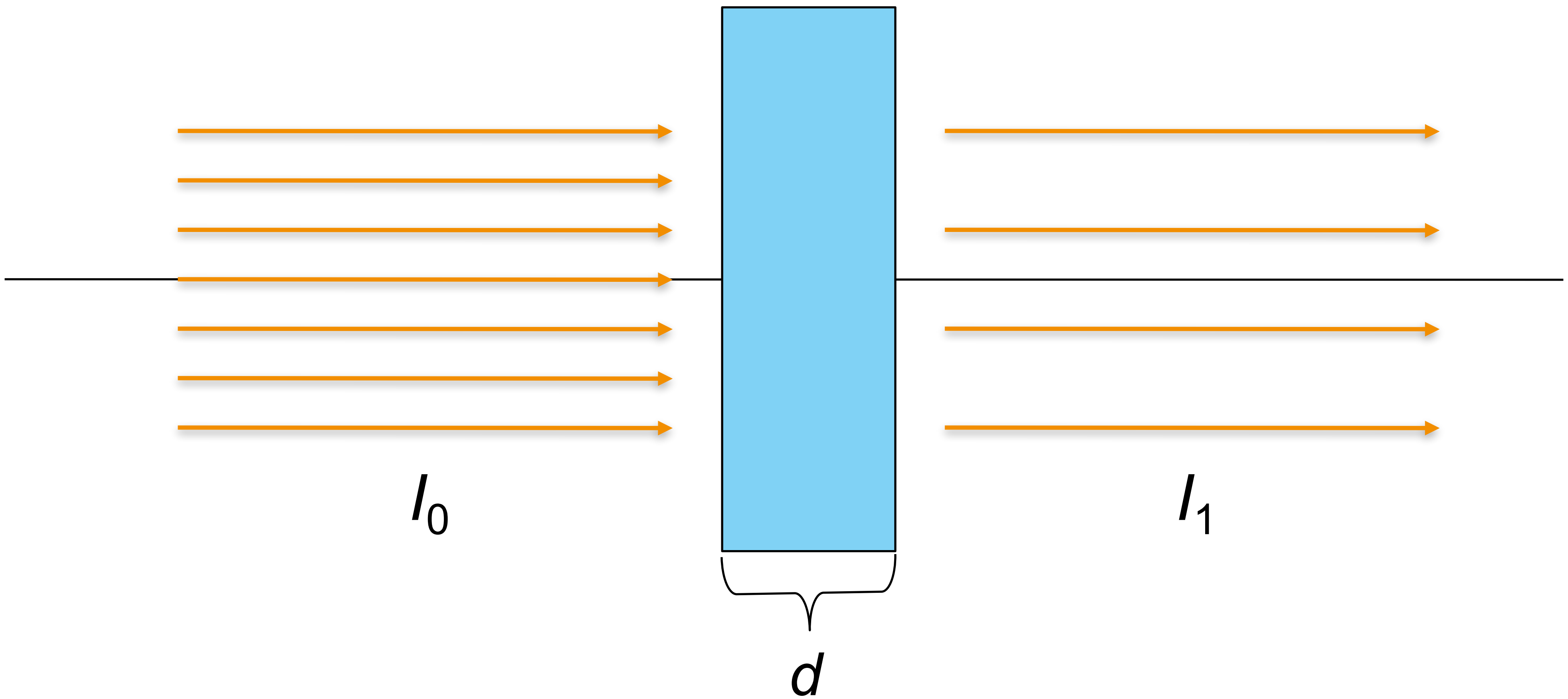
X-Ray Absorption: Lambert-Beer Law

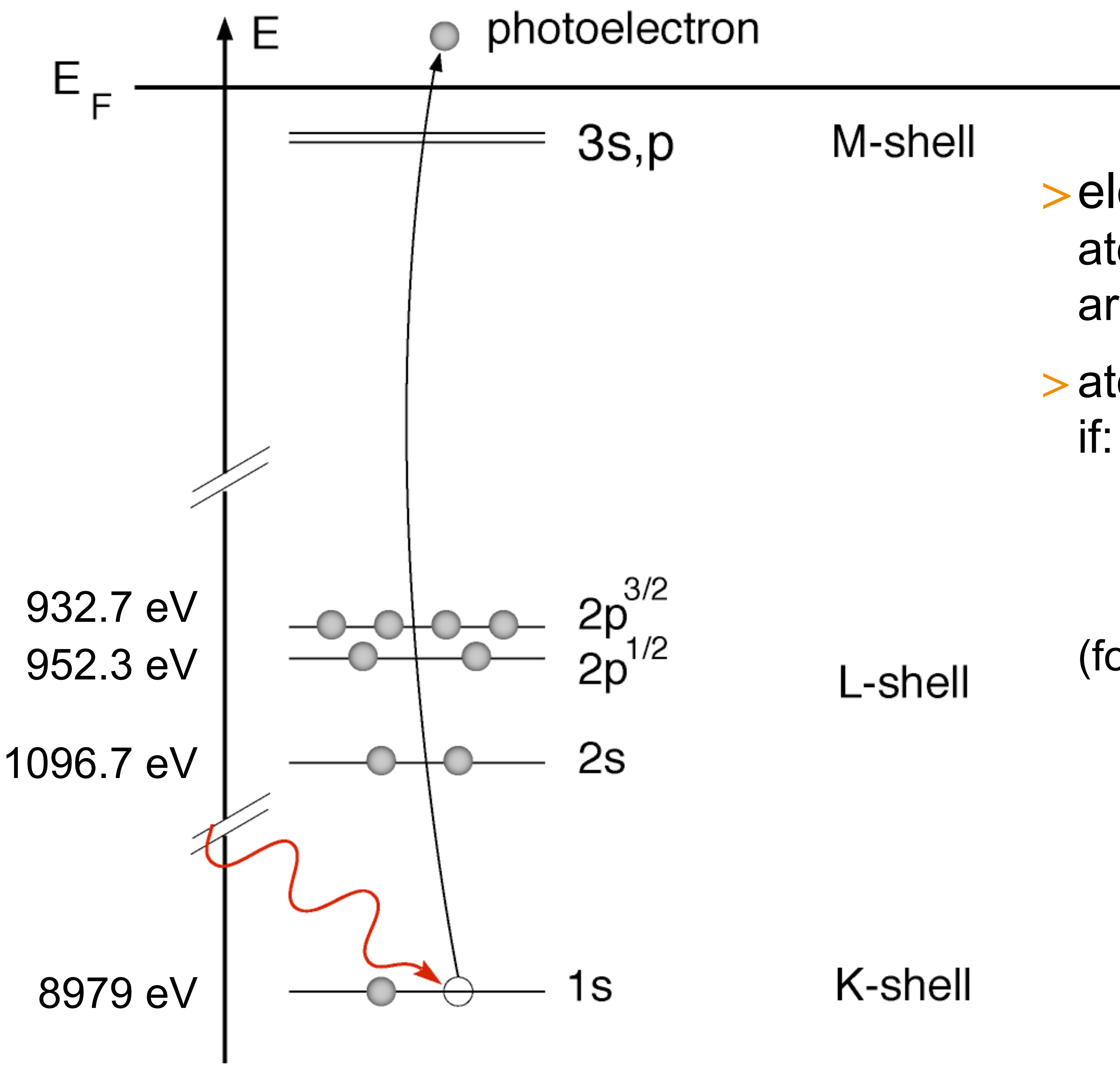


$$I_1(E) = I_0(E) \cdot \exp\left[-\mu(E) d\right]$$

μ(*E*): linear attenuation coefficient

$$\mu(E) \cdot d = \ln\left(\frac{I_0}{I_1}\right)$$

Photo Absorption



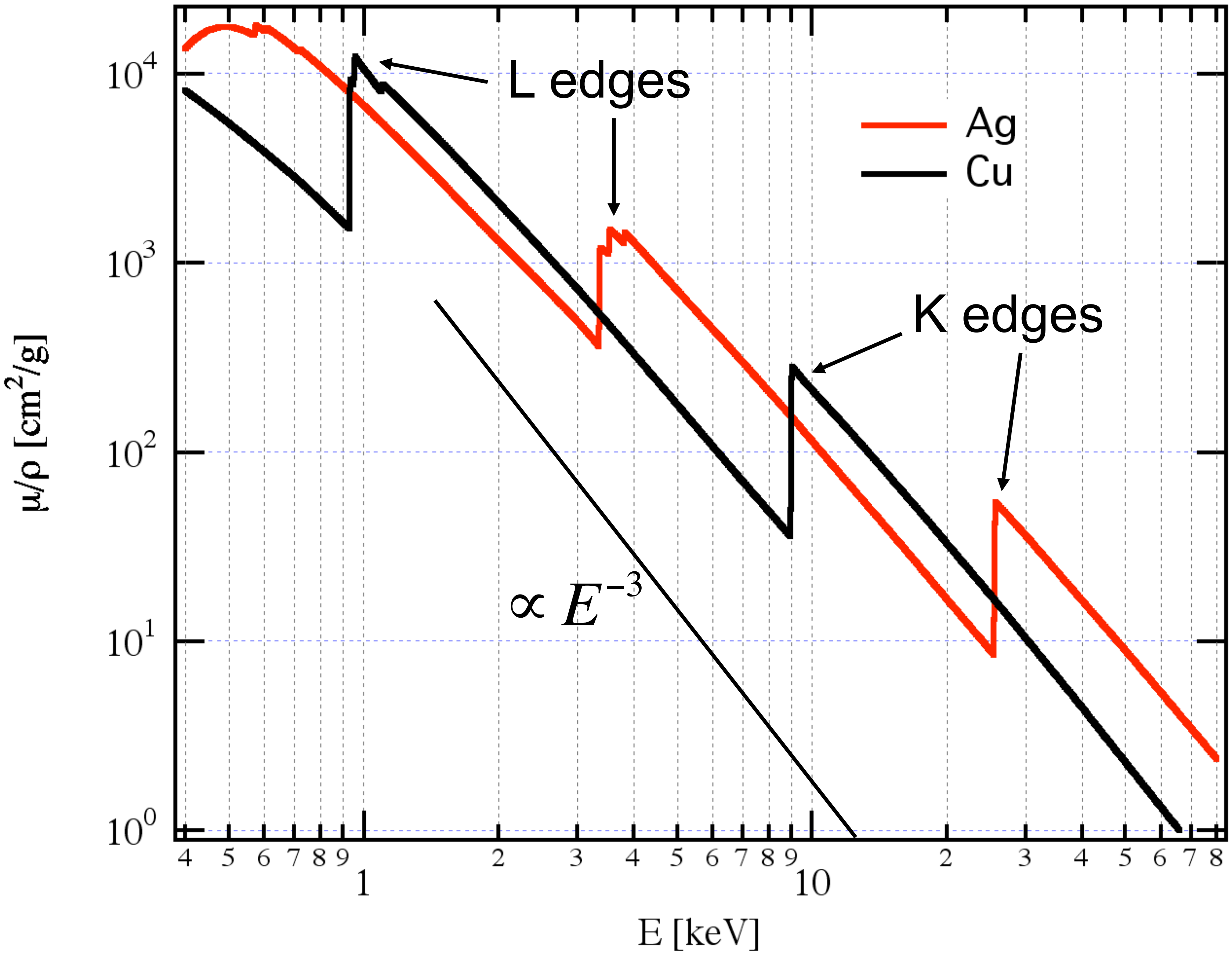
- electrons populate specific atomic states: binding energies are atom specific: K, L, M
- atom can absorb x-ray photon if:

$$E_{\mathrm{photon}} > E_{\mathrm{ionization}}$$

(follows from Pauli principle)

Example: Absorption in Cu & Ag

μ(E): linear attenuation coefficient



- mainly atomic effect
- strong dependence on x-ray energy:

$$\propto E^{-2.78}$$

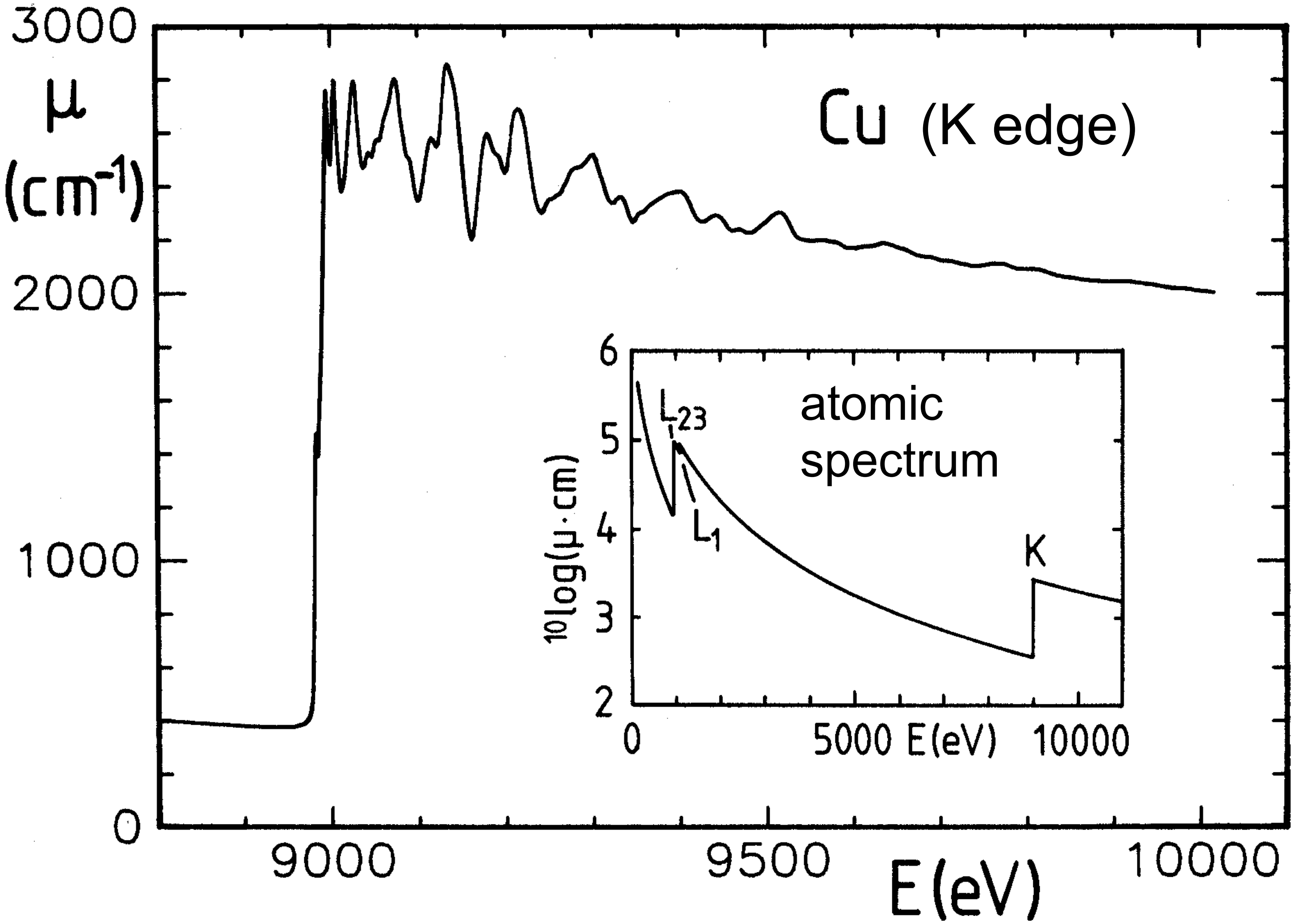
- strong dependence on atomic number:

$$\propto Z^{2.7}$$

- larges contribution from inner shells

Example: Absorption in Cu

μ(E): linear attenuation coefficient



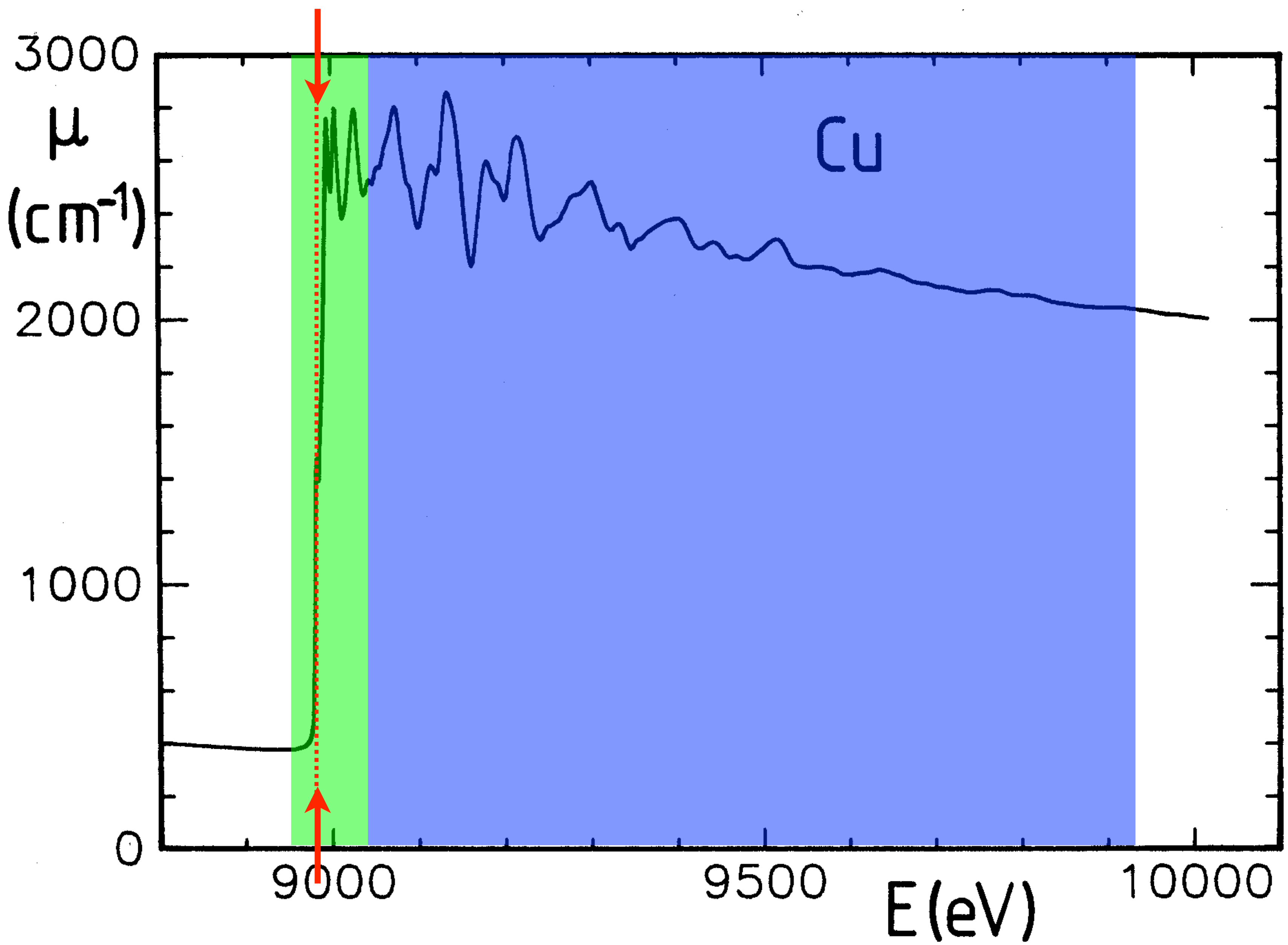
Metallic Cu:

- mainly atomic effect
- fine structure in solid:
  X-ray Absorption Fine Stucture

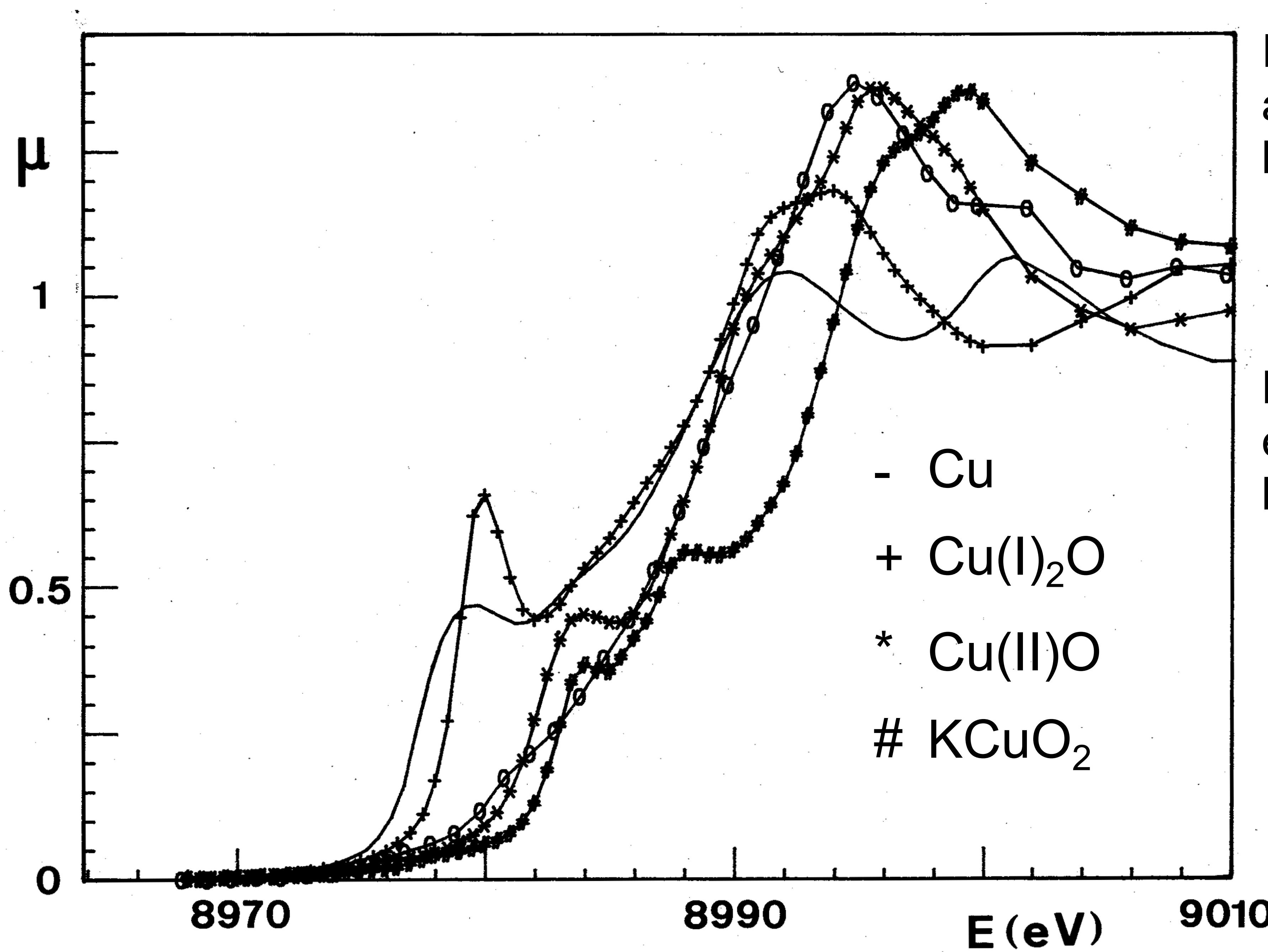
X-ray Absorption Spectrum

Three characteristic features:

- Energy of absorption edge: oxidation state
- Near-edge region: (XANES: x-ray absorption near edge structure) local, projected density of states
- Extended fine structure: (EXAFS: extended x-ray absorption fine structure) local chemical environment of atomic species



Energy of Absorption Edge

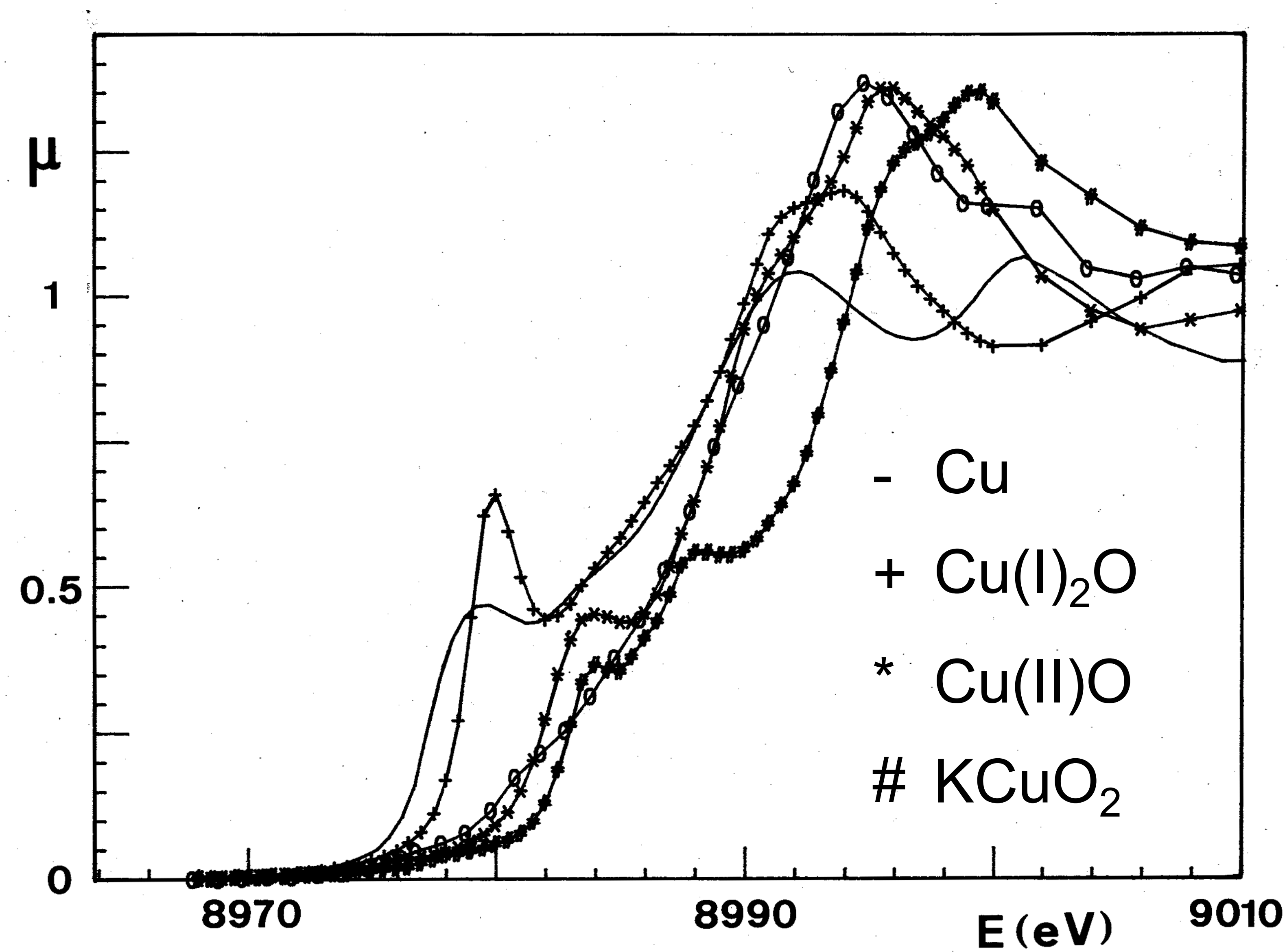


Increasing oxidation state: absorption edge shifts to higher x-ray energies

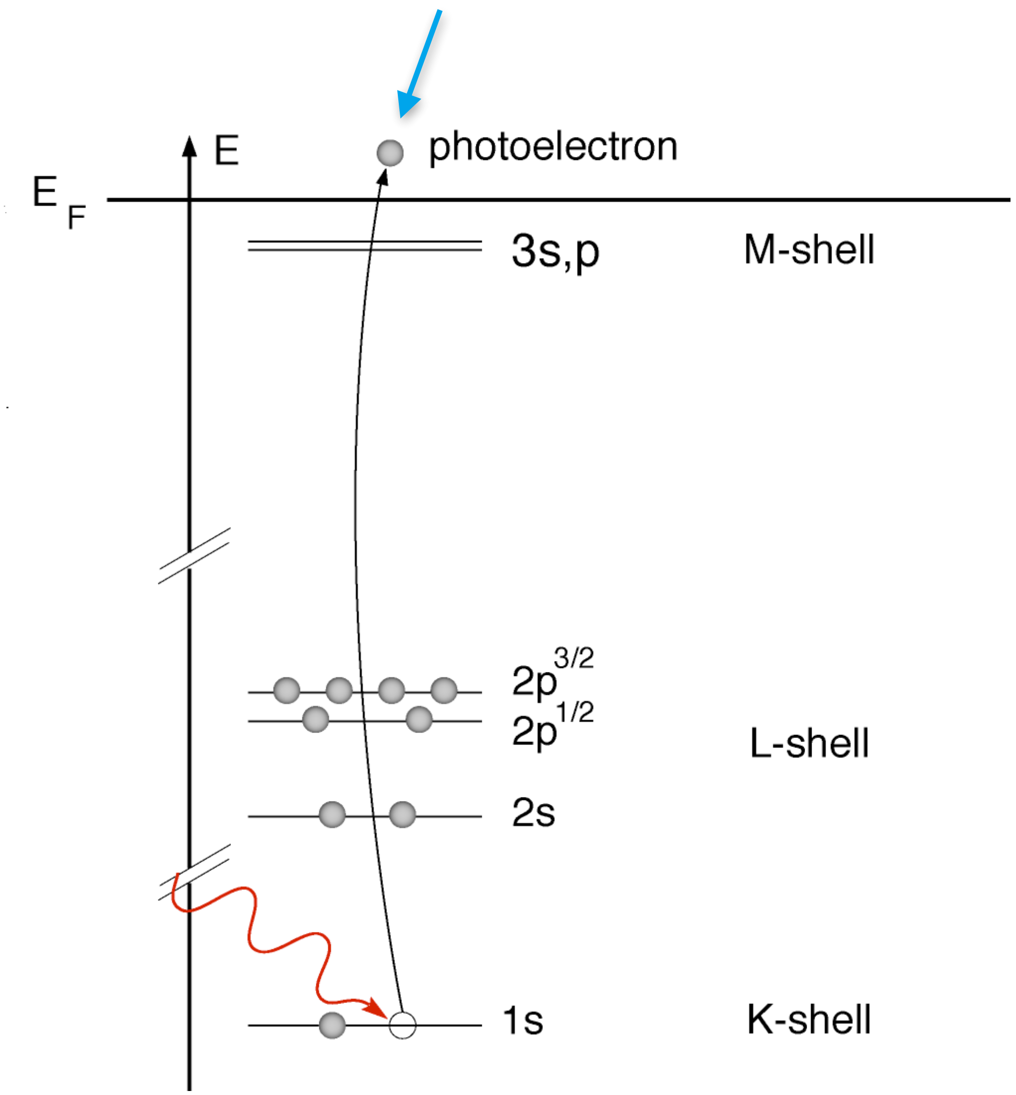
Reduced screening of electric field of nucleus by valence electrons:

other electrons more tightly bound!

Shape of Near-Edge Spectrum



depends on density of states
available to photoelectron



Shape of spectrum:

> can be modeled by methods in theoretical solid state physics
> can be used as „fingerprint" to identify a given chemical environment

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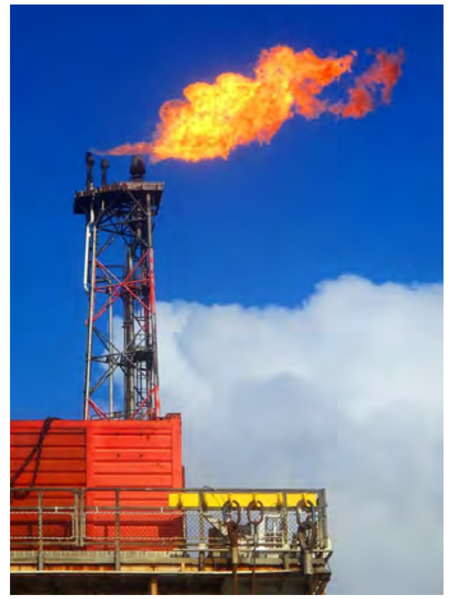

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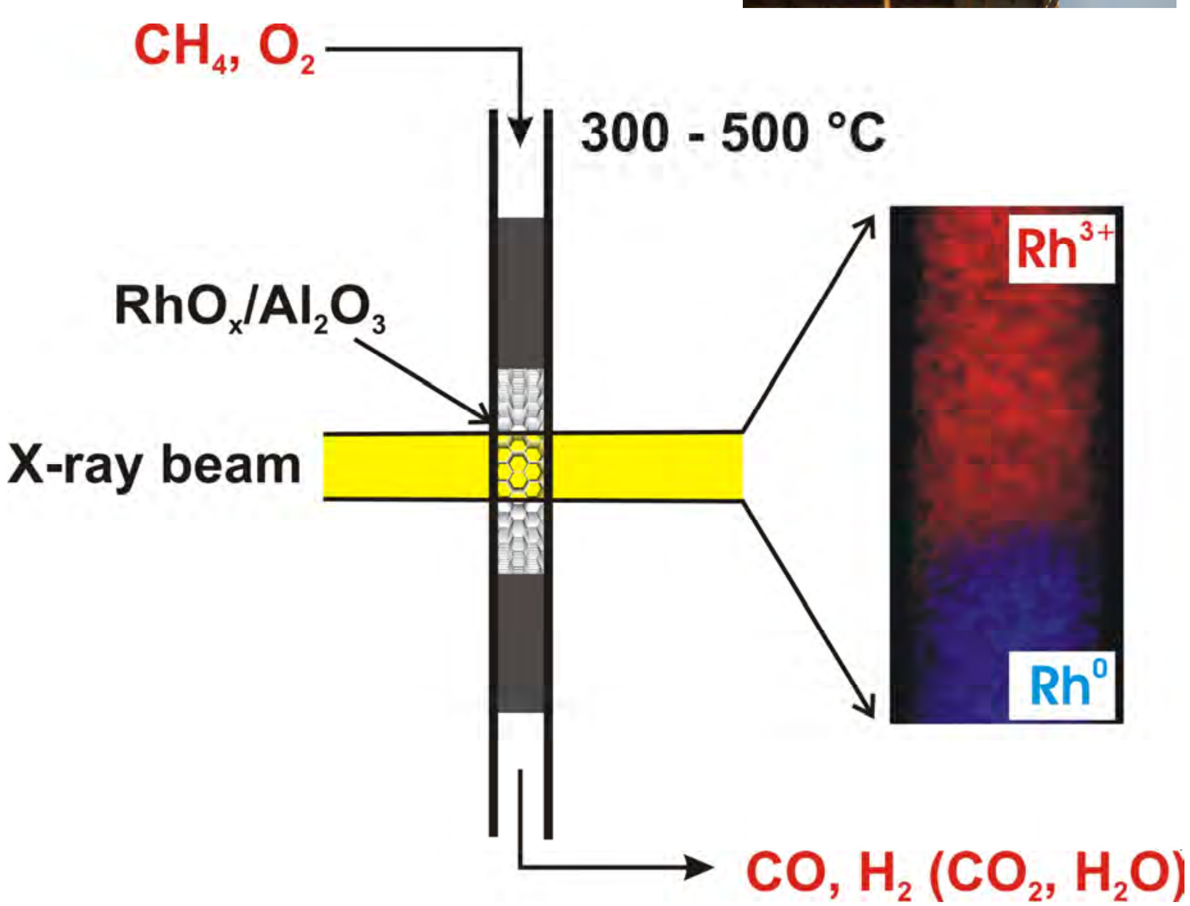

Grunwaldt, et al.,
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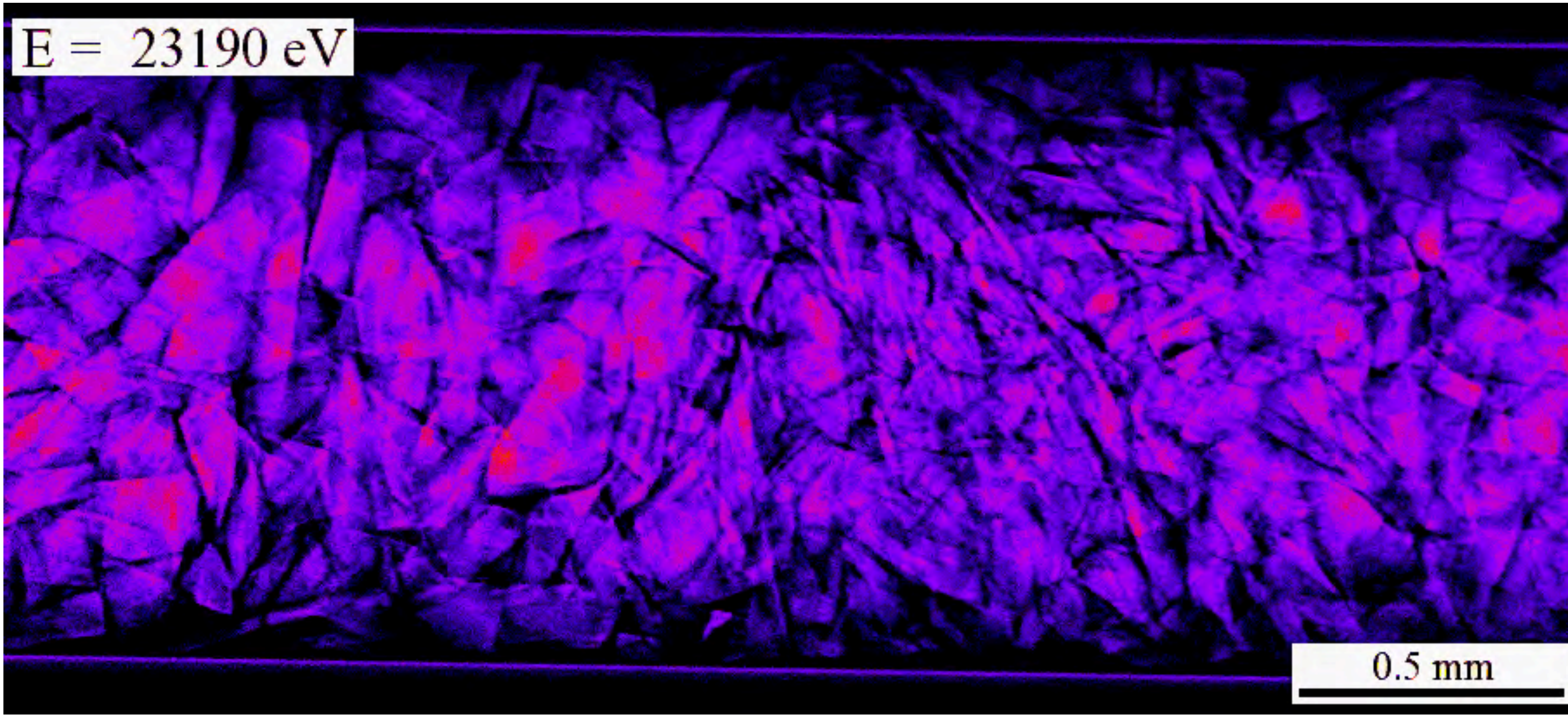
Visualize Catalysis

In-situ transmission imaging of catalyst bed inside chemical reactor



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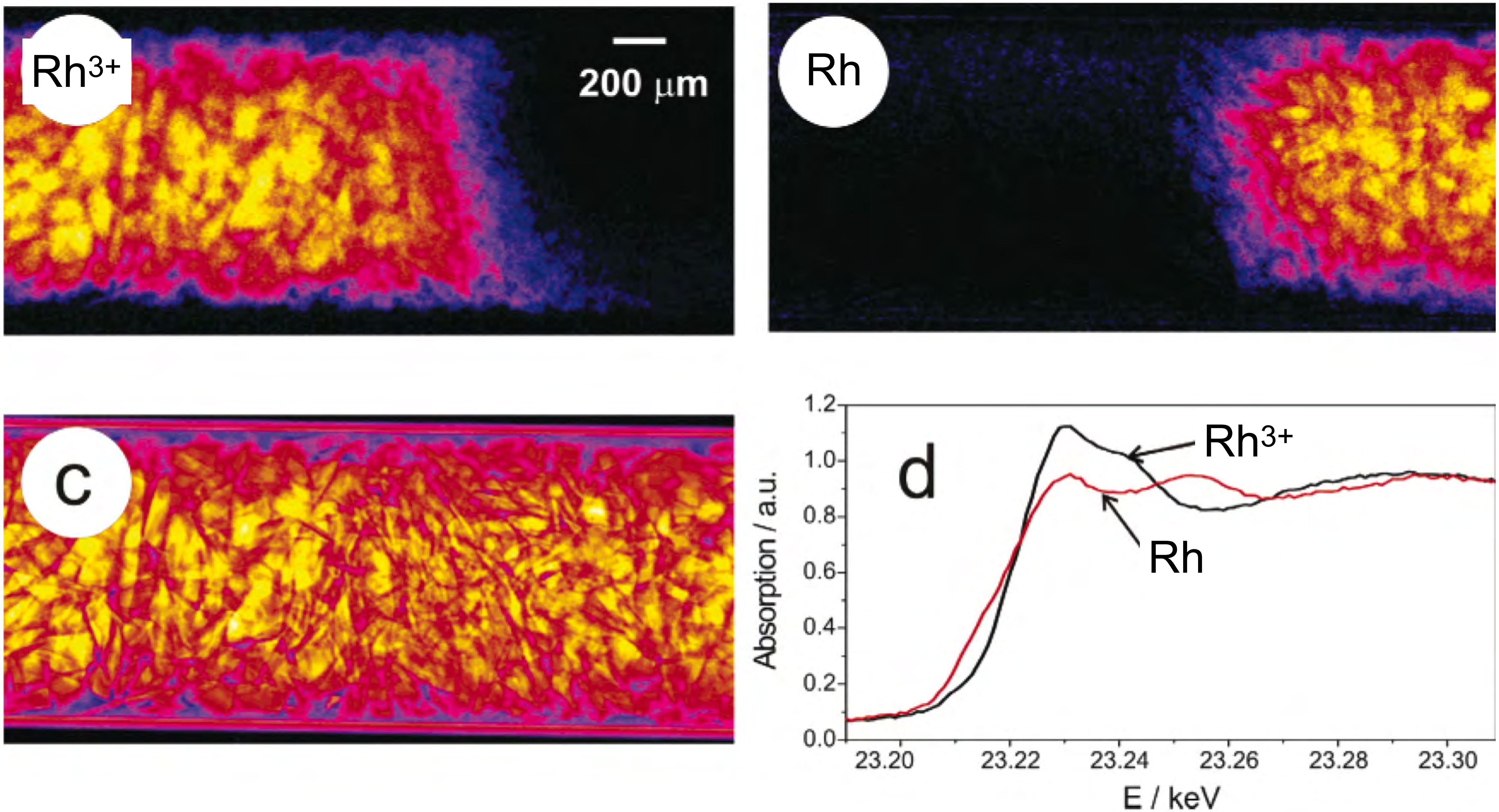
Visualize Catalysis



Grunwaldt, et al.,
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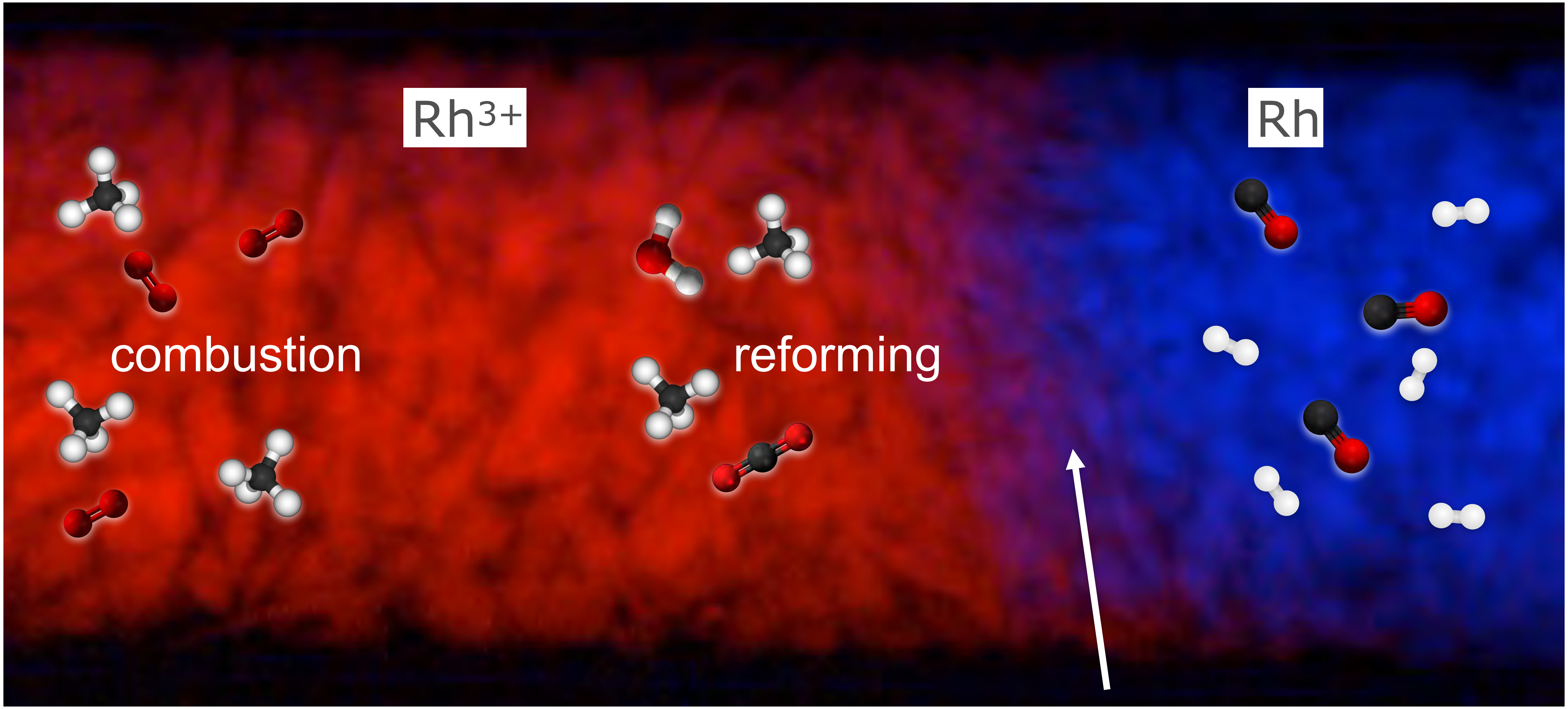
Visualize Catalysis

$$2\ CH_4 + O_2 \rightarrow 2\ CO + 4\ H_2$$

direction of flow



production of hydrogen
Rh is reduced!

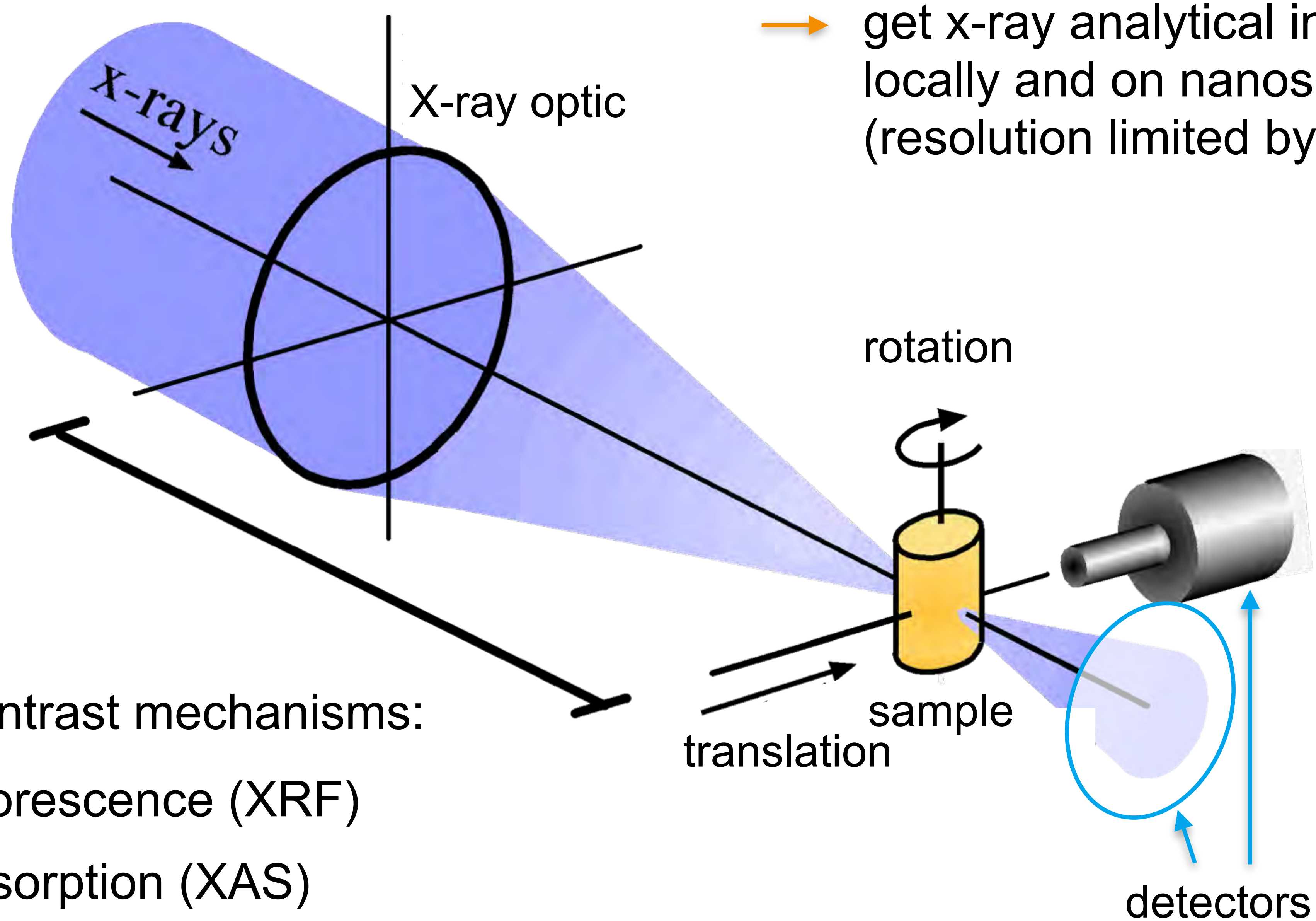
Grunwaldt, et al.,
J. Chem. Phys. B **110,** 8674 (2006)

Scanning Microscopy and Tomography: Nanoprobe

X rays are focused onto the sample

raster scan sample through beam:

→ get x-ray analytical information locally and on nanoscale (resolution limited by focus size).

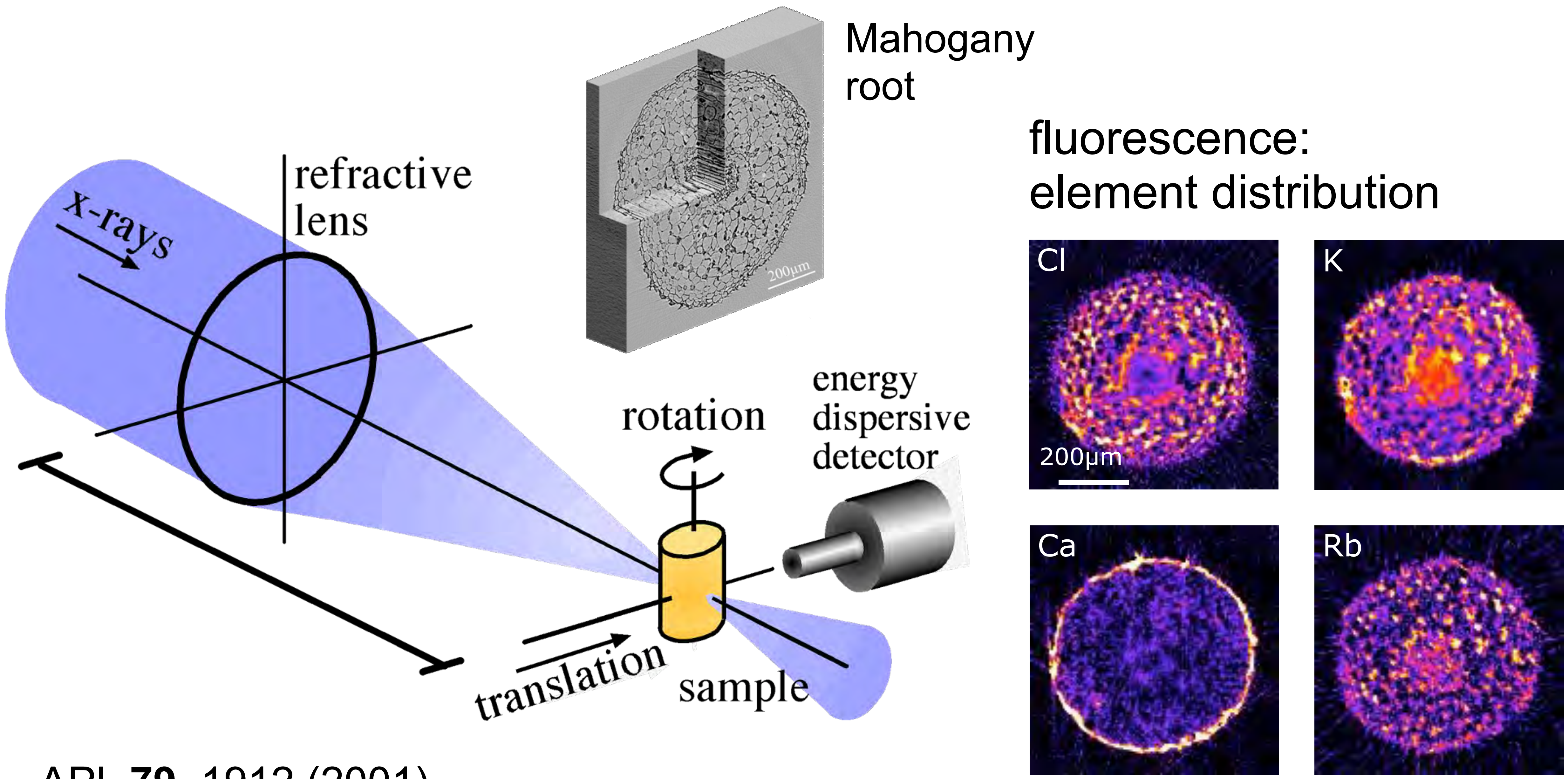


different contrast mechanisms:

- x-ray fluorescence (XRF)
- x-ray absorption (XAS)
- x-ray diffraction (XRD, SAXS, WAXS)
- maybe in future even IXS (RIXS)
- …

X-Ray Scanning Microscopy and Tomography

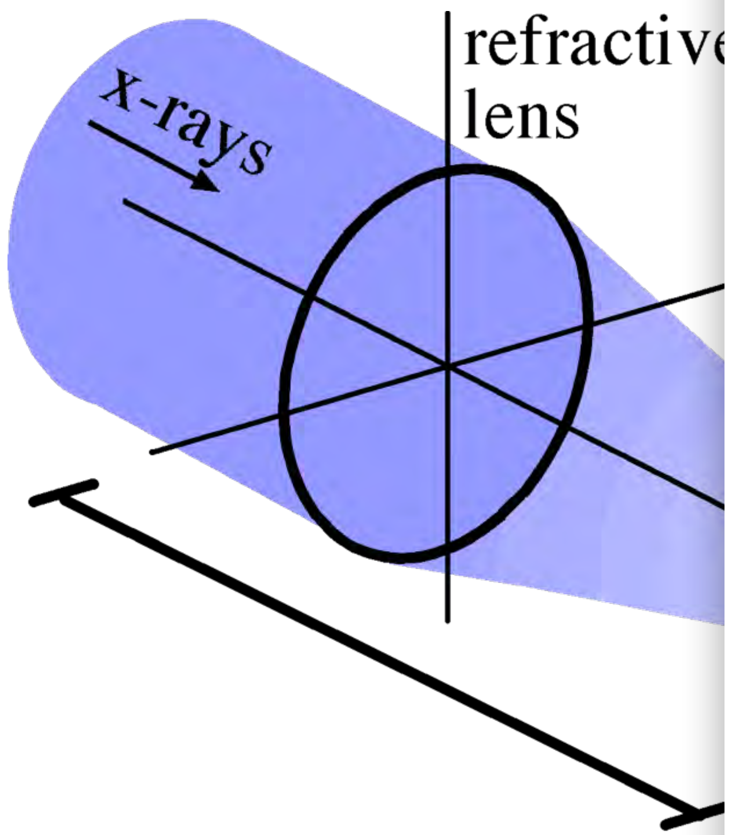
> Fluorescence microtomography



APL **79**, 1912 (2001).

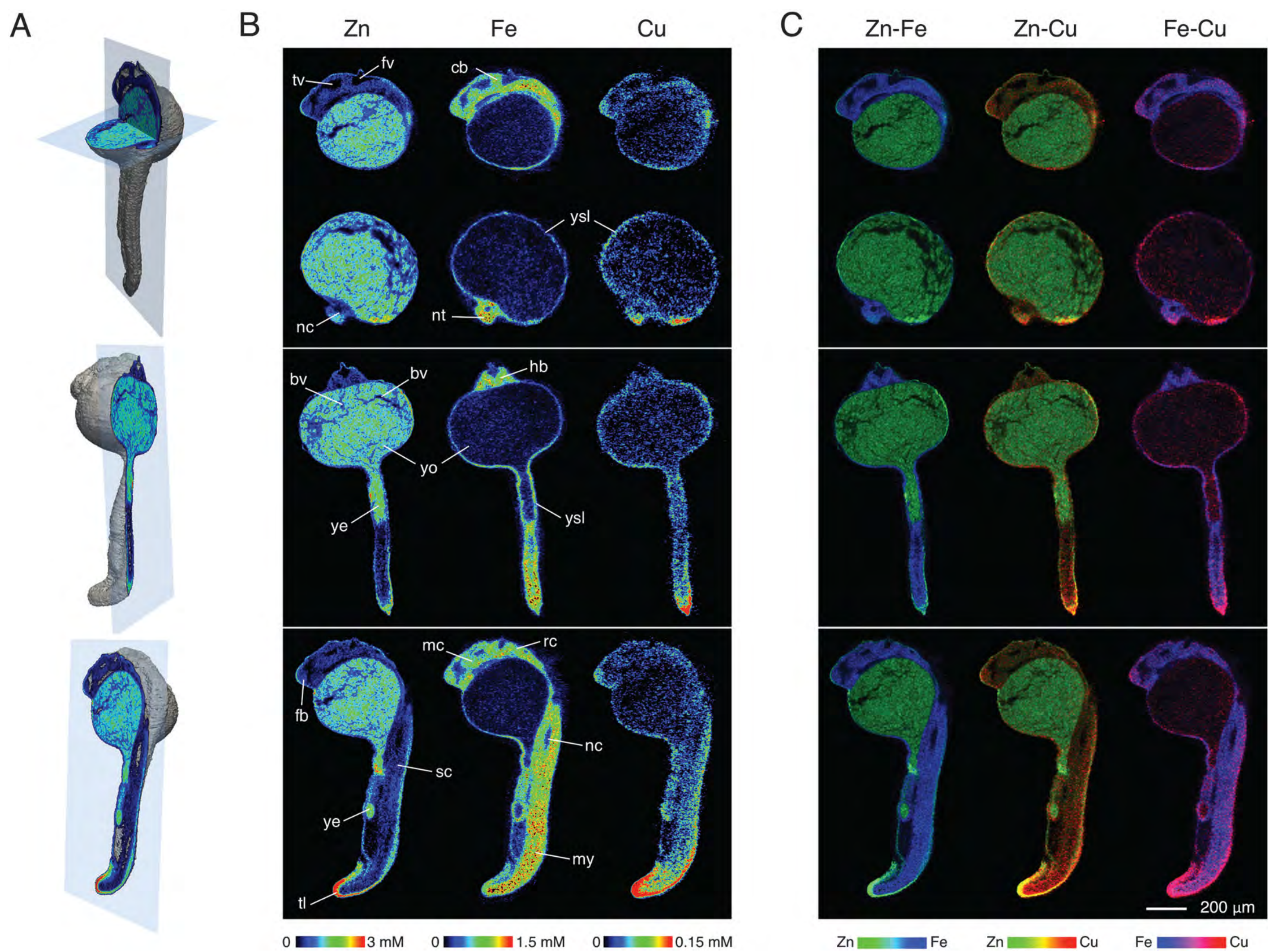
X-Ray Scanning Microscopy and Tomography

> Fluorescence microt



APL **79**, 1912 (2001).

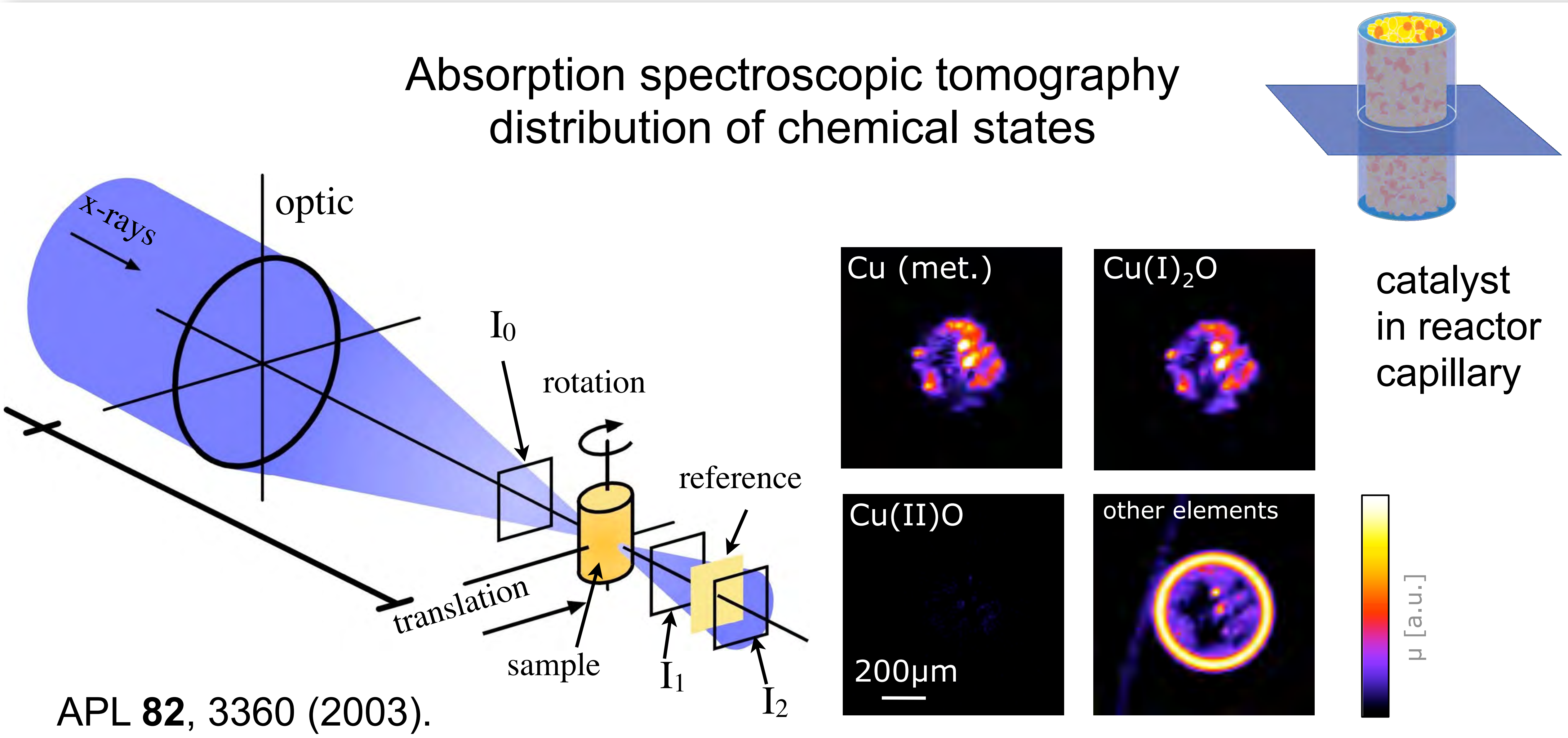
Full fluorescence tomogram of zebra fish embryo



D. Bourassa, et al., Metallomics 6, 1648 (2014).

X-Ray Scanning Microscopy and Tomography

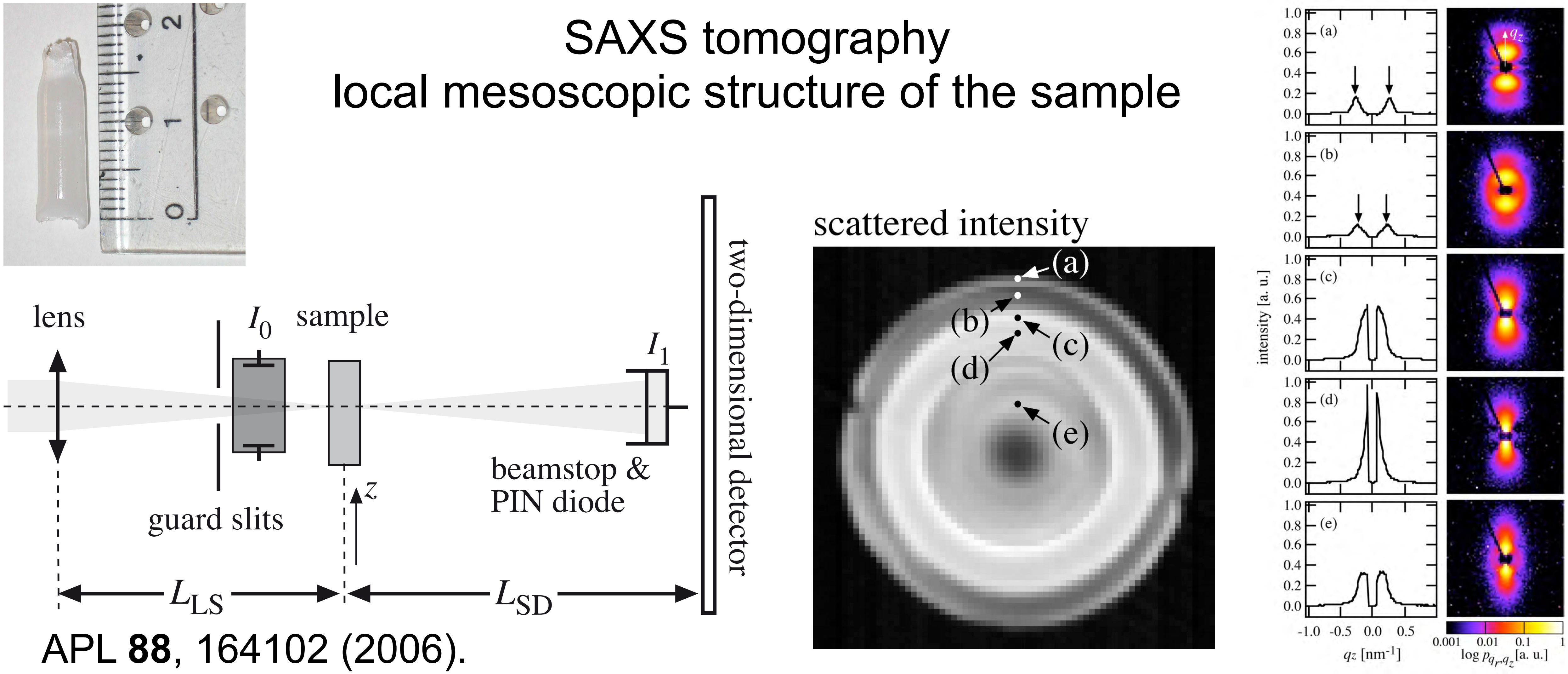
> Fluorescence microtomography
> Tomographic absorption spectroscopy (XANES tomography)



APL **82**, 3360 (2003).

X-Ray Scanning Microscopy and Tomography

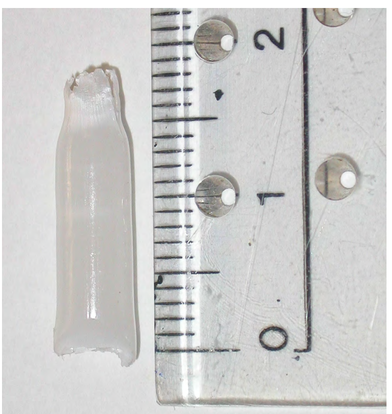
- Fluorescence microtomography
- Tomographic absorption spectroscopy (XANES tomography)
- Small-angle x-ray scattering tomography (SAXS tomography)



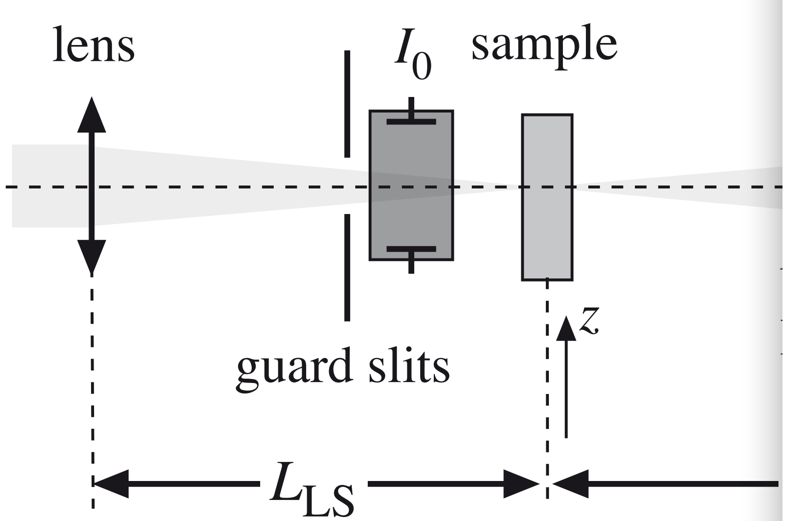
APL **88**, 164102 (2006).

X-Ray Scanning Microscopy and Tomography

- Fluorescence microtomography
- Tomographic absorpti
- Small-angle x-ray sca

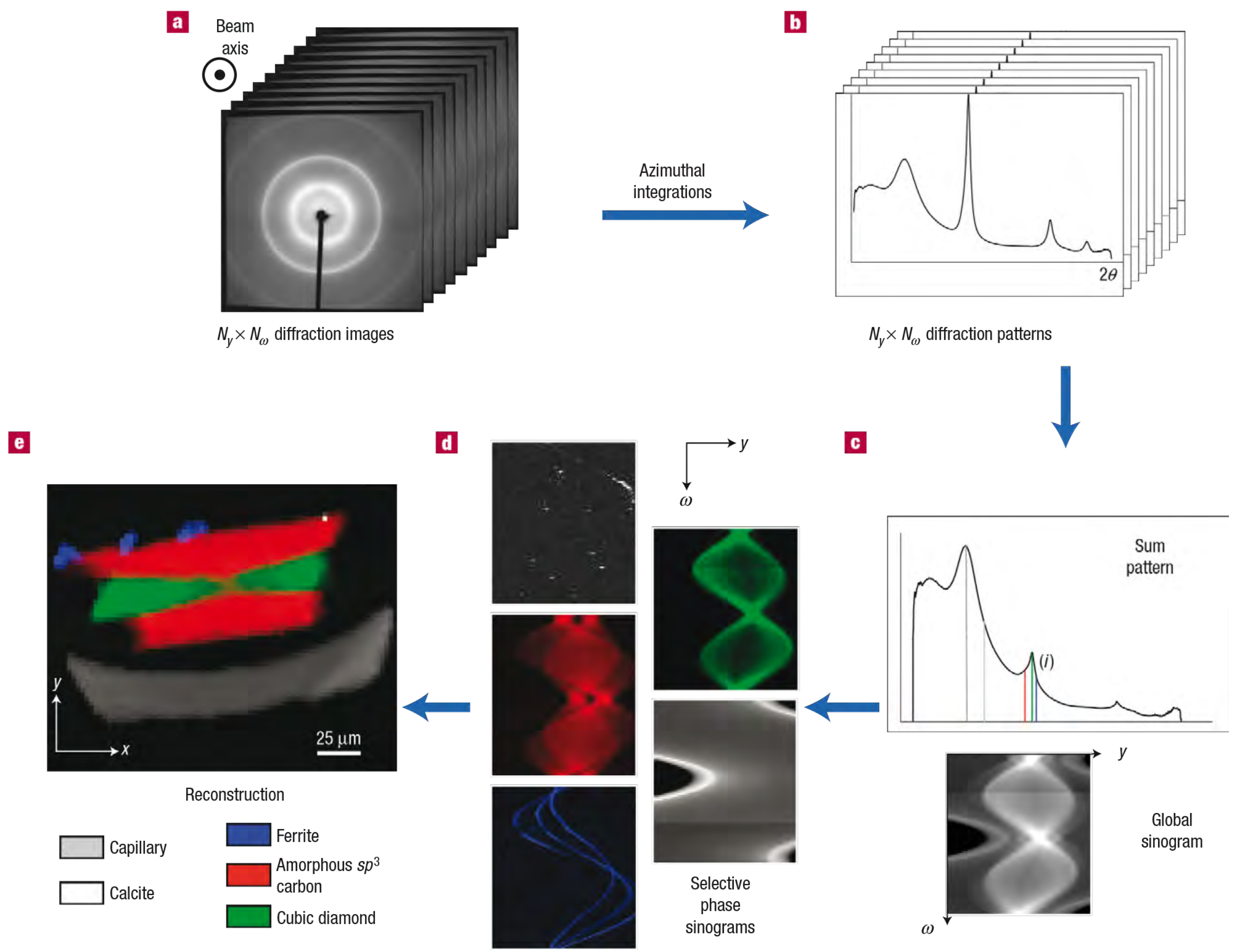


local me



APL **88**, 164102 (2006).

Wide-Angle-X-Ray-Scattering (WAXS) Tomography



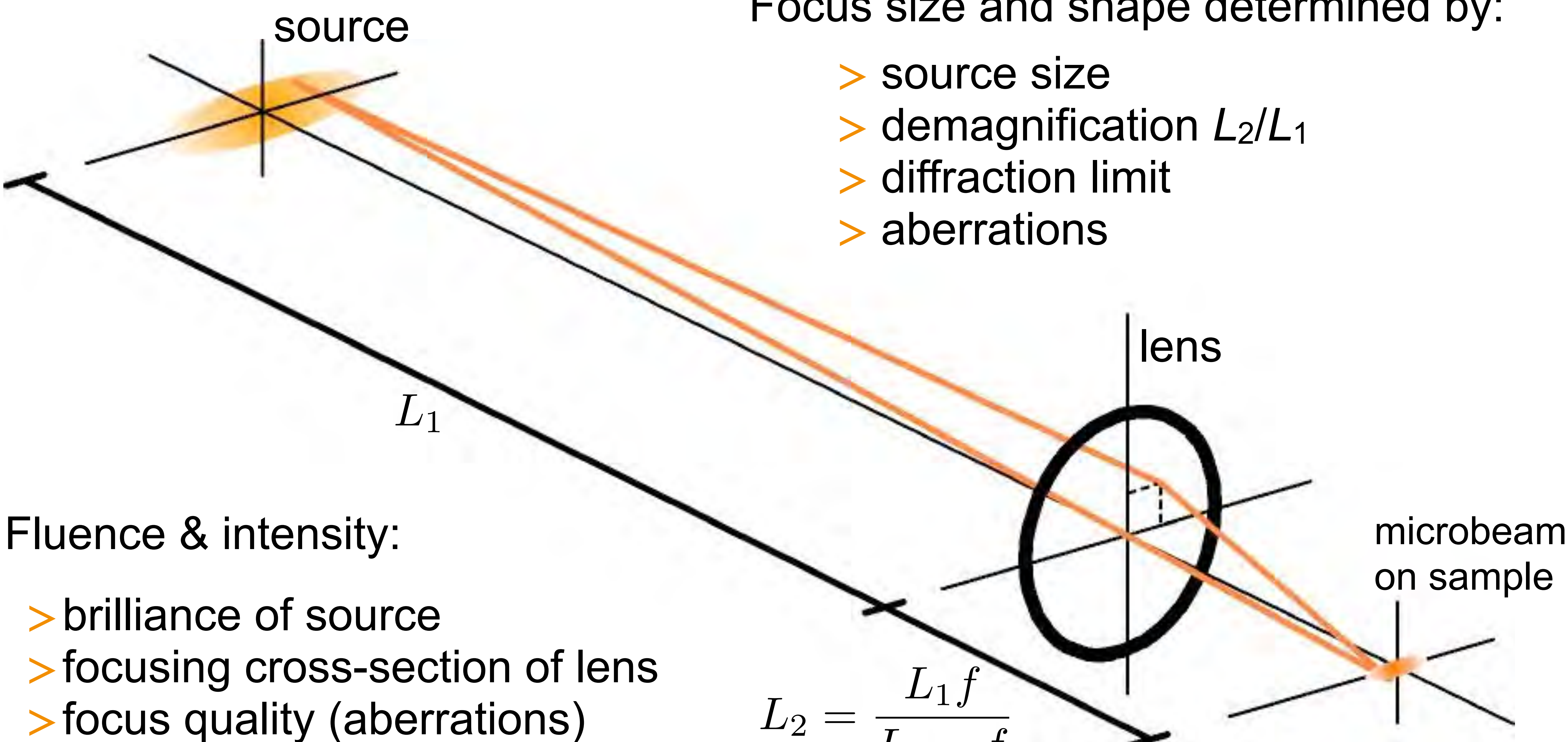
P. Bleuet, et al., Nat. Materials **7**, 468 (2008).

Scanning Microscopy with Hard X-Rays

Source is imaged onto the sample to create an intensive micro-/nanobeam:

Focus size and shape determined by:

- source size
- demagnification L_2/L_1
- diffraction limit
- aberrations



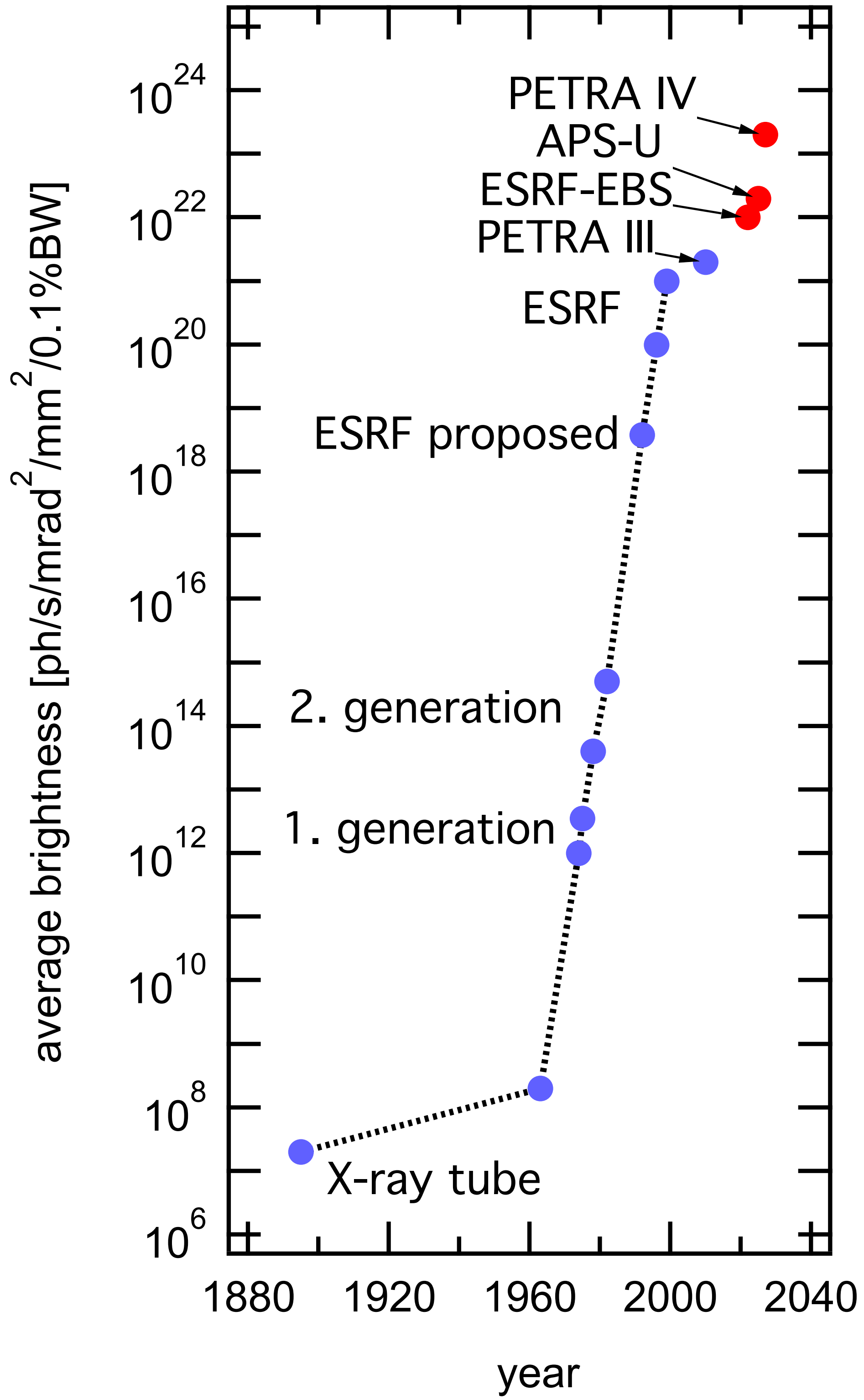
$$L_2 = \frac{L_1 f}{L_1 - f}$$

Fluence & intensity:

- brilliance of source
- focusing cross-section of lens
- focus quality (aberrations)

Spectral Brightness

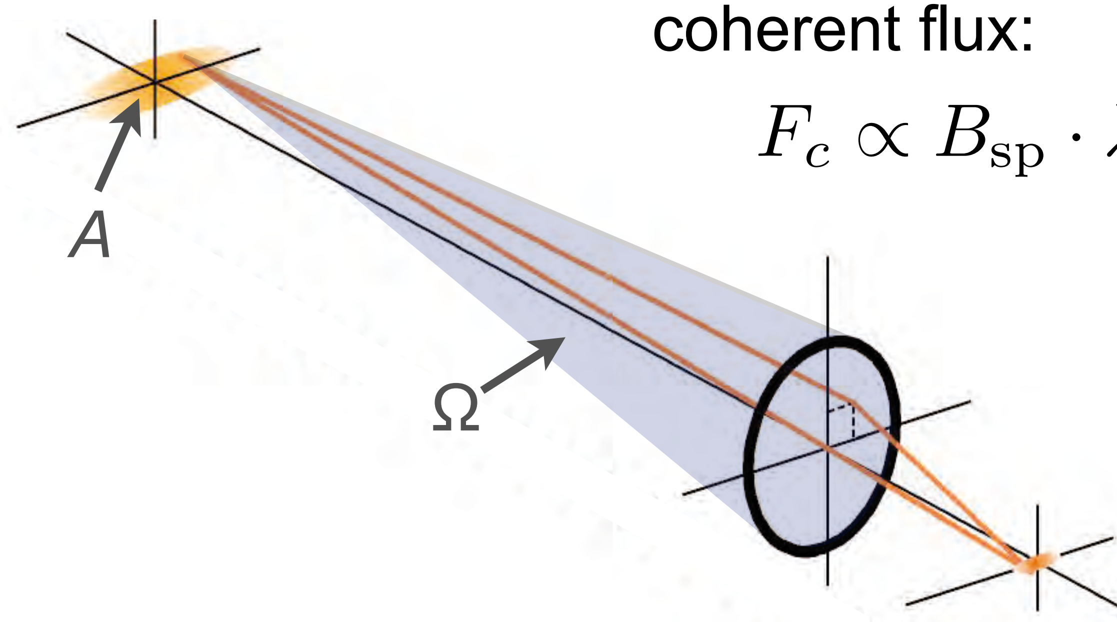
10000x more light per decade (since 1965)



Spectral brightness:

$$B_{\mathrm{sp}} = \frac{F}{\Omega \cdot A \cdot \Delta E / E}$$

Flux per phase-space volume



coherent flux:

$$F_c \propto B_{\mathrm{sp}} \cdot \lambda^2 \cdot \frac{\Delta E}{E}$$

Improvements in brightness:

- > faster measurements (time resolution)
- > nano-imaging (spatial resolution)
- > spectroscopy (energy resolution)

Fluorescence Tomography

Example: investigating the ion transport in plants

Fluorescence analysis of plants:

- strong diffusion of elements
- cell structure complicated and delicate

Difficult sample preparation

- cryo sections
- fracture surfaces

ideal:

nondestructive probe of
inner structures of sample

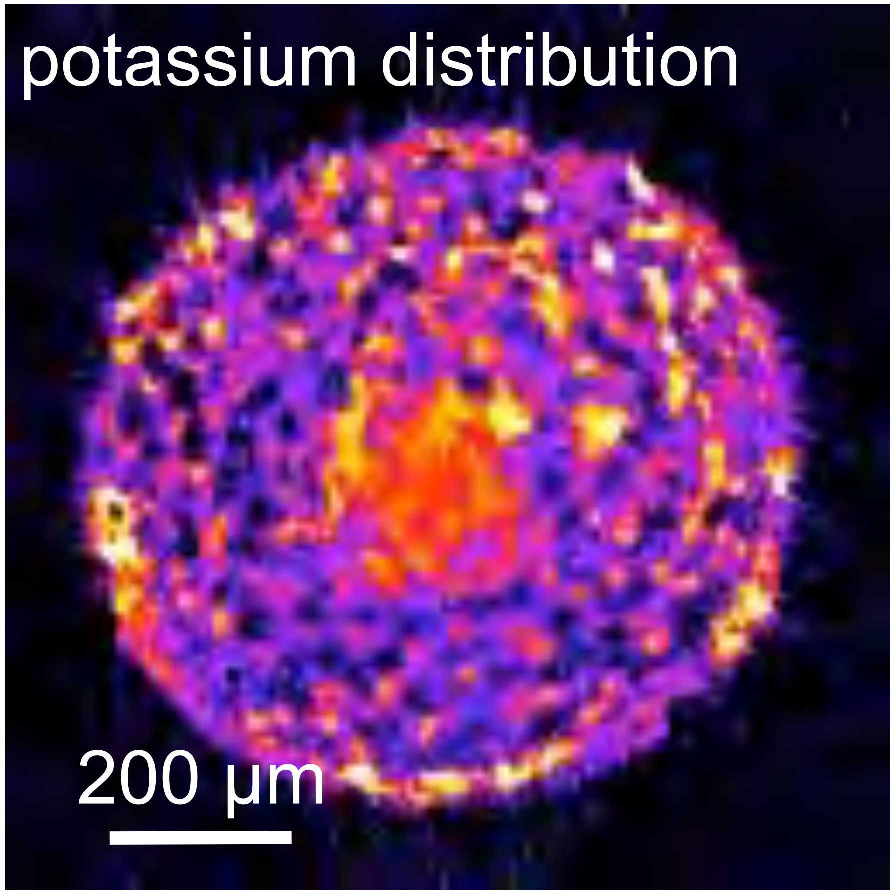
root of Mahogany tree

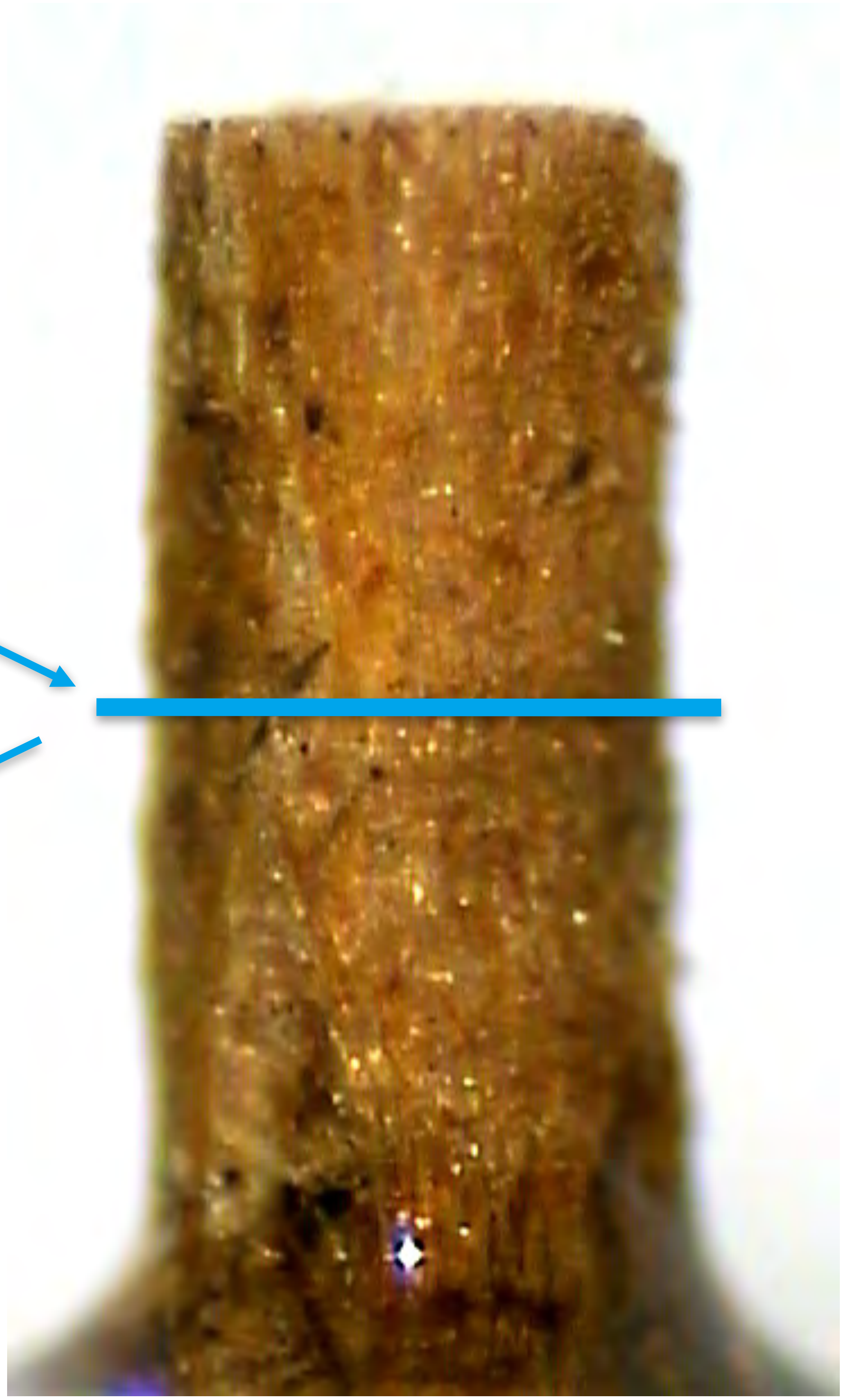
Fluorescence Tomography

Root of Mahogany tree

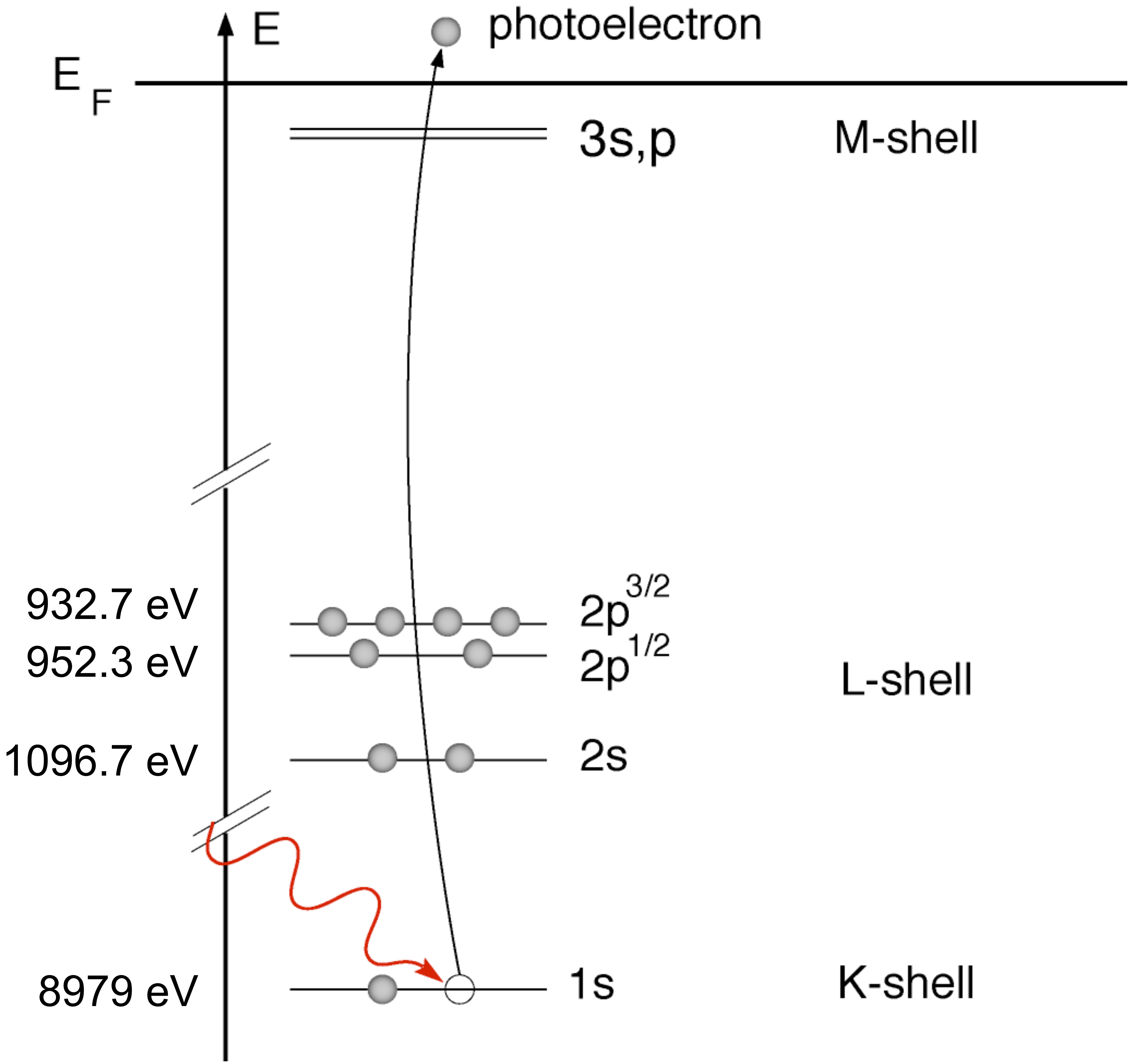
element distribution on virtual section through sample

Example:





X-ray Fluorescence & Auger Process



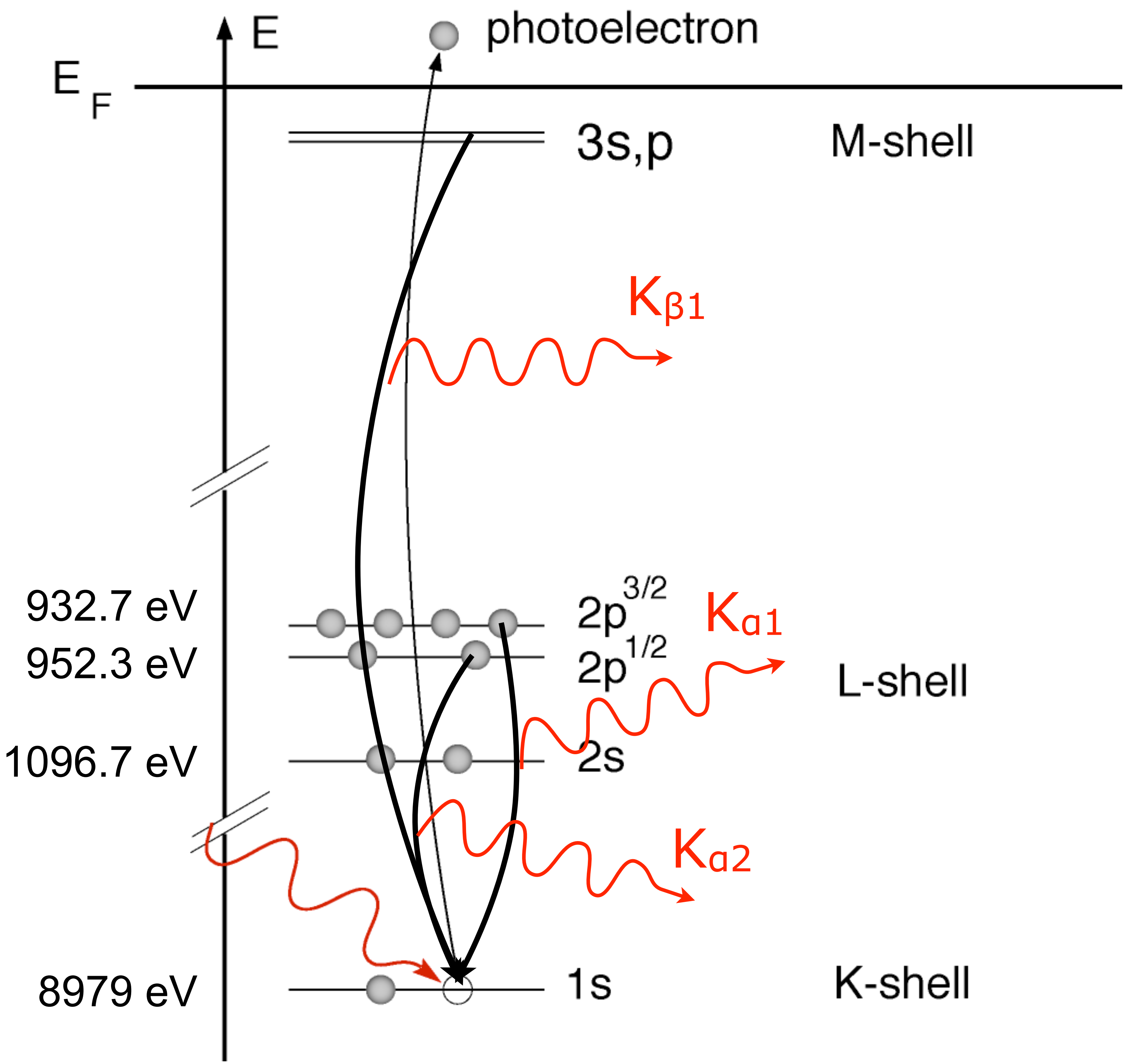
X-ray Absorption

leaves atom in excited state (core hole):

Secondary processes:

> fluorescence

X-ray Fluorescence & Auger Process



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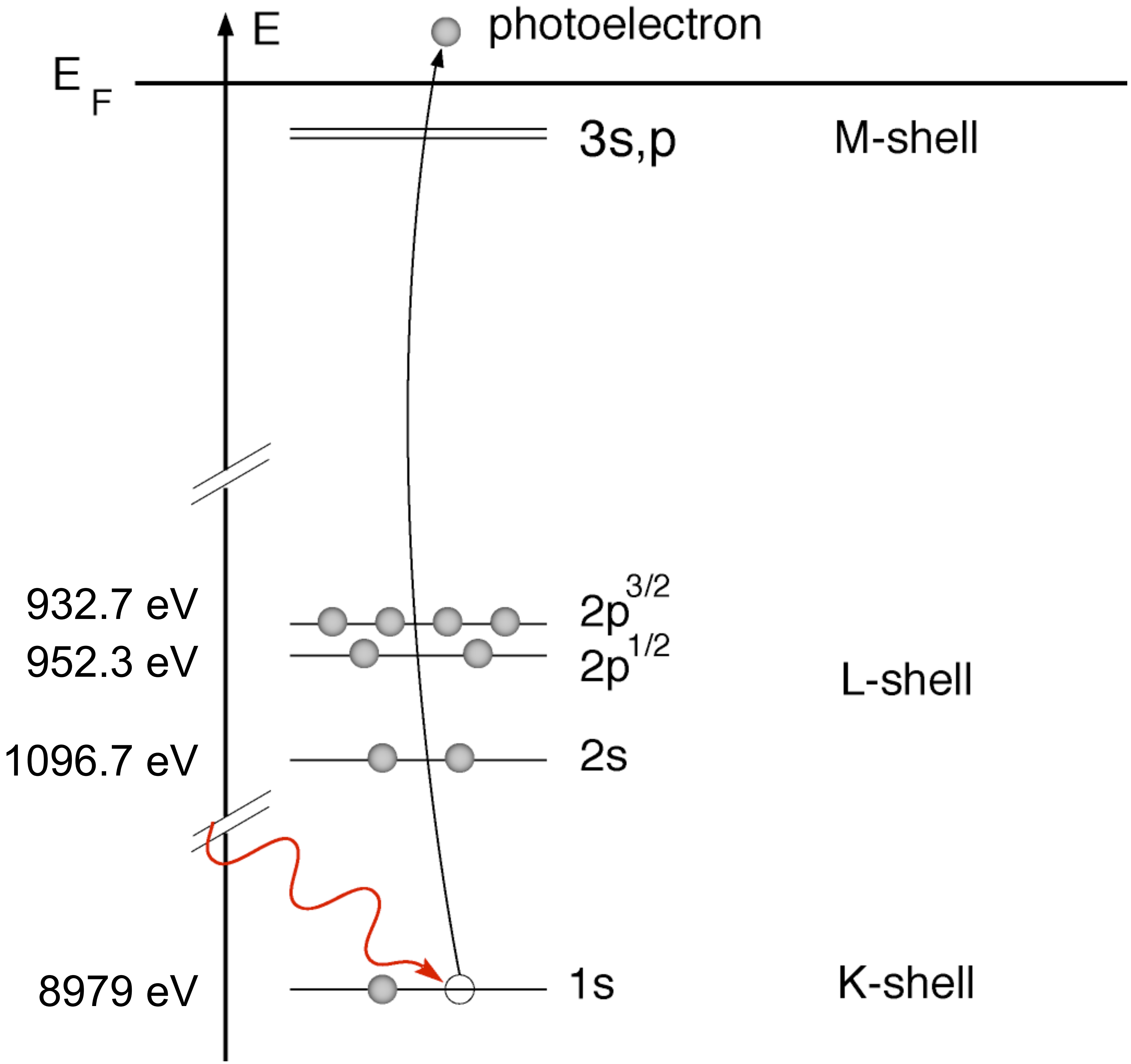
X-ray Fluorescence & Auger Process



X-ray Absorption

leaves atom in excited state (core hole):

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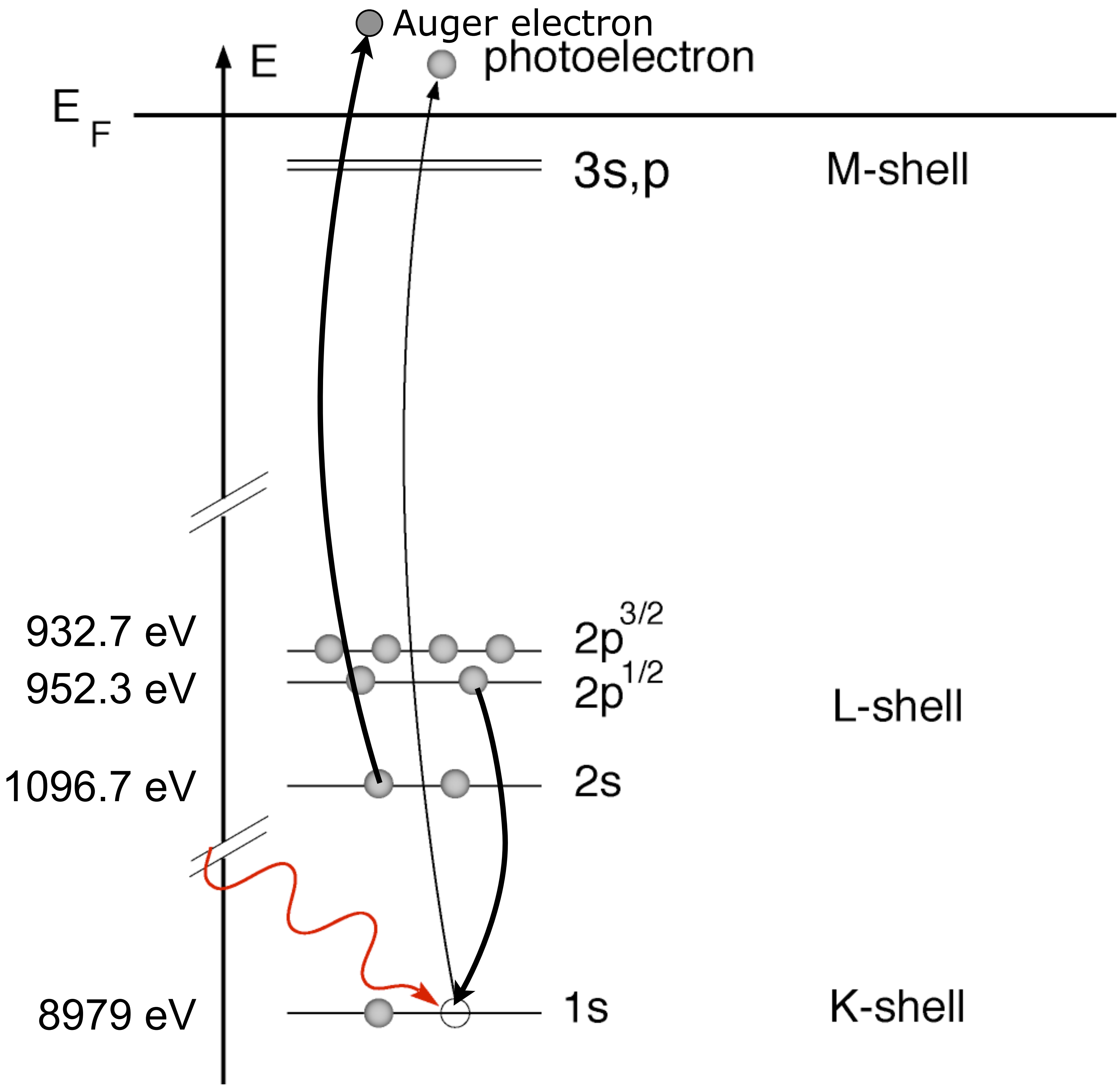
X-ray Fluorescence & Auger Process



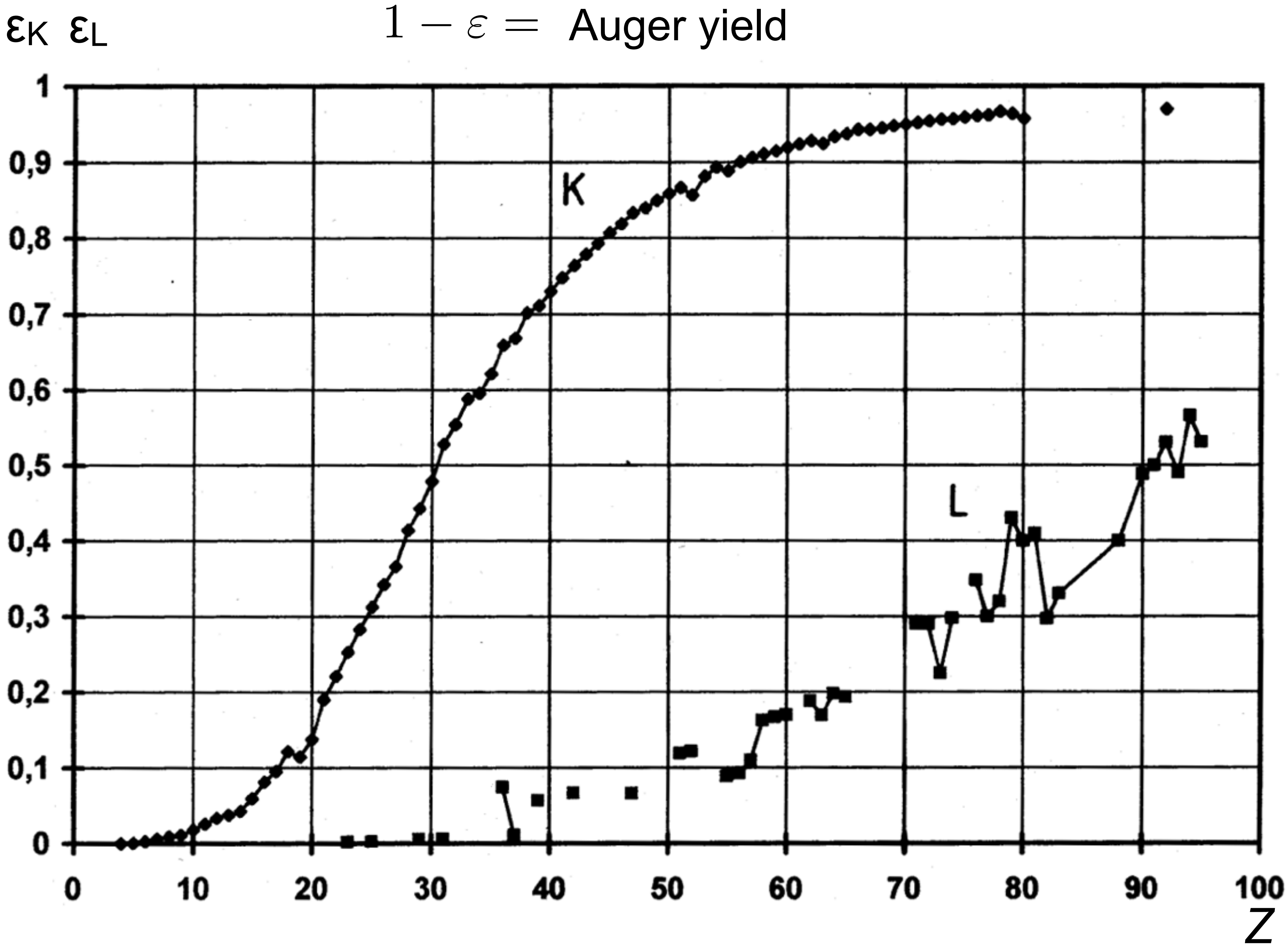
X-ray Absorption

leaves atom in excited state (core hole):

Secondary processes:

- fluorescence

Fluorescence Yield

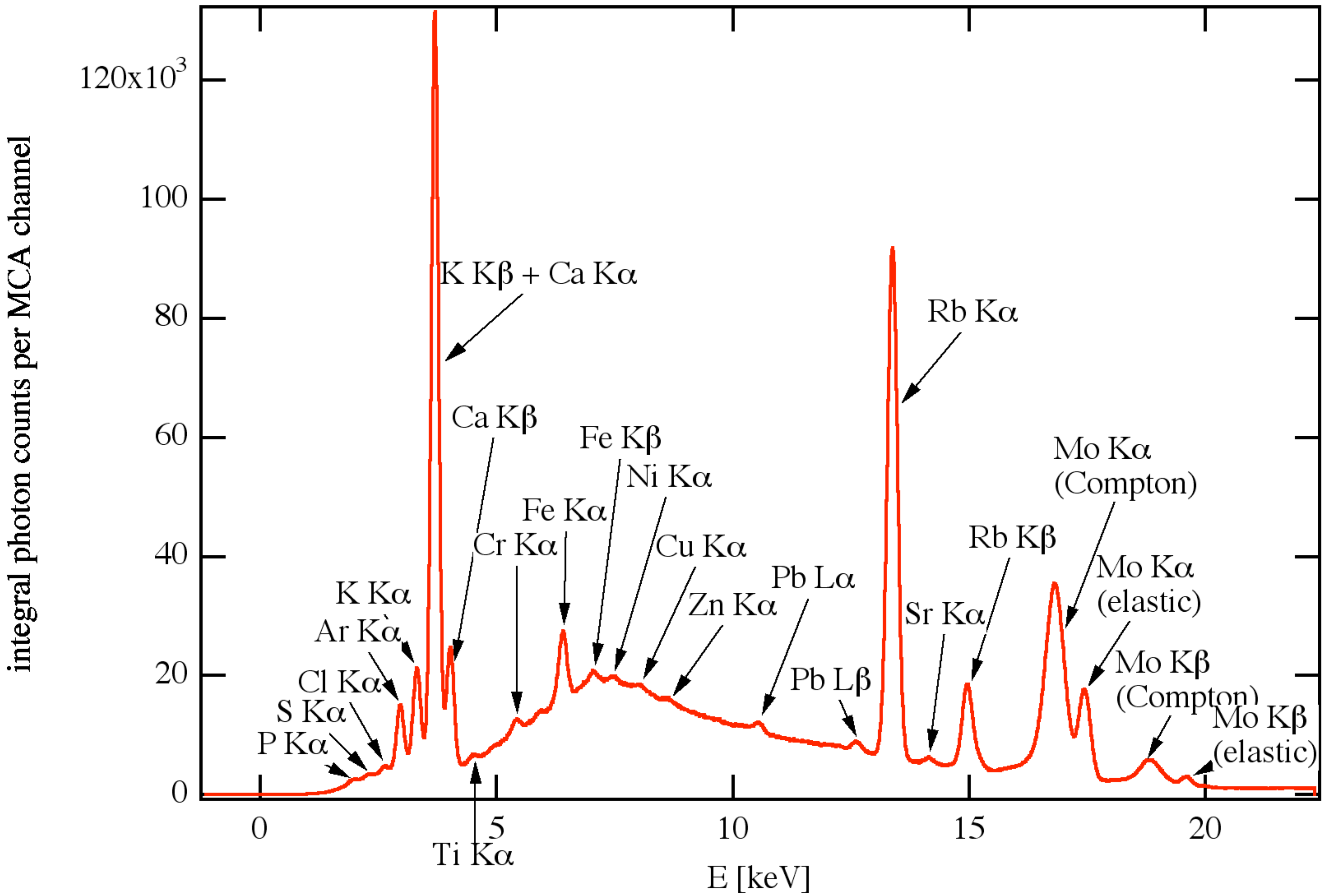


$1 - \varepsilon =$ Auger yield

element specific

Fluorescence dominates at higher binding energies for core hole excitation (growing with atomic number Z)

Fluorescence Spectrum



Illuminated atoms emit characteristic fluorescence radiation!

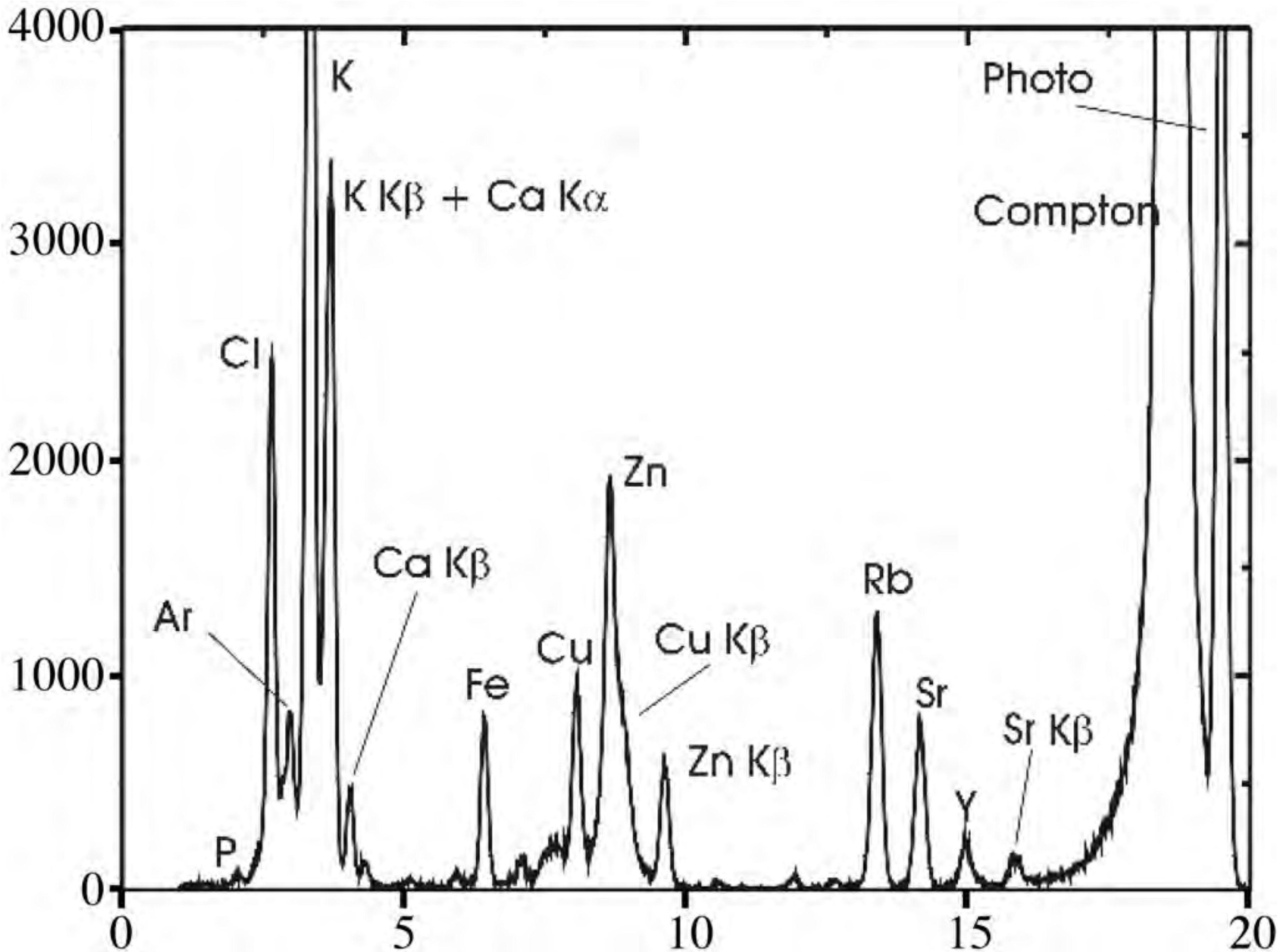
Example spectrum excitation with X-ray tube:

Background due to scattered spectrum of X-ray tube

Limitation of detection limits by background!

Excitation with Monochromatic Synchrotron Radiation

Example: undulator radiation (Si 111 monochrom.): 19.5 keV

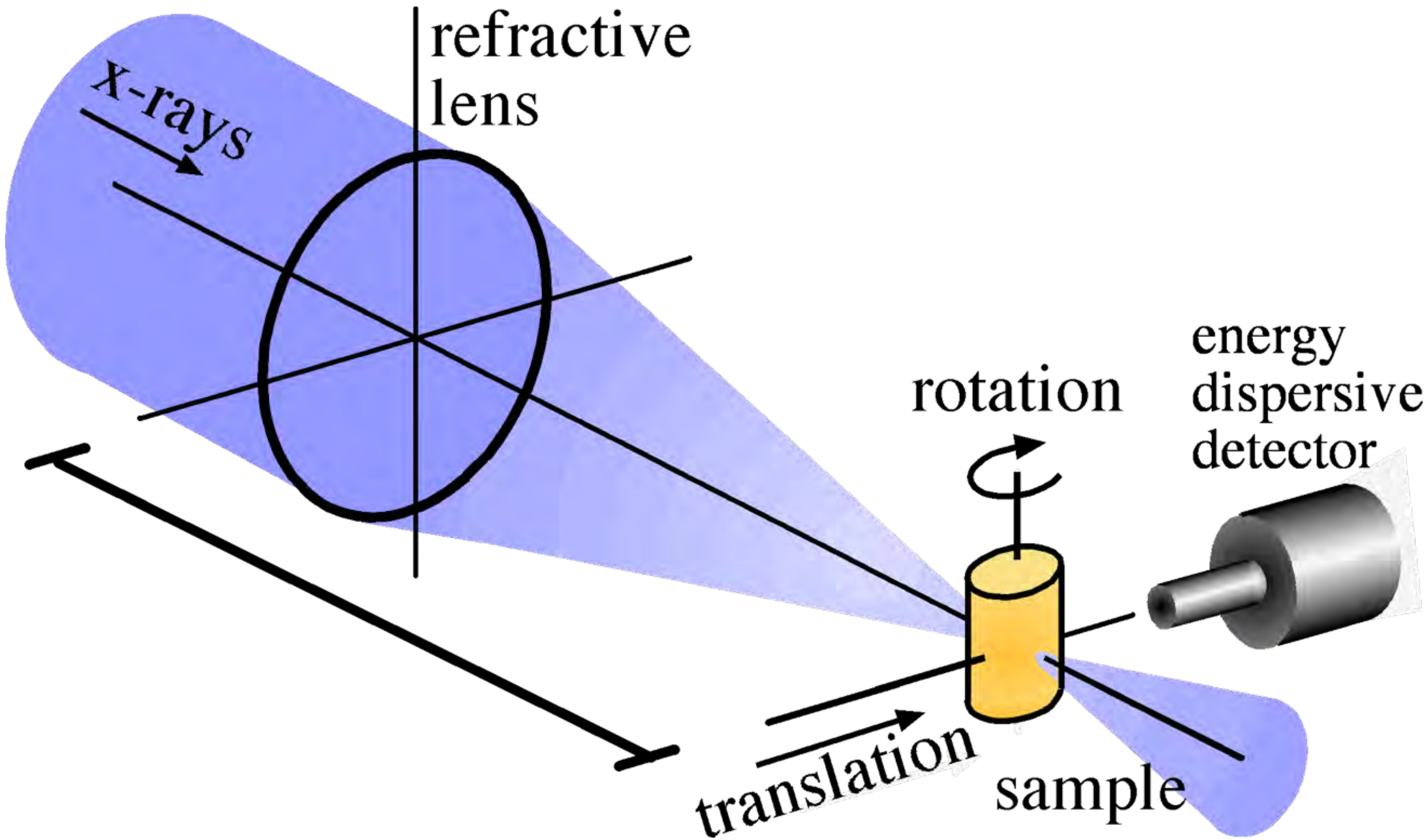


No background due to scattered radiation at fluorescence energy

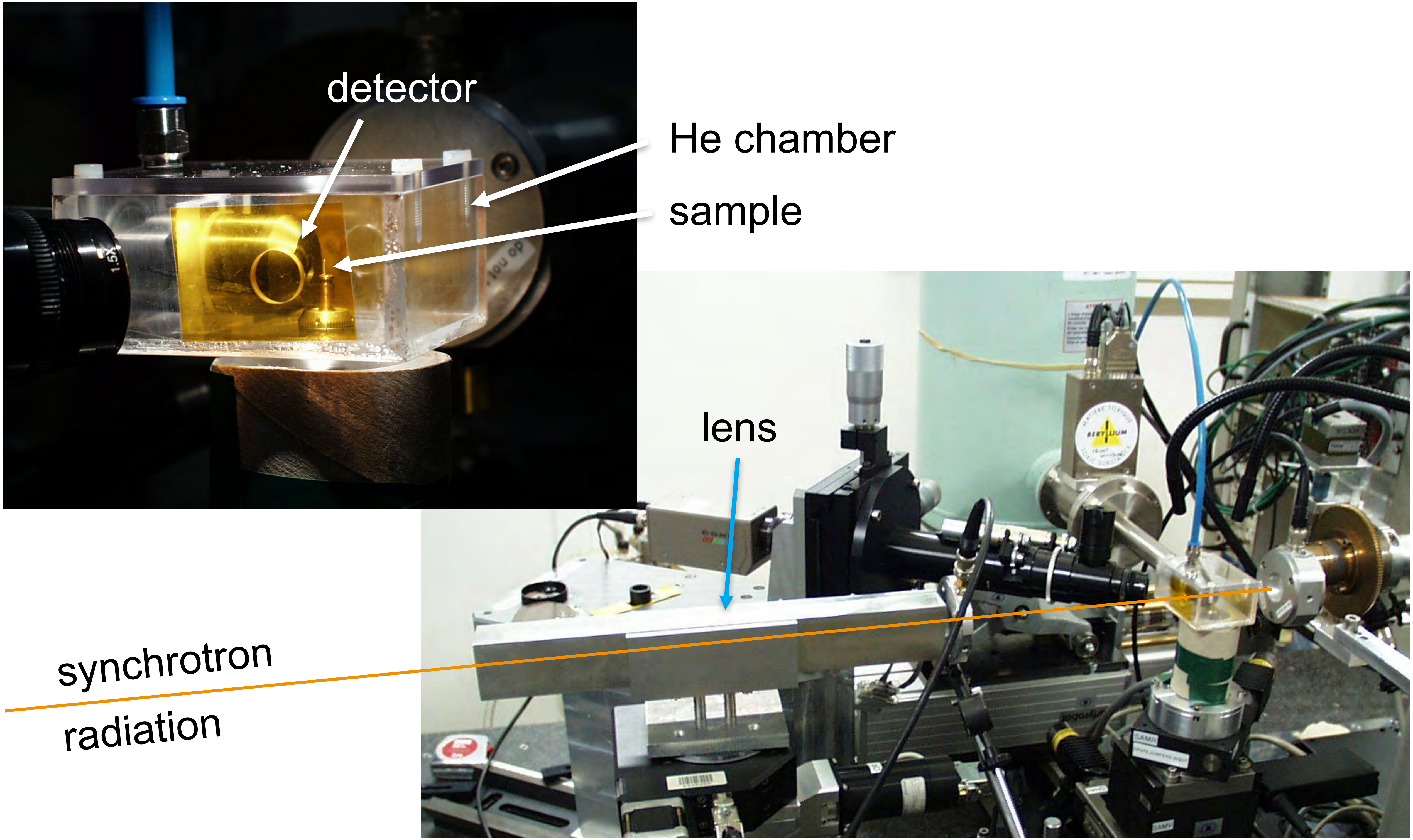
High signal-to-background ratio!!

very low detection limits possible (ppb-level)!

Scanning Probe: Fluorescence Microtomography



Fluorescence Microtomography

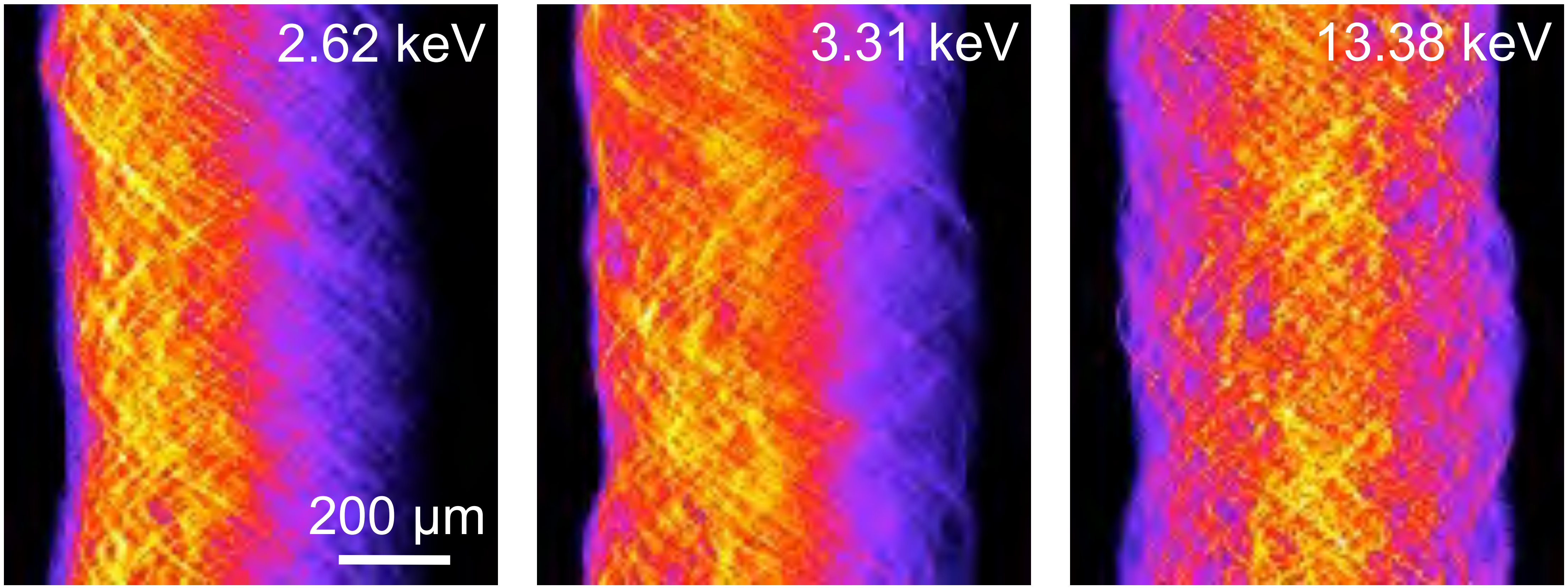


Fluorescence Tomography: Measured Data

Sinograms:

Cl Kα | K Kα | Rb Kα

152 rotations



$0^\circ - 360^\circ, \delta\varphi = 2.38^\circ$

translations: 128, 6µm

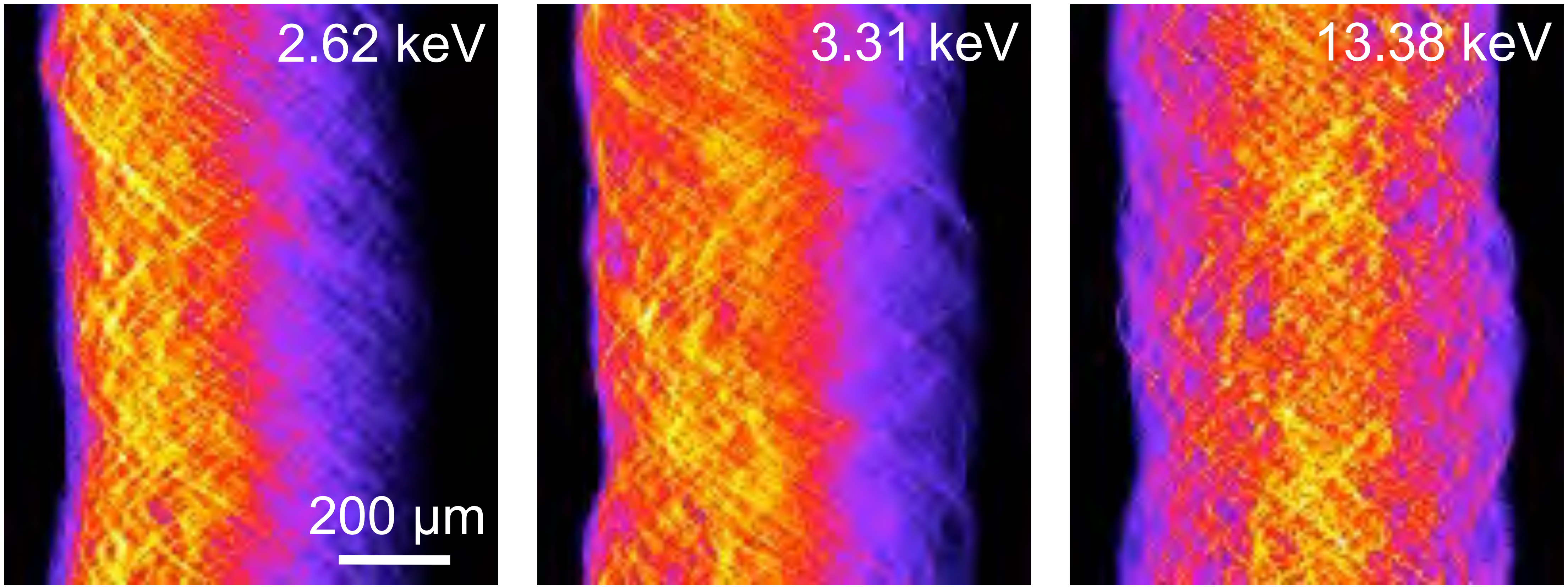
experimental parameters:

- energy: 19.5 keV
- refractive lens (Al): *N* = 150, f = 45.4 cm, m = 1/127
- beam size: 1.5 x 6µm^2, flux: 1.1 · 10^{10} ph/s

Fluorescence Tomography: Measured Data

Sinograms:

Cl Kα | K Kα | Rb Kα



152 rotations

$0^\circ - 360^\circ, \delta\varphi = 2.38^\circ$

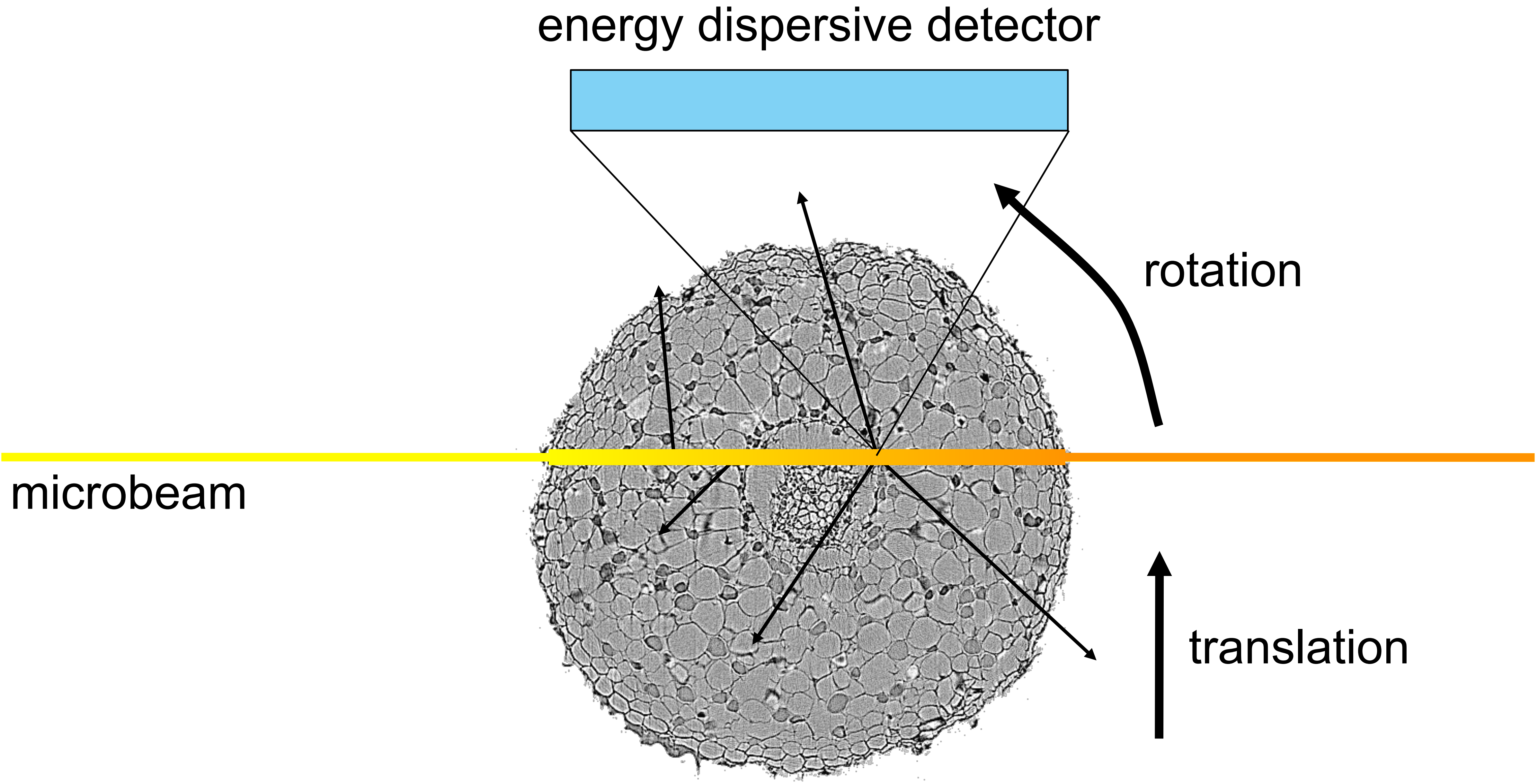
translations: 128, 6µm

Symmetry:

$$I_{i\nu}(-r, \varphi + \pi) = I_{i\nu}(r, \varphi)$$

only holds (approx.) for Rb!
Absorption of fluorescence radiation: asymmetry in sinogram.

Fluorescence Tomography: Model

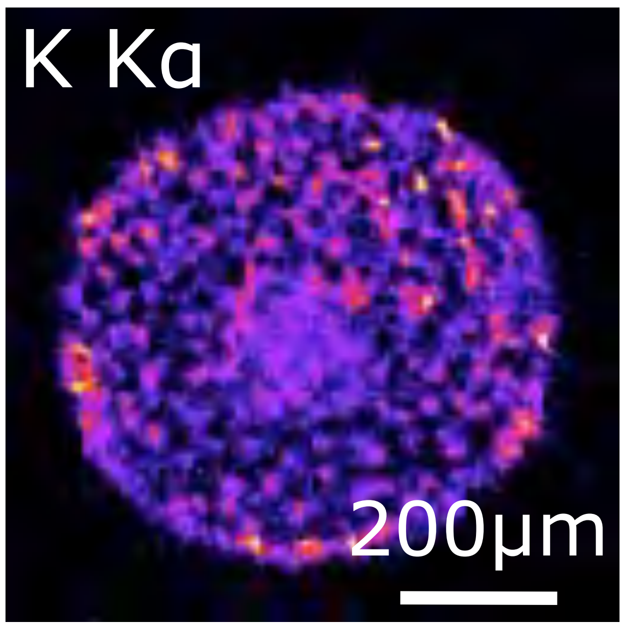


$$I_{i\nu}(r,\varphi) = I_0 \int ds \left[e^{-\int_{-\infty}^{s} ds' \mu_0(x,y)} \cdot p_{i\nu}(x,y) \cdot \int d\gamma e^{-\int dr' \mu_{i\nu}(x,y)} \right]$$

Absorption Correction

Example: potassium distribution in Mahogany root

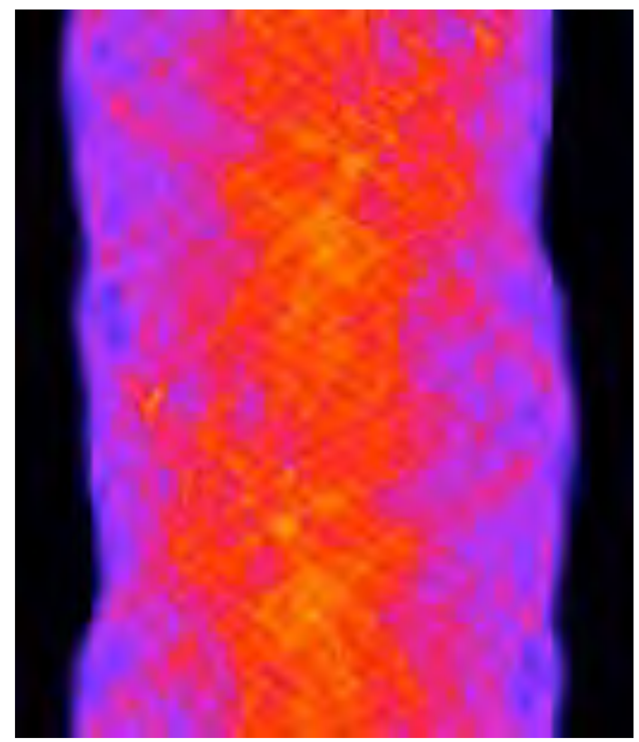
Disregarding attenuation of fluorescence:



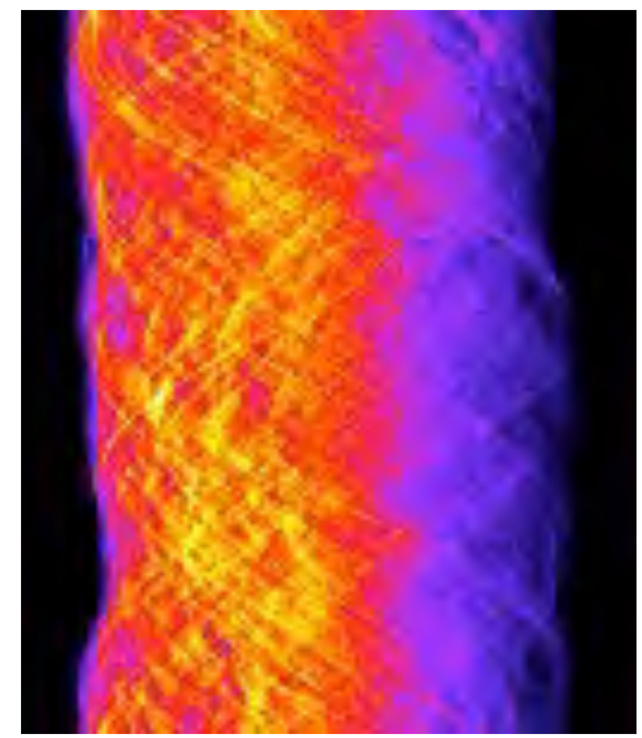
simple tomographic model →

$$\mu_{\mathrm{K\,K}\alpha} = 0$$

reconstructed sinogram



measured sinogram

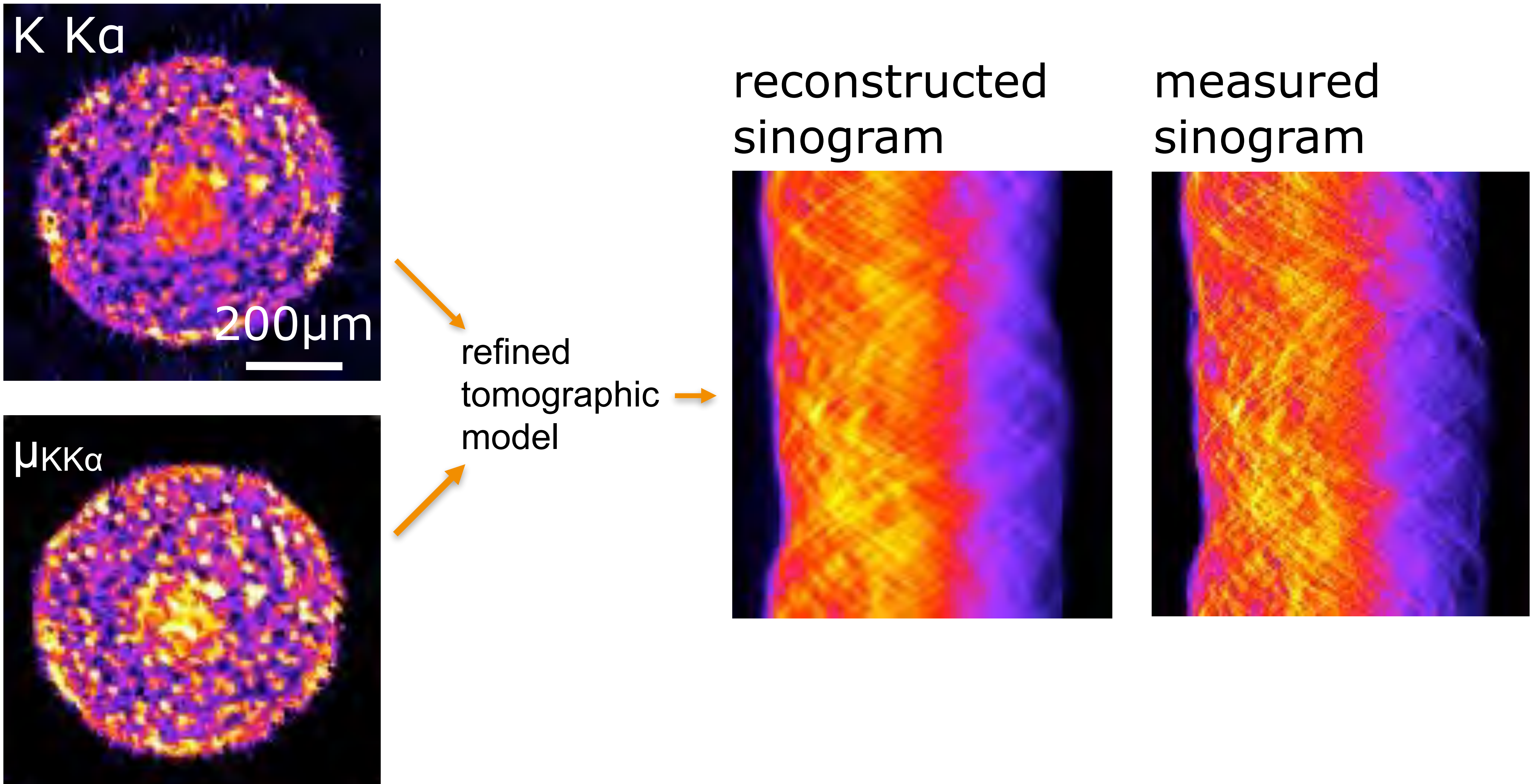


C. Schroer, Appl. Phys. Lett. **79**, 1912 (2001).

Absorption Correction

Example: potassium distribution in Mahogany root

Accounting for attenuation of fluorescence:

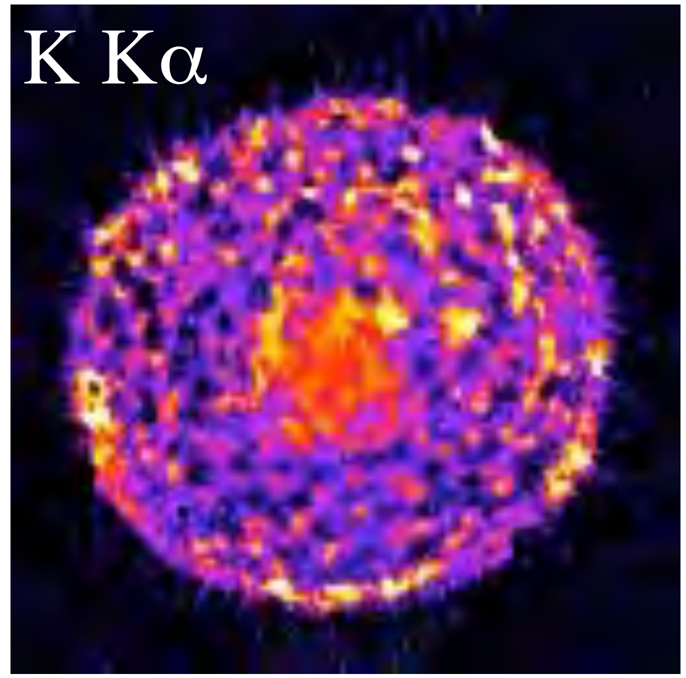


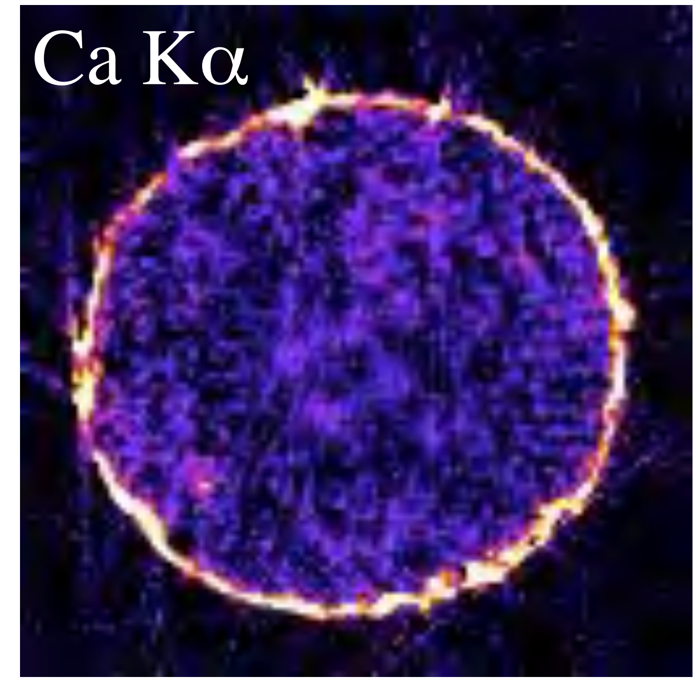
C. Schroer, Appl. Phys. Lett. **79**, 1912 (2001).

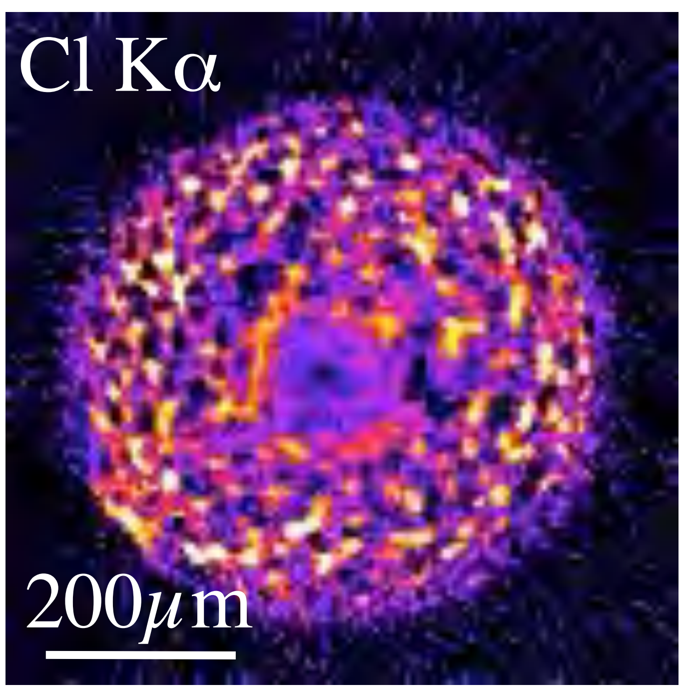
Fluorescence Tomography

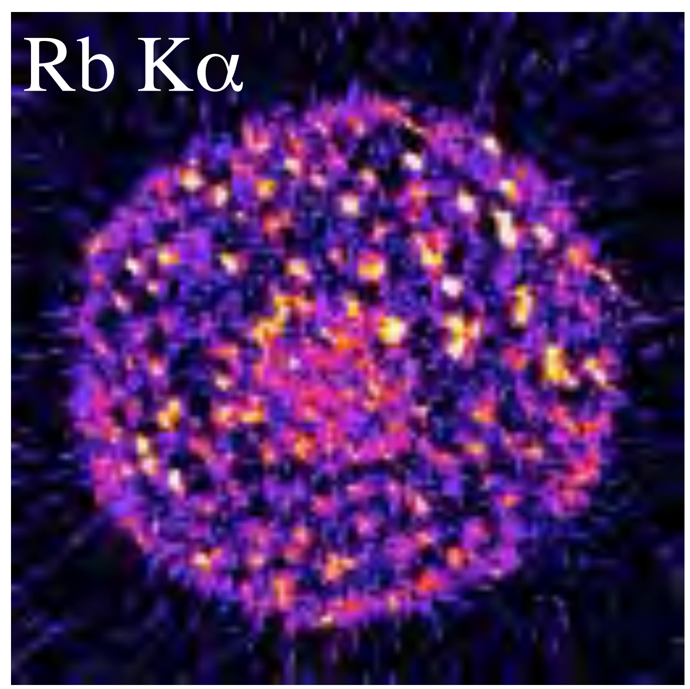
root of Mahogany tree

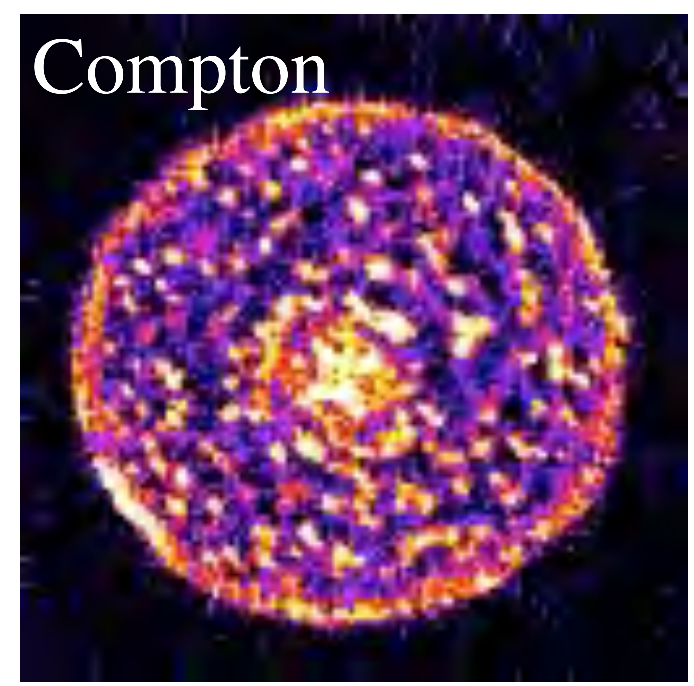
pixel size: 6 µm











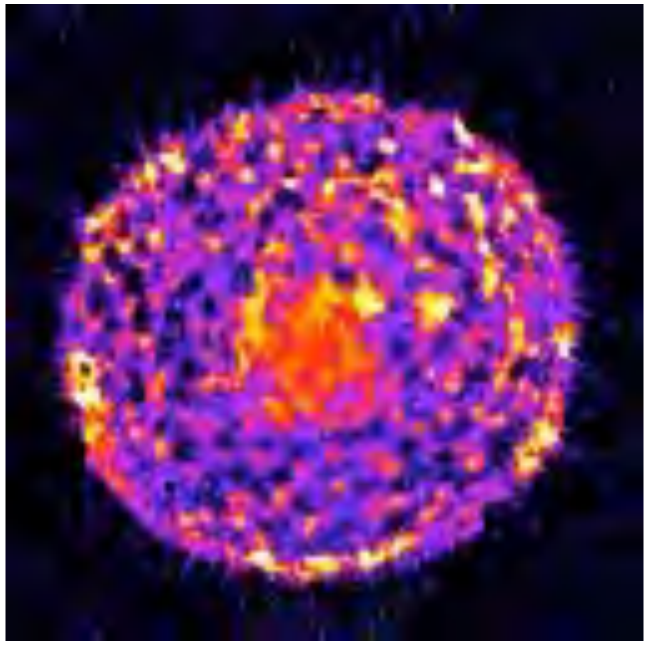
Fluorescence Tomography

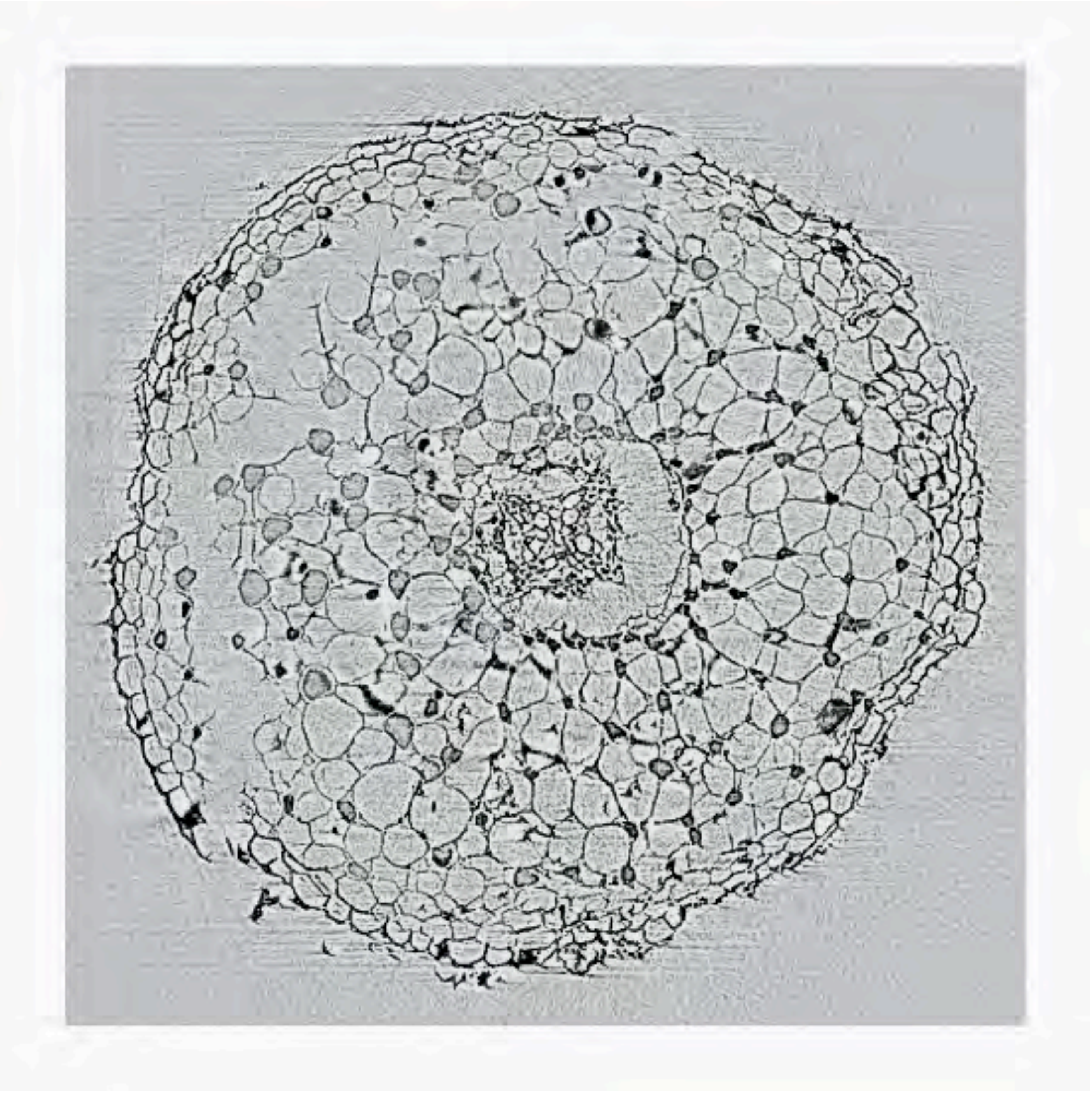
Take advantage of:

- >large penetration depth of x-rays
- >element specific contrast

Compare with structural data
from transmission tomogram:

K Kα





SAXS Tomography: Local Nanostructure

SAXS: Small-Angle X-ray Scattering

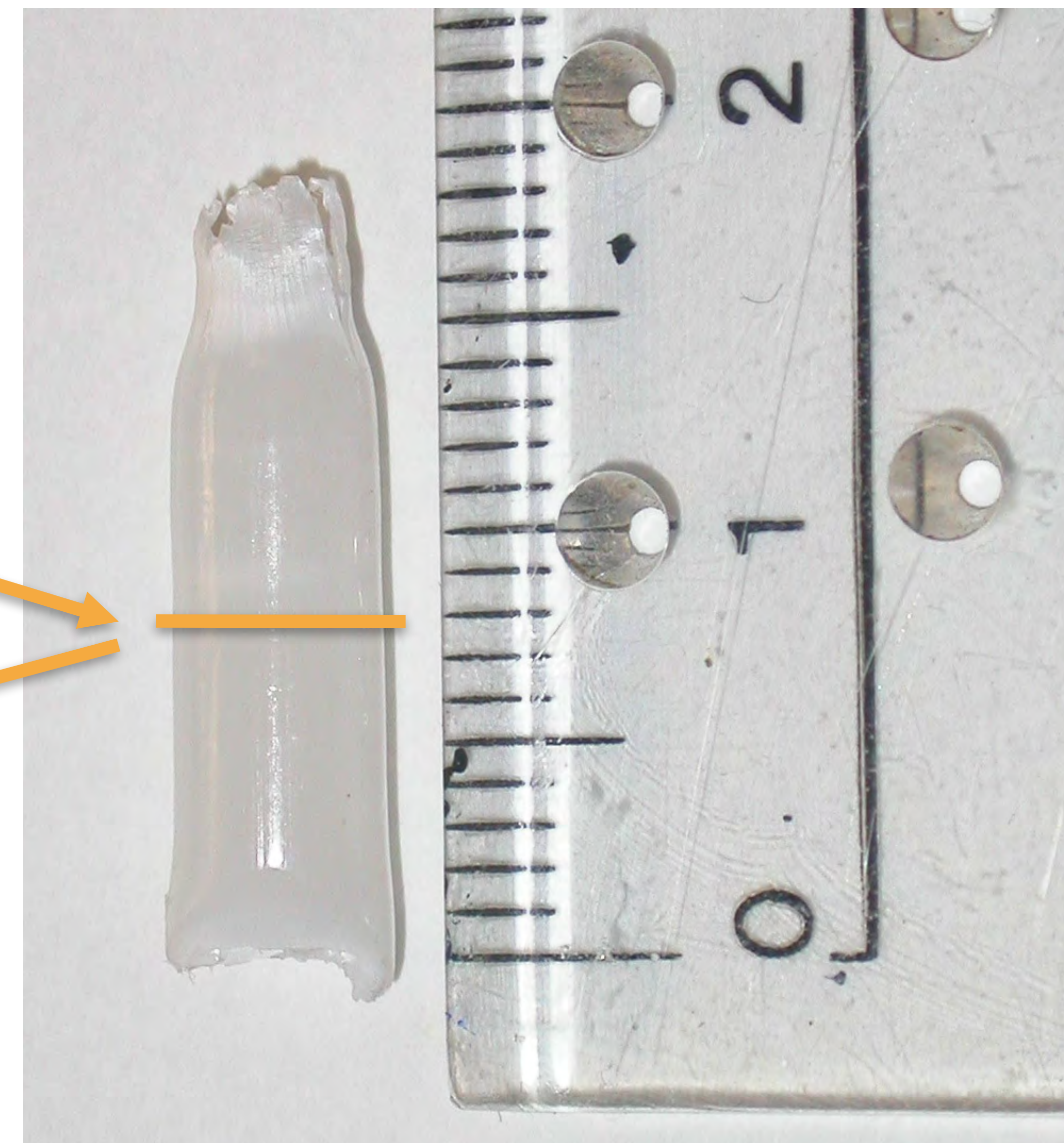
Investigating the local nanostructure on a virtual section through sample

Sample:

Non-destructive investigation
of inner structure of sample

virtual section

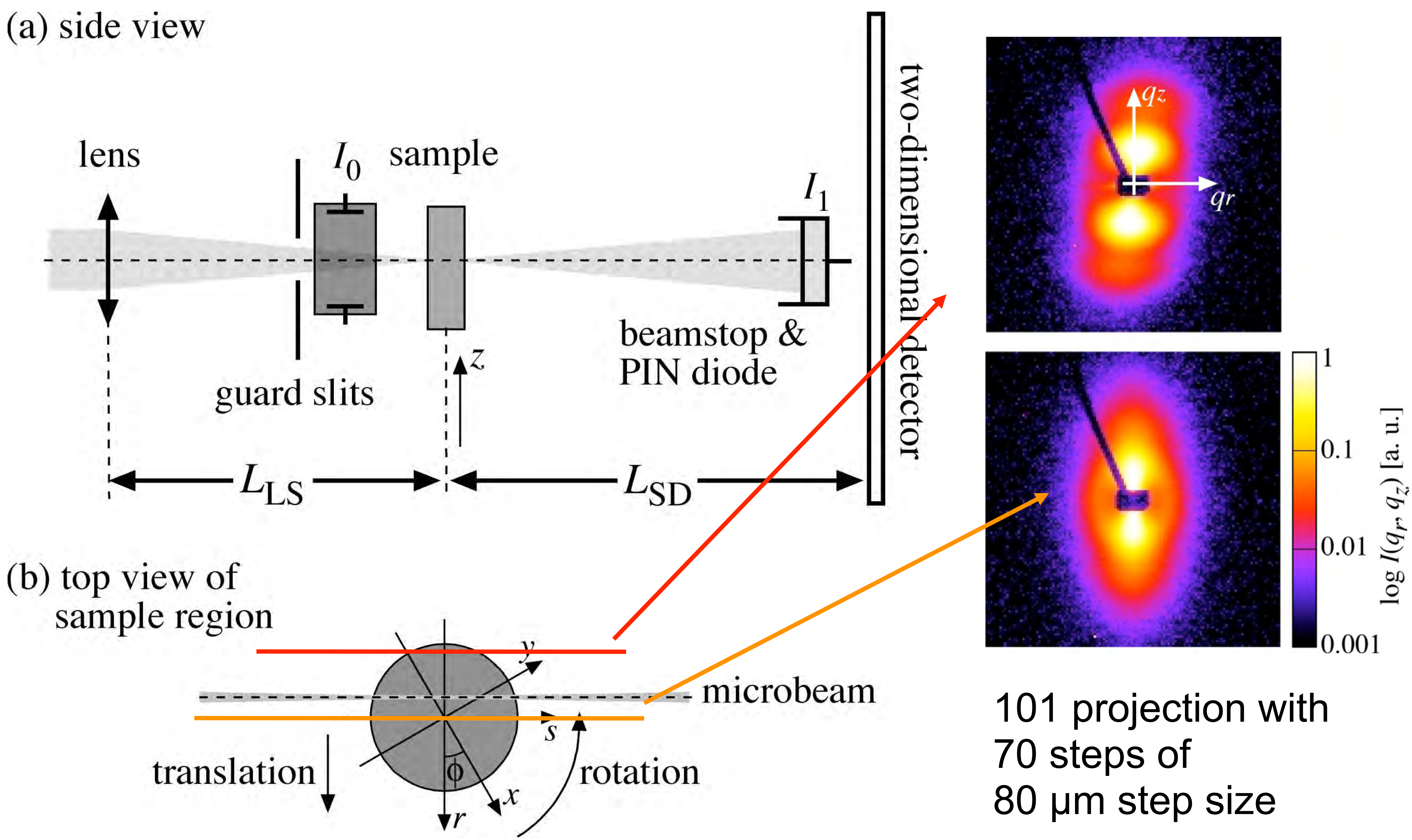
reconstructed SAXS
cross section at each point
on the virtual section



polyethylene rod

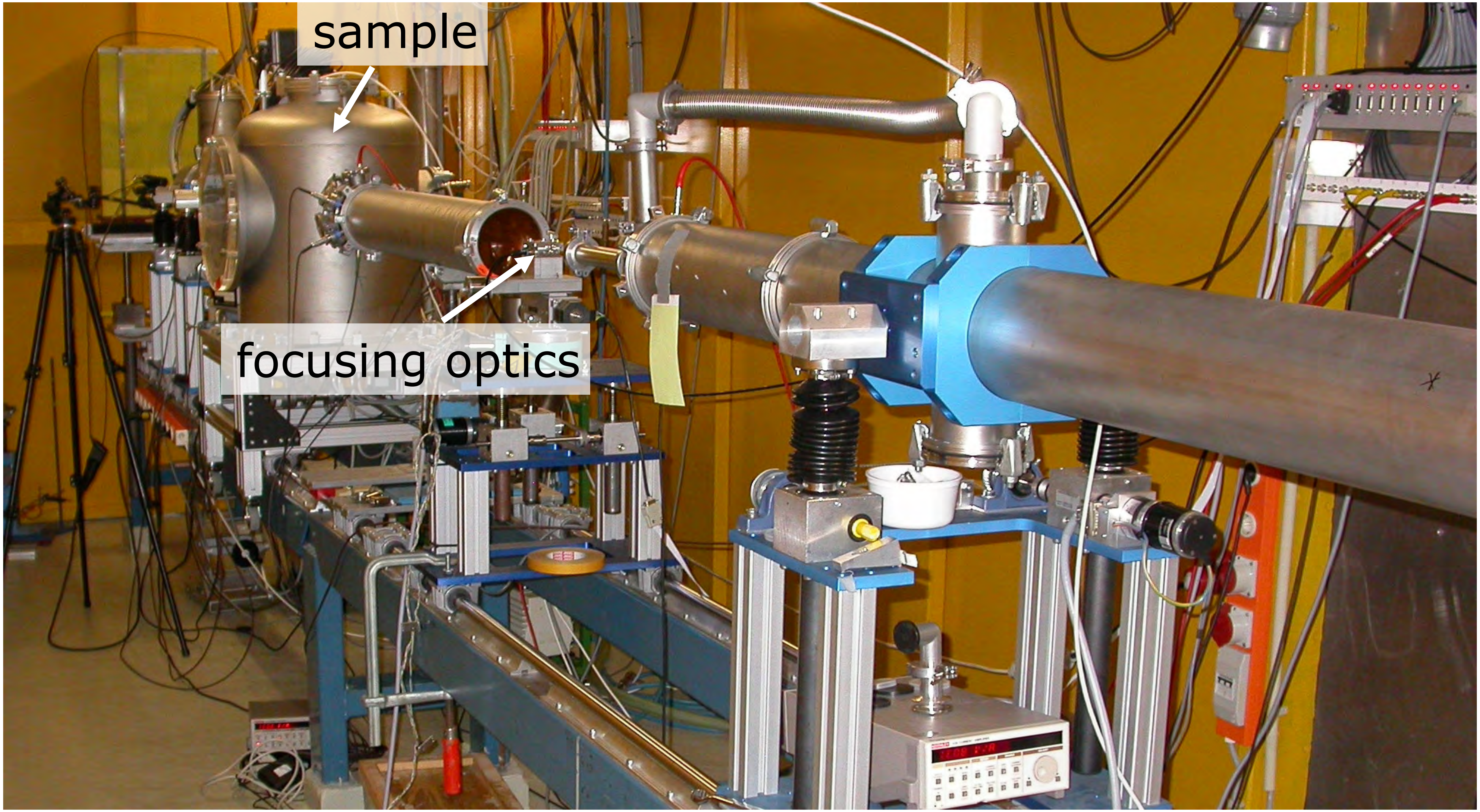
C. Schroer, et al., Appl. Phys. Lett. **88**, 164102 (2006)

Tomographic Small-Angle X-Ray Scattering

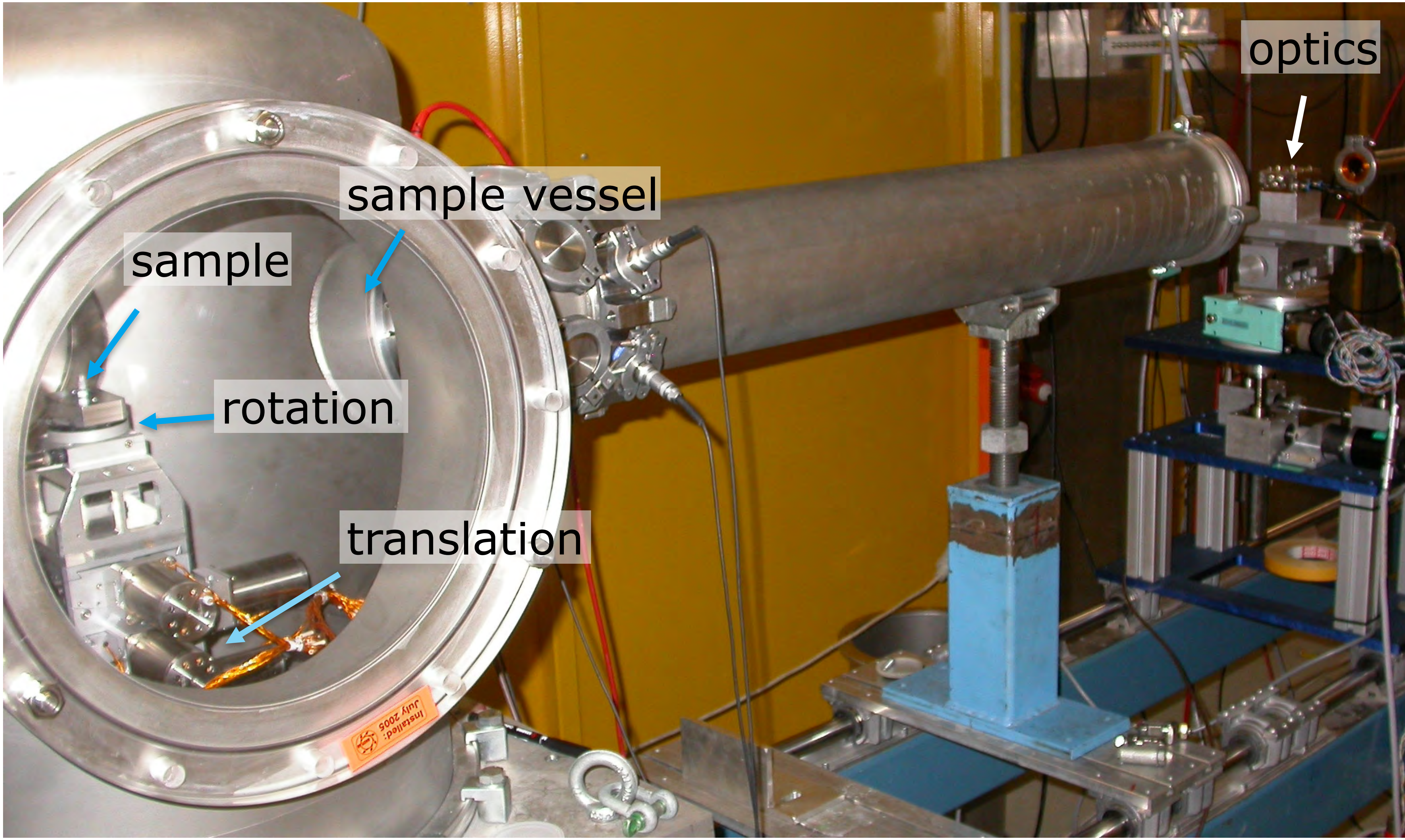


101 projection with
70 steps of
80 µm step size

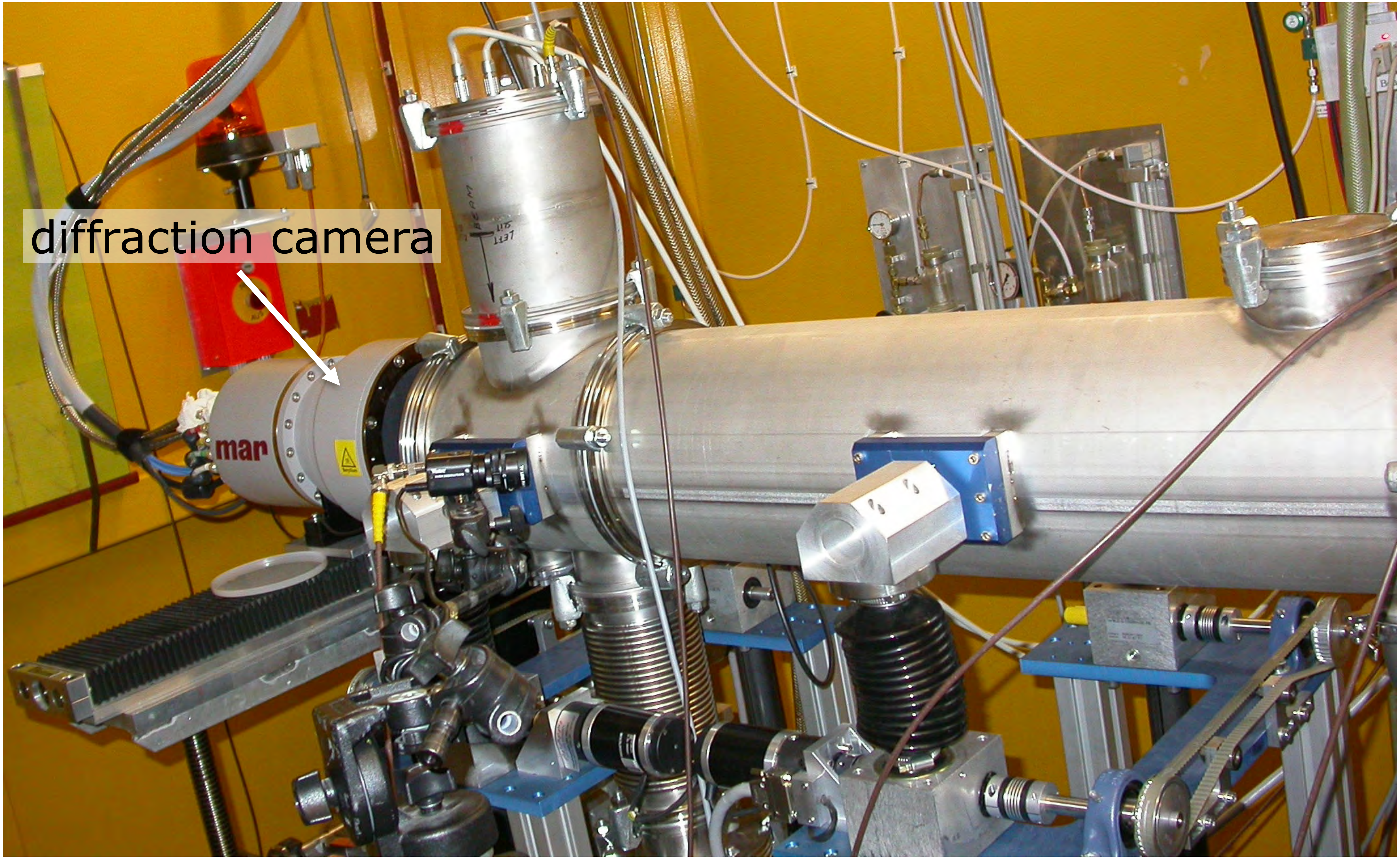
SAXS Tomography



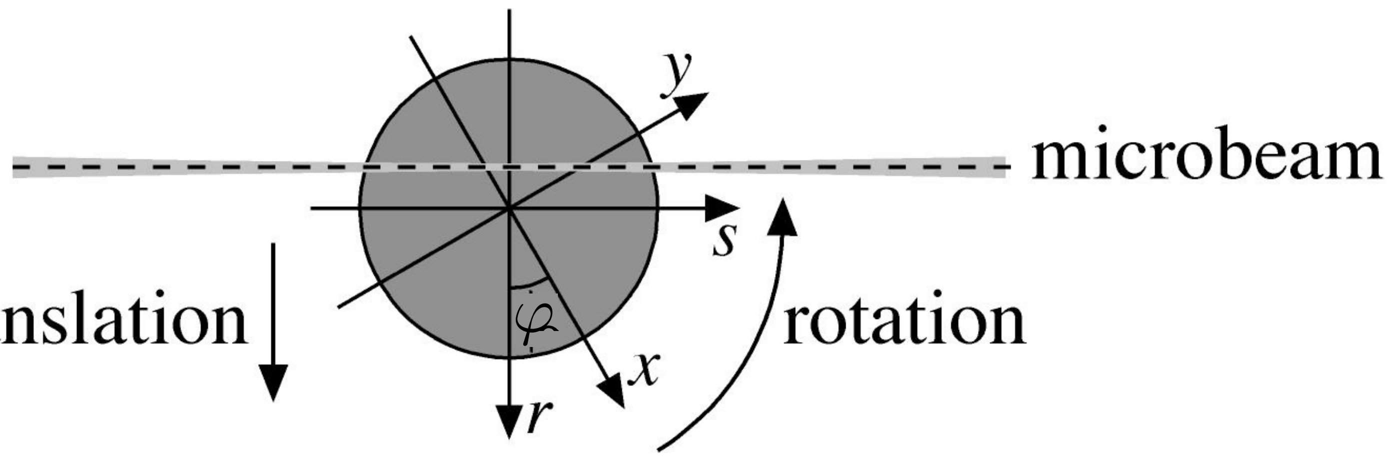
SAXS Tomography



SAXS Tomography



SAXS Tomography



Transmitted beam:

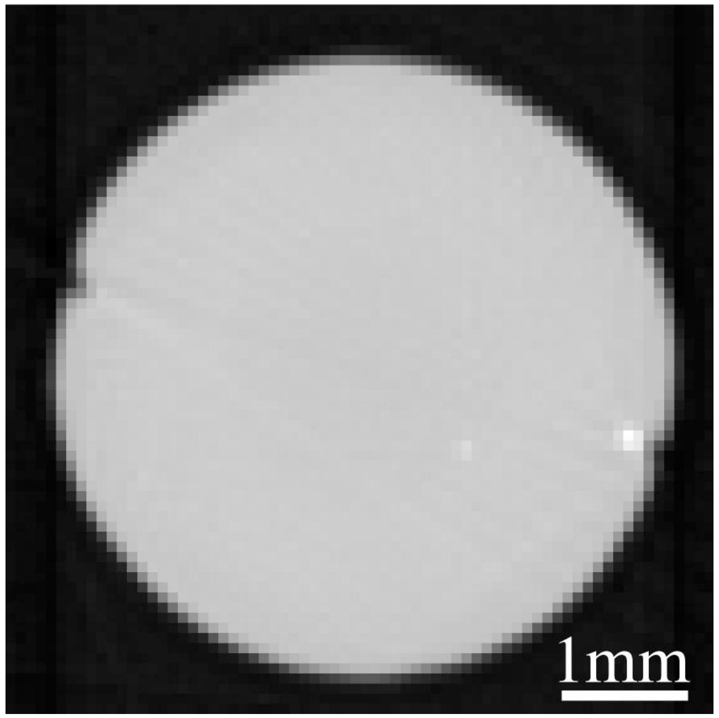
$$I_1(r,\varphi) = I_0 \exp\left\{ -\int ds'\mu\left[x(s',r),y(s',r)\right]\right\}$$

Standard tomography:

homogeneous density (polyethylene):

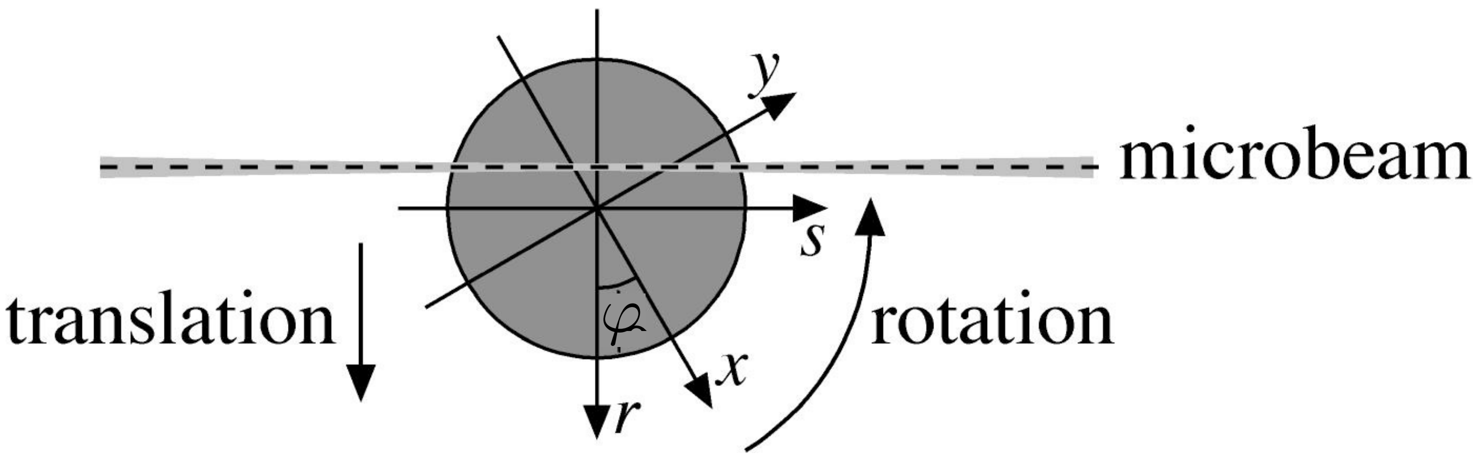
$$\rho = \left[0.88 \pm 0.04\right]\mathrm{g/cm}^3$$

attenuation



C. Schroer, et al., Appl. Phys. Lett. **88**, 164102 (2006)

SAXS Tomography



scattered signal:

$$I_{\vec{q}}(r,\varphi) = I_0 \int ds\ f(\varphi,s,r)p_{\vec{q},\varphi}(x,y)g(\varphi,s,r)$$

attenuation of primary beam:

$$f(\varphi,s,r) = \exp\left\{-\int_{-\infty}^{s} ds'\ \mu(x,y)\right\}$$

attenuation of scattered beam

$$g(\varphi,s,r) = \exp\left\{-\int_{s}^{\infty} ds'\ \mu(x,y)\right\}$$

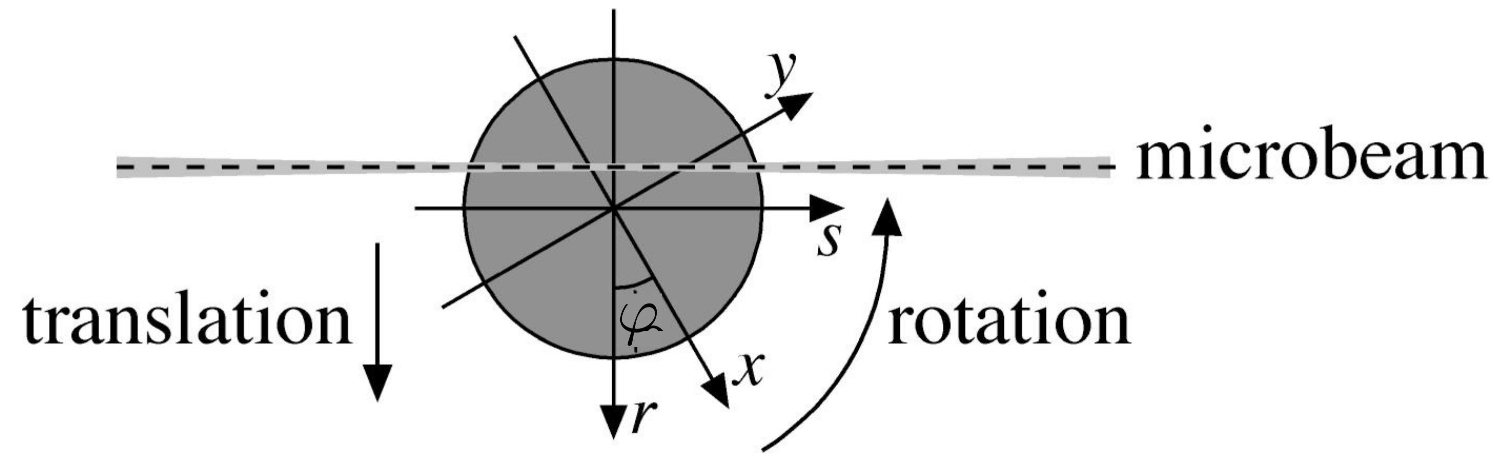
Diffraction signal in forward direction:

$$I_1(r,\varphi) = I_0(r,\varphi)\cdot f(\varphi,s,r)\cdot g(\varphi,s,r)$$

independent of *s*

C. Schroer, et al., Appl. Phys. Lett. **88**, 164102 (2006)

SAXS Tomography



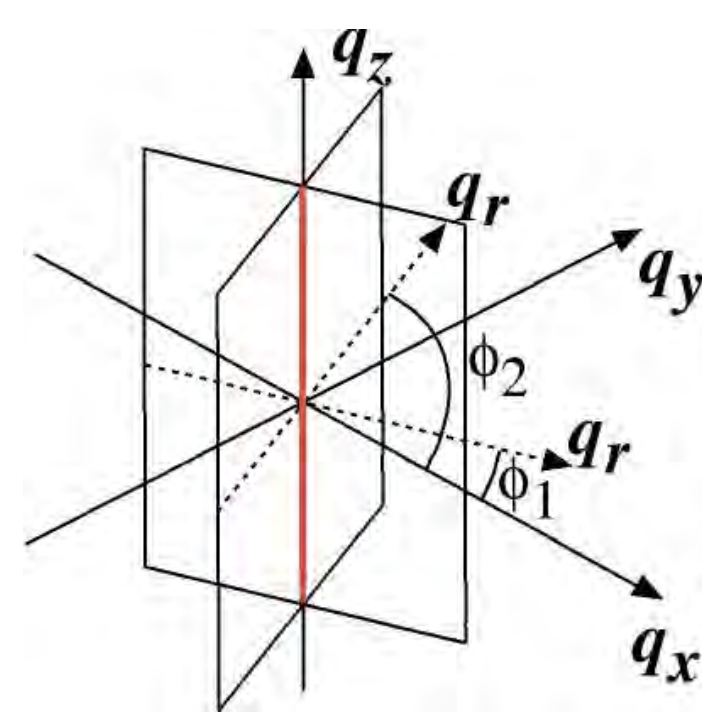
scattered signal:

$$I_{\vec{q}}(r,\varphi) = I_1 \int ds\; p_{\vec{q},\varphi}(x,y)$$

tomography works only if $p_{\vec{q},\varphi}(x,y)$ is independent φ

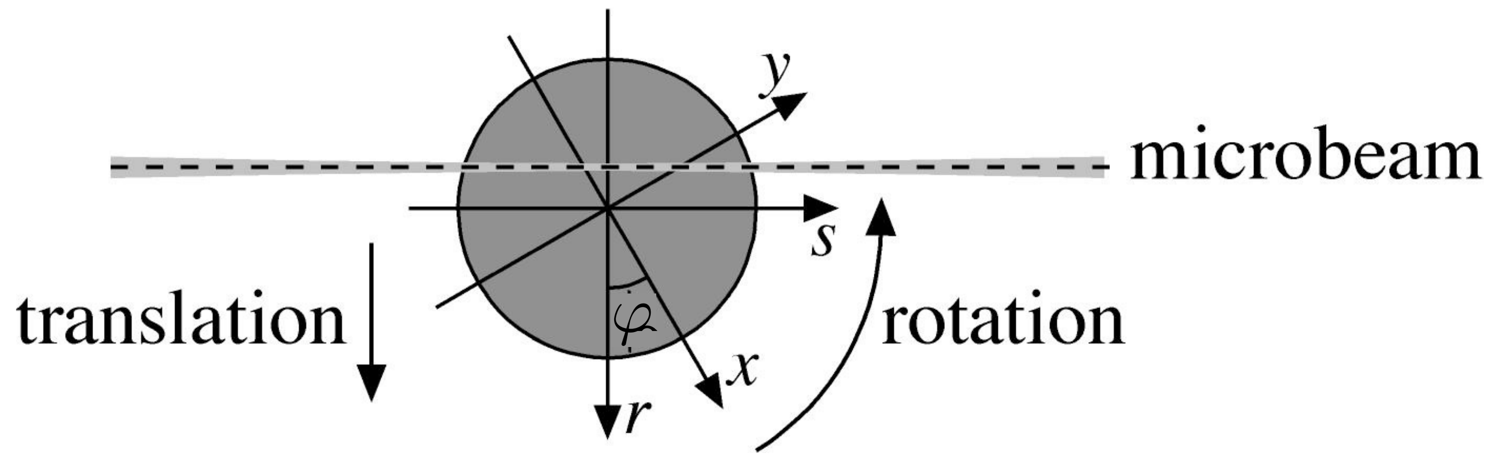
general case: $p_{\vec{q},\varphi}(x,y)$ complicated function

reconstruction only for $q_r = 0$
(q along rotation axis)



C. Schroer, et al., Appl. Phys. Lett. **88**, 164102 (2006)

SAXS Tomography



scattered signal:

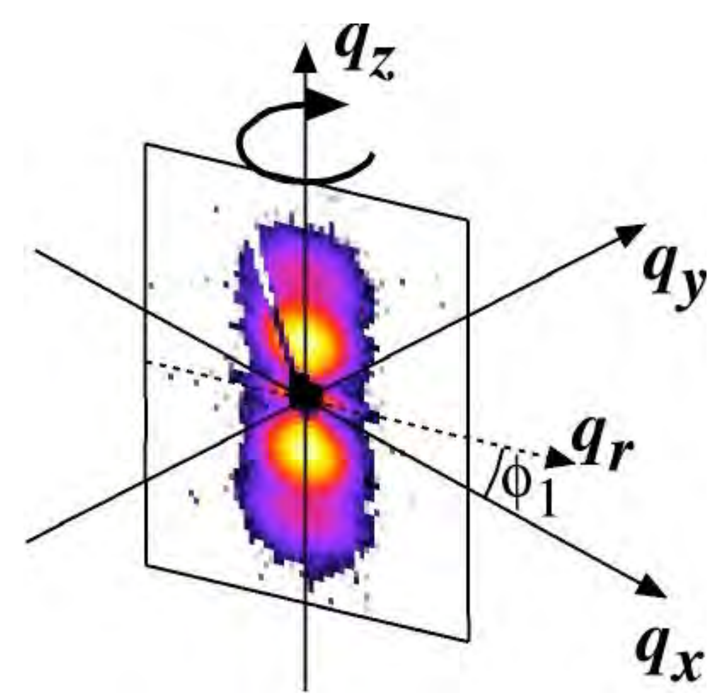
$$I_{\vec{q}}(r,\varphi) = I_1 \int ds\; p_{\vec{q},\varphi}(x,y)$$

tomography works only if $p_{\vec{q},\varphi}(x,y)$ is independent φ

special case: $p_{\vec{q},\varphi}(x,y)$ has rotation symmetry

around rotation axis

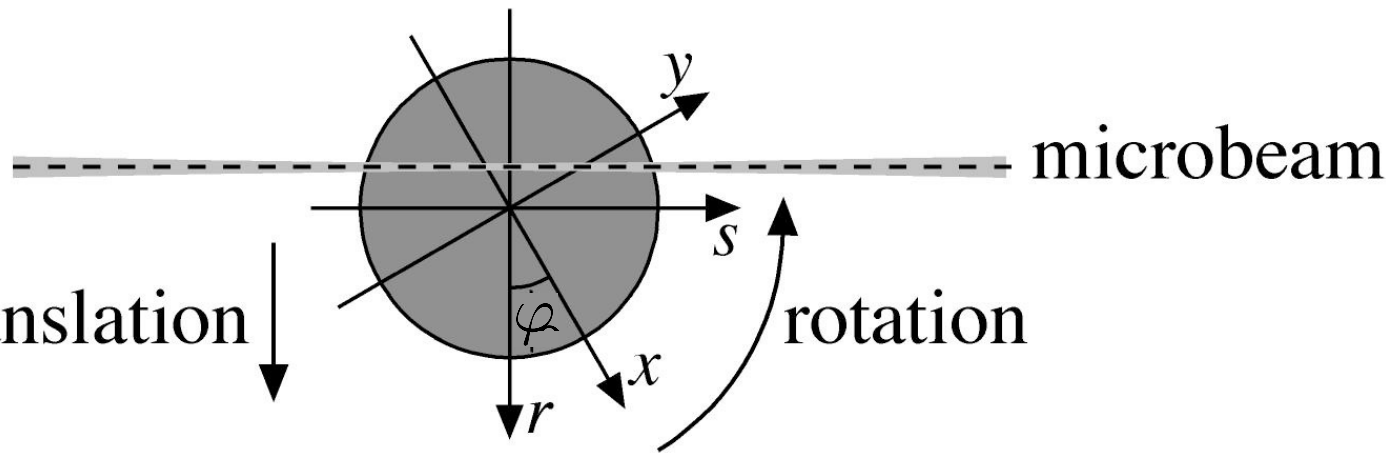
reconstruction of full SAXS cross section
in the vicinity of *q* = 0



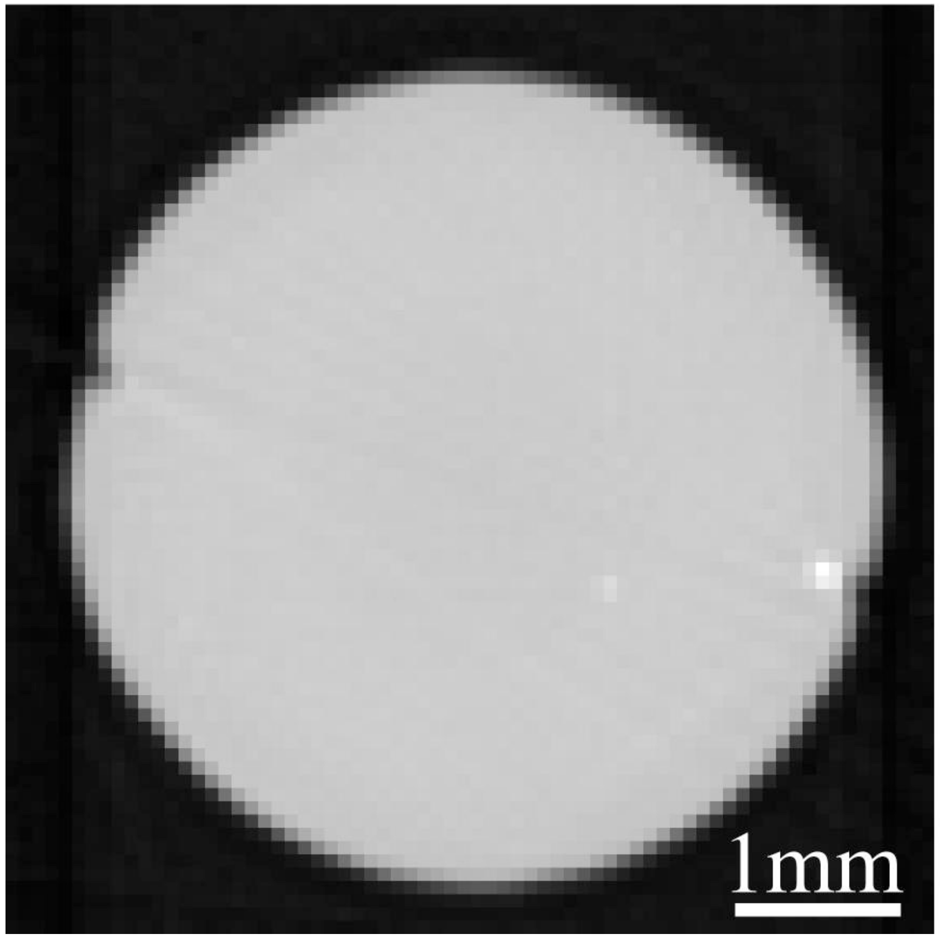
C. Schroer, et al., Appl. Phys. Lett. **88**, 164102 (2006)

SAXS Tomography

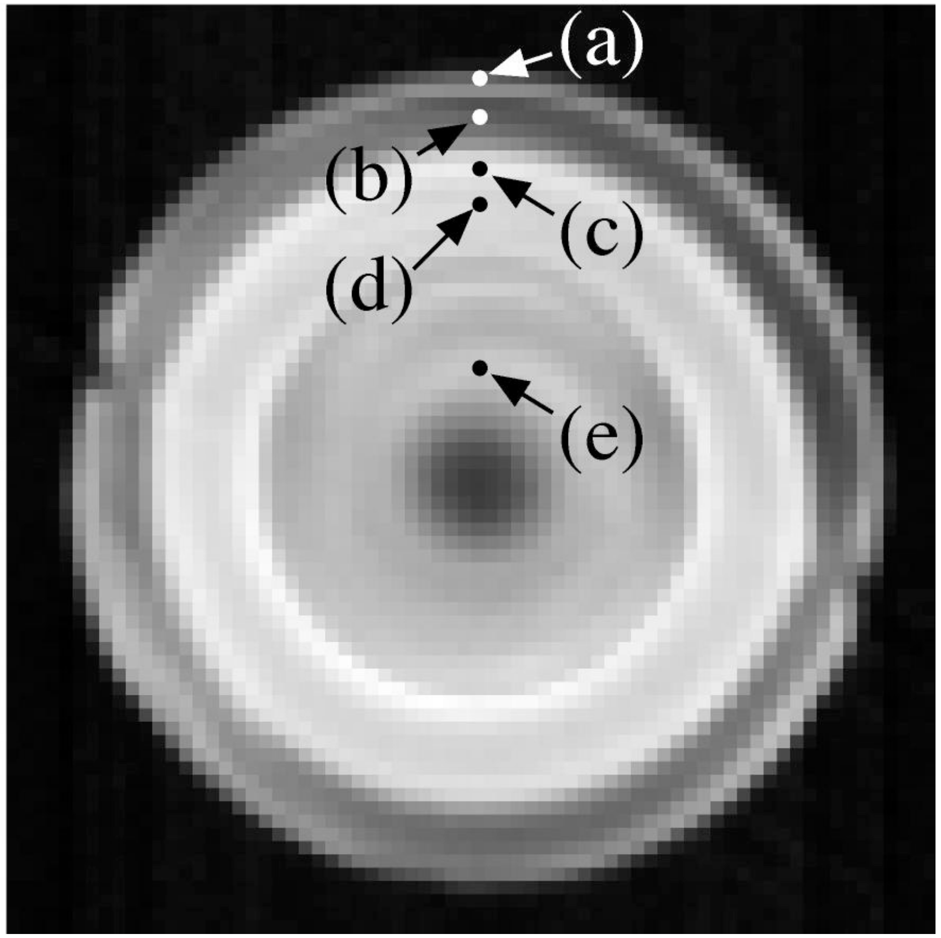
reconstruction:



attenuation



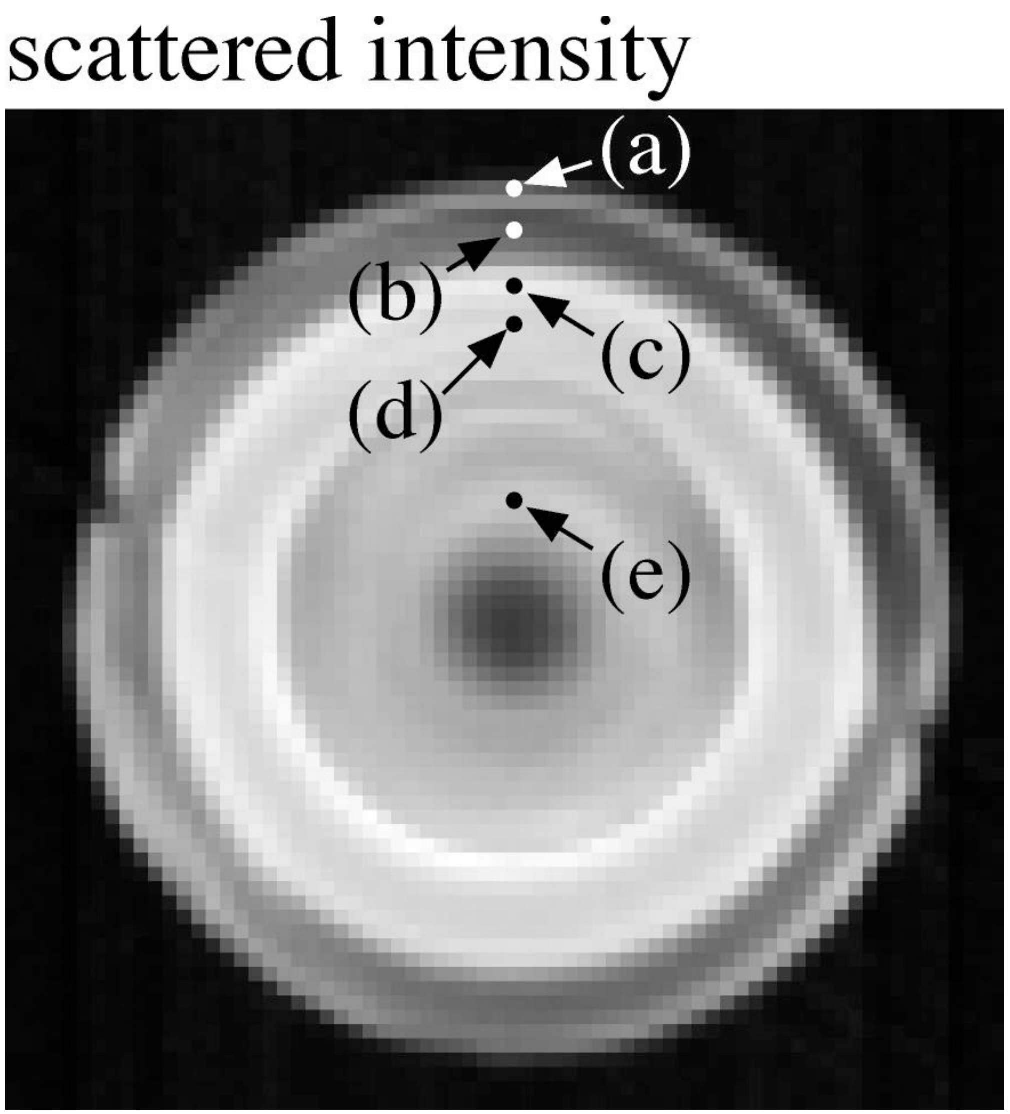
scattered intensity

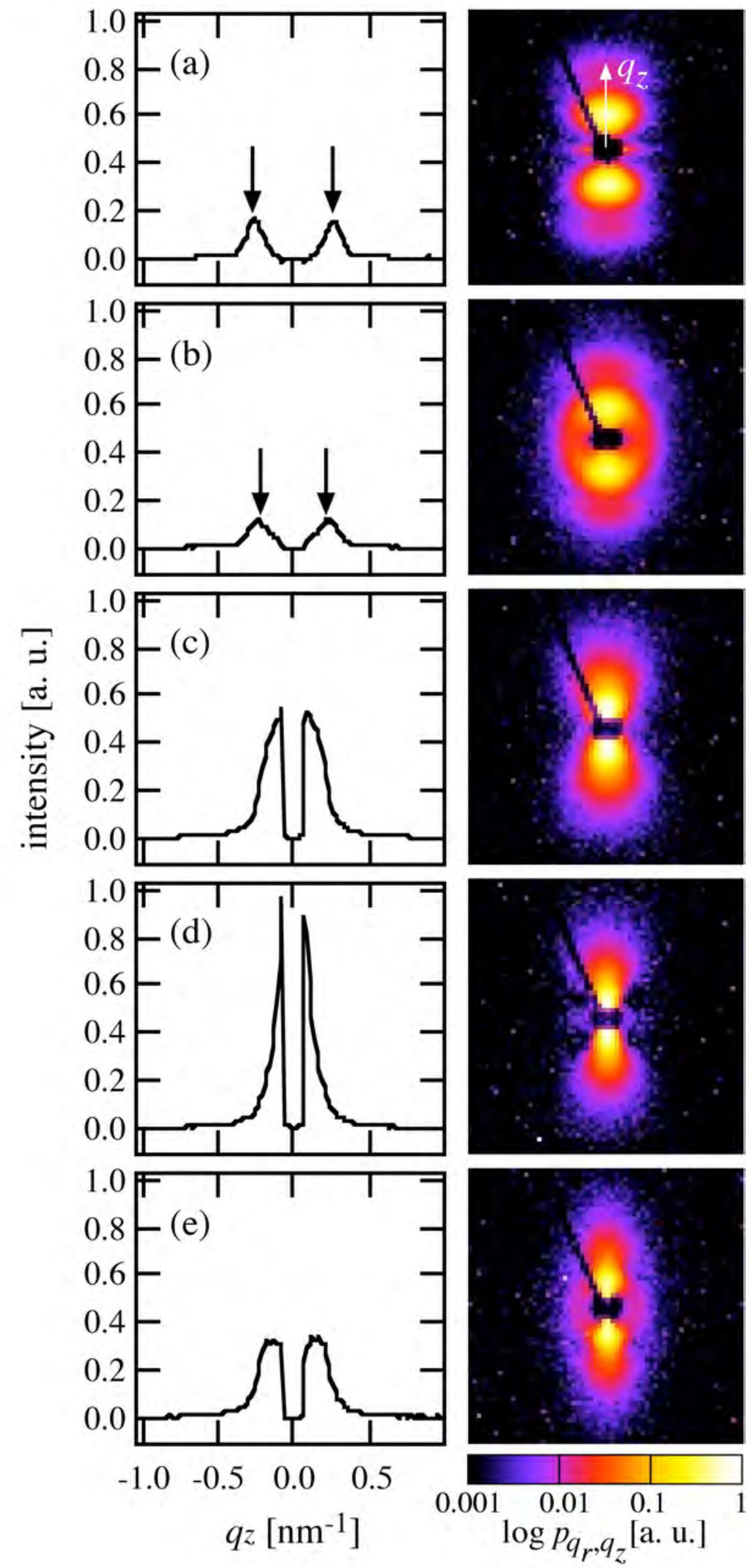


integral scattering cross section along rotation axis

SAXS Tomography

Sample with fibre texture:



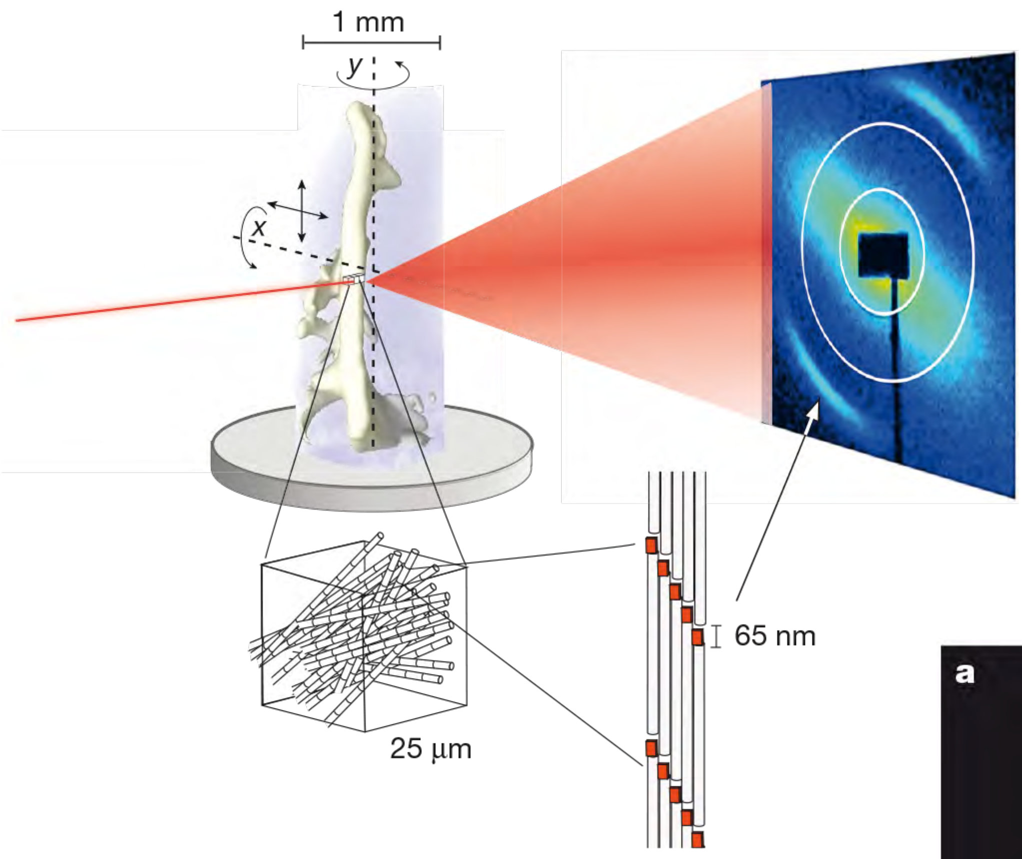


inhomogeneous nanostructure

scattering cross section in each pixel (rotation symmetry)!

C. Schroer, et al., Appl. Phys. Lett. **88**, 164102 (2006)

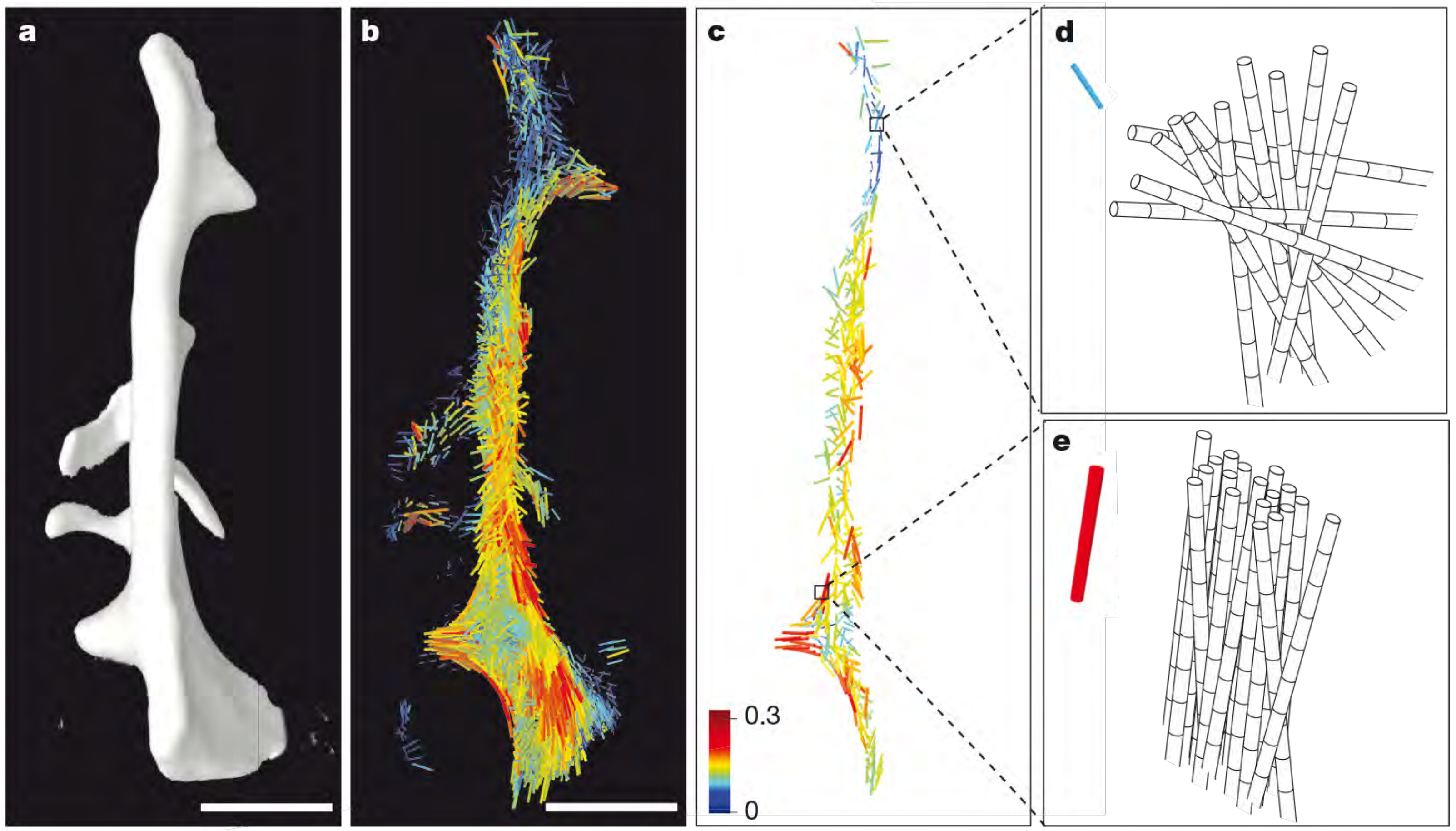
SAXS Tomography in 3D



general SAXS-tomographic oroblem

in general: measure 6 dimensional information!

Scan in 4 dimensions and record 2D patterns (coarse mesh due to time limitations)



Liebi, M., et al., Nature, **527**(7578), 349–352. (2015).

Conventional X-Ray Microscopy

X-ray microscopy as a quantitative local measurement:

- Full-field microscopy: attenuation and phase contrast

→ measure complex refractive index of sample

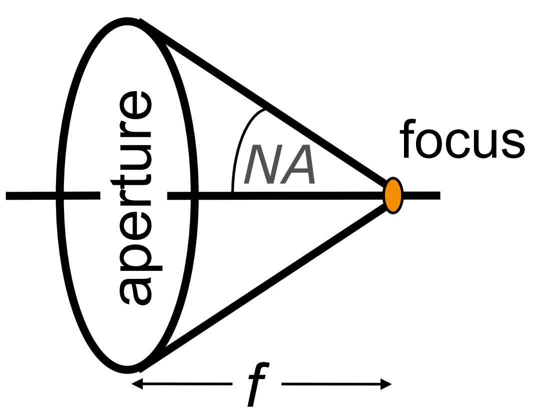
- scanning microscopy:

→ all x-ray analytical techniques can be used as contrast:

  - x-ray fluorescence (XRF): chemical composition (quantitative analysis)
  - x-ray absorption spectroscopy (XAS): chemical state of given element (e. g. oxidation)
  - x-ray diffraction and scattering (SAXS & WAXS): local nanostructure
  - …

Full-field and scanning microscopy require x-ray optics

→ resolution limited by numerical aperture of optics



Tomorrow: what are the limits and how can we overcome them?